

Composition-Dependent Growth Kinetics of Intermetallic Phases in U-Mo vs. Al Alloy Diffusion Couples Annealed at 550° and 600°C

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Nuclear Energy

Nuclear reactors will play a larger role for generating energy:

- Currently 104 reactors provide
 ~20% of electricity in U.S.
 - This equals what is used by California, Texas, and New York.
 - Over 430 reactors worldwide (provides 17% of electricity).



Help to reduce emissions of greenhouse gases.

Currently, more advanced reactors are being developed for a variety of applications:

- GEN IV reactors are being developed to provide energy efficiently, cost-effectively, and with minimal waste generation.
- NGNP is being developed to provide process heat to support hydrogen production.

Space reactors are being developed to power space craft and provide power at remote space locations (e.g., Mars).

Nuclear Energy and Advanced Fuels

To support development of these different types of reactors, need fuels that will withstand more aggressive reactor conditions.

This includes higher temperatures, higher burn-ups (heavy metals fissioned), etc.

Many types of fuel materials are being investigated:

Ceramics, metallic alloys, molten salts, ceramic-ceramic composites, metal-matrix composites, etc.

Fuel Performance and Challenges:

 Swelling: The accumulation of two fission product atoms for each atom fissioned. This is aggravated by the fact that some of the fission products are gases.

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Melting

Thermal conductivity / Thermotransport

Fuel restructuring

Fuel/Cladding Interaction (FCI)

Generation IV Reactors

- Six reactor concepts are being evaluated.
 - Lead-cooled fast reactor
 - Molten salt reactor
 - Sodium-cooled fast reactor
 - Supercritical water cooled reactor
 - **Gas Fast Reactor**
 - **High Temperature Gas Reactor**



Early Prototype Commercial Power



Generation I

Reactors



1950

1960



- LWR-PWR, BWR

1980

1990

- CANDU - AGR

1970

Generation II

LWRs

Generation III

Advanced



- ABWR - System 80+

Gen III

2010

2000



Resistant

Gen IV

2030

Sodium fast reactors can utilize almost all the energy in natural uranium versus the 1% utilized in thermal spectrum systems.



Gen III+

2020



Next Generation Nuclear Plant (NGNP)

- Prismatic block design or pebble bed design.
- Uses He coolant. Refractory coated fuel.
 - Graphite moderator
 - High temperature stability.
 - Provide process heat to supply a hydrogen production plant.





- 360,000 "pebbles" in core.
 - 350 discharged daily.
 - One pebble discharged every 30 seconds.
 - ✓ Average pebble cycles through core 15 times.

Stand Pipe 7m(23 ft) Control Rod Drive Assembly Cold leg Core Coolant Upper Plenum Control Rod Guide tubes Upper Core Restraint Upper Plenam Structure Shroud Central Reflector 8.2m(27ft) Dia Graphite Vessel Flange Control Rods Reactor Vessel Annular shaped 23.7m(78ft) Active Core Cross Vessel Nipple Outer Side Reflector Graphite Hot Duct Structural Core Exit Hot Gas Element Plenum Core Inlet Floy Graphite Core Support Columns Core Insulation Layer for Metallic Outlet Core Support Plate Flow 2.2m(7ft) Shutdown Cooling Hot Duct System Module Insulation Module Metallic Core Support Structure

Refueling

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NGNP Fuel Particle in Pebble Bed Design







The layers that surround fuel kernel include an inner pyrolytic carbon (IPyC) layer, a SiC layer, and an outer pyrocarbon (OPyC) layers.

- Layers act as pressure vessel for fission product gases.
- Barrier to migration of other fission products.

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Space Reactors

Radio-isotope Thermal Generator (RTG) employed for "New Horizons" mission to Pluto. **V** Use natural decay of Pu238 to provide heat and electricity. **Ultimately reactors** will be used to provide energy in space.



Generation IV Reactor Fuel and Cladding

- Experimental Breeder Reactor-I (EBR-I) and EBR-II were successfully operated using metallic fuels (at INL).
 - The fuels used were primarily U-base (e.g., U-Zr, U-Pu-Zr) alloys.
 The reactors demonstrated a high degree of safety, a compact design, and relatively small amounts of waste generation.
- The maximum fuel operating temperature allowed for a fuel/cladding interface temperature of 650°C.
- Multicomponent, multiphase diffusion is commonly observed in irradiated metallic nuclear fuels.
 - This diffusion occurs in the presence of temperature gradients, power gradients, the generation of fission products, fission density gradients, etc.
 - The resulting phases that form affect the performance of irradiated nuclear fuels.

Eventually the fuel will swell and contact the cladding.
 Interdiffusion occurs between the constituents in the cladding and fuel, along with generated fission products.

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Example of Fuel/Cladding Interaction in An Irradiated U-Pu-Zr Alloy Fuel

Fuel

Cladding

50 un

Fission Products

U-16Pu-23Zr, D9 cladding, 11.3 at% Burnup

Swelling and interdiffusion between the fuel and cladding alloy constituents results in the formation of reaction zones at the inner surface of the cladding. These zones are brittle and can form cracks.

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Machined U-10Mo, ~8 g-U/cm³

Atomized U-10Mo, ~8 g-U/cm³

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Further Consideration in Fuel Development: Low-Enriched Uranium-base Fuels

- **Dispersion Fuel Fabrication Process:**
 - Fuel powder production (powder size 20 to 125µm)
 - Grinding/Machining/Communition (ANL, AECL)
 - Atomization (KAERI)
 - Fuel-matrix powder blending
 - Press fuel-matrix blend into a compact
 - Insert compact into aluminum frame and weld to assemble

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- Hot roll assembly to final thickness, ~1.25-mm
- Blister anneal to verify aluminum bonding
- Radiograph to locate fuel zone
- Shear to final plate length and width

Low-Enriched Uranium-base Fuels: U-Mo Alloys

U-10Mo fuel irradiated in RERTR-2 ✓ Low temperature, < 100°C ✓ High burnup, 70% U²³⁵



Powder

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50 micron

- Interaction of fuel and cladding in irradiated fuel produces phases that can potentially contribute to swelling of fuel plates:
 - Al-rich phase (U,Mo)Al, may contribute to increased swelling.



Objectives

To examine the growth and composition of the intermetallic compound layers that develop in U-Mo/AI system to improve the performance and service life of U-Mo/AI dispersion fuels.

Examine the interdiffusion behavior between U-Mo alloys and Al-alloys with specific Si alloying additions.

Findings and understanding from this program will provide strategies and solutions to minimize the interdiffusion-induced degradation of U-Mo alloys.

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Experimental Details







Cladding

Claddinc

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- Solid-to-solid diffusion couple alloys were sectioned, polished and assembled under a controlled Ar atmosphere in a glove box.
- Diffusion couples were wrapped in Ta foil, and encapsulated in quartz capsule in Ar atmosphere after Argon flush for heat treatment.
- Diffusion anneal performed using a Lindberg/Blue 3-Zone horizontal tube furnace.
- Diffusion structures examined by optical and scanning electron microscopy.
 - Concentration profiles determined by Electron Probe Microanalysis (EPMA) using pure standards and ZAF correction.

Experimental Details

Diffusion couples were assembled using U-7Mo, U-10Mo and U-12Mo with pure AI (99.999%), AI-2Si and AI-5Si (wt.%) and commercially available 4043AI and 6061AI alloys.

Alloys cast at INL (U-alloys) and ANL (AI-Si alloys).

Homogenization at UCF.

The couples were heat-treated in Ar-atmosphere for 24 hours at 550°C and 600°C. Anneal at 500° and 300°C will be carried out.

The Matano plane was approximated from experimental concentration profiles.

The interdiffusion flux was calculated directly from the concentration profiles.

The integrated interdiffusion coefficients have been determined for the major components.

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Interdiffusion Flux and Integrated Interdiffusion Coefficients

Interdiffusion fluxes of all components can be determined directly from their concentration profiles without the need of the interdiffusion coefficients:

$$\tilde{J}_{i} = \frac{1}{2t} \int_{C_{i}^{-} \text{ or } C_{i}^{+}}^{C_{i}(x)} dC_{i} \quad (i = 1, 2, ..., n)$$

C(x)

Integrated interdiffusion coefficients for a component *i* on either side of the Matano plane can be defined as:

$$D_{i,L}^{int} = \int_{-\infty}^{x_o} \widetilde{J}_i(x) dx$$
 and $D_{i,R}^{int} = \int_{x_o}^{+\infty} \widetilde{J}_i(x) dx$

The Total Integrated interdiffusion coefficients for a component *i* over the entire concentration profile:

$$D_{i,Tot}^{int} = \int_{-\infty}^{-\infty} \widetilde{J}_i(x) dx$$

M. A. Dayananda, Y.H. Sohn, Scripta Mater., 35 (1996) 683.

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U-12Mo vs. AI-2Si Diffusion Couple (550°C for 24hr)



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Observed Behavior of Si in the Interdiffusion Zone



 For diffusion couples with 4043, AI-2Si and AI-5Si Alloys:

- Si build-up in the interdiffusion zone near the Al-alloy/Intermetallic interface for couples with U-7Mo alloys.
- Si build-up in in the interdiffusion zone near U-Mo/Intermetallic interface for couples with U-10Mo and U-12Mo alloys.



Integrated Interdiffusion Coefficients

550°C for 24 Ho

600°C for 24 Hours

Diffusion Couple	D ^{int} _{<i>i</i>,Tot} 10 ⁻¹⁴		
	Al	Мо	U
U-7Mo vs. Al	5.97	-1.0	-4.9
U-7Mo vs. 6061AI	49.9	-7.4	-40.2
U-10Mo vs. Al	13.5	-2.8	-10.5
U-10Mo vs. 6061AI	161.0	-34.1	-126.0
U-12Mo vs. Al	3.48	-0.9	-2.53
U-12Mo vs. 6061AI	75.0	-6.16	-17.3

 Integrated interdiffusion coefficients offer a measure of the accumulated interdiffusion flux and its direction for a component in diffusion couples.

Diffusion coefficients in atf•m²/sec.

Diffusion	$D_{i,Tot}^{\text{int}}$ 10-16		
Couple	Al	Мо	U
U-7Mo vs. Al	*	*	*
U-7Mo vs. 6061AI	3535.8	662.5	-4188.2
U-7Mo vs. 4043AI	4.6	-0.7	
U-7Mo vs. AI-2Si	7.1	-1.3	-5.5
U-7Mo vs. Al-5Si	27.5	-3.4	-12.4
U-10Mo vs. Al	*	*	*
U-10Mo vs. 6061AI	1473.3	-321.9	-1039.4
U-10Mo vs. 4043AI	*	*	*
U-10Mo vs. Al-2Si	*	*	*
U-10Mo vs. Al-5Si	56.0	-7.8	-29.7
U-12Mo vs. Al	/ 1555.0	-419.4	-1106.5
U-12Mo vs. 6061AI	1004.1	-109.1	-321.2
U-12Mo vs. 4043AI	5.9	-1.9	-5.8
U-12Mo vs. Al-2Si	49.5	-10.4	-29.5
U-12Mo vs. Al-5Si	18.5	-3.34	-9.5
* Analysis in Progress		John h	1000

Interdiffusion Zone Developed Thickness

600°C for 24 Hours

Diffusion Couple	Average Thickness (µm)
U-7Mo vs. 6061AI	658
U-7Mo vs. Al	275
U-10Mo vs. 6061AI	1550
U-10Mo vs. Al	405
U-1 <mark>2Mo vs</mark> . 6061AI	925
U-12Mo vs. Al	223

 Diffusion couples with 6061 developed large interdiffusion zones.

Diffusion couples with Si containing Alloys developed interdiffusion zones roughly one order of magnitude smaller than those with 6061.

550°C for 24 Hours

		Average	
	Diffusion Couple	Thickness	
4		(µm)	
2	U-7Mo vs. 4043AI	24	
	U-7Mo vs. 6061AI	741	
	U-7Mo vs. Al	*	
	U-7Mo vs. Al-2Si	24	
ł	U-7Mo vs. Al-5Si	22	
5	U-10Mo vs. 4043AI	*	
-	U-10Mo vs. 6061AI	400	
1	U-10Mo vs. Al	*	
4	U-10Mo vs. Al-2Si	24	
	U-10Mo vs. Al-5Si	89	
	U-12Mo vs. 4043AI	28	
Ì	U-12Mo vs. 6061AI	467	
1	U-12Mo vs. Al	540	
	U-12Mo vs. Al-2Si	65	
	U-12Mo vs. AI-5Si	45	
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* Analysis in Progress

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Composition Dependent Growth Kinetics of UAI₄ Intermetallic Phase at 550°C

The thickness of intermetallic layer increases with increasing concentration of Mo in the UMo alloy.

AI-2Si

U-7Mo

(U,Mo)(Al,Si)

50µm





Phase Identification of AI-Rich Intermetallic Phases in U-Mo-AI System



Summary

Introduction of Si into the Al Alloy in the diffusion couples appears to effectively reduce the growth rate of the intermetallic layers in diffusion couple studies.

The composition of the intermetallic phase in all diffusion treated at 550°C and 600°C couples maintain the UAl₄ with minor solid solutioning, but with some build-up near interfaces.

Growth rate of the UAl₄ intermetallic layer increases slightly with Mo content and decreases rapidly with Si content.

 Diffusion couples with Si in the Al-alloy exhibits a build-up of Si within the intermetallic layer in the interdiffusion zone.

Current Work in Progress

Diffusion anneal at 500°C and 300°C.

Determination of activation energy of interdiffusion based on integrated interdiffusion coefficients.

 Detailed phase identification within diffusion couples using TEM via FIB-INLO with particular emphasis on precipitates near interfaces.

Other alloying additions into fuel or cladding or as coatings: ✓ U-Mo-X (X = Si, Ti, Zr, Nb, etc.) ✓ Diffusion barrier coatings

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