

Quasi-isentropic compression
by ablative laser loading
and on Be at Z

Damian C. Swift, P-24, LANL

With thanks to:

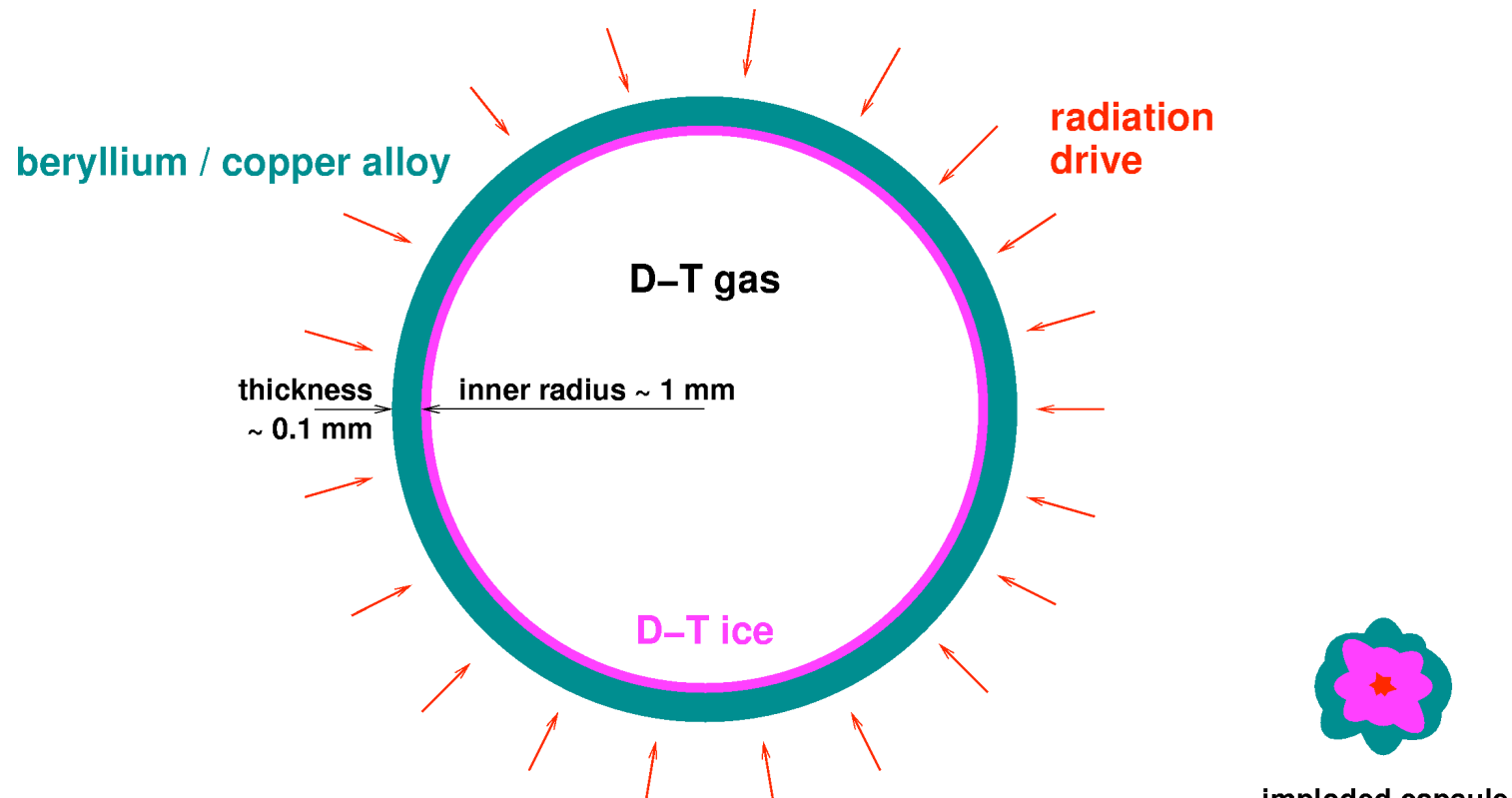
Randall P. Johnson, Thomas E. Tierney, Sheng-Nian Luo,
Scott R. Greenfield, Aaron C. Koskelo, Kenneth J. McClellan
(LANL),

Marcus D. Knudson (SNL), Pedro P. Peralta (ASU)

Outline

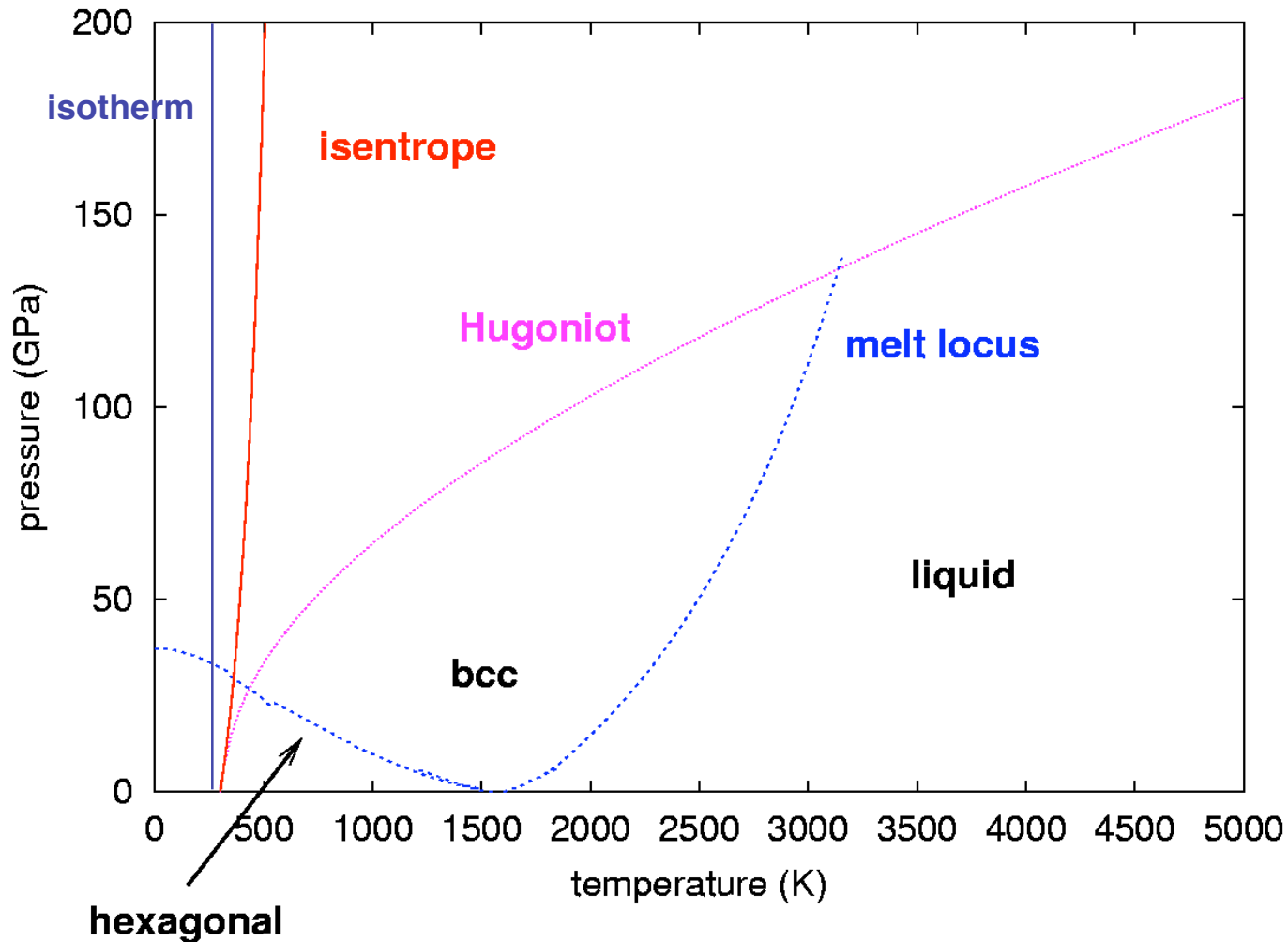
- Motivations:
 - Physics-based material dynamics
 - Capsule properties for ICF
- Z ICE experiments on Be
- ICE by laser ablation:
 - Ablative loading: method and modeling
 - LICE at TRIDENT
 - Example data
- Diagnostics

Beryllium capsule design for ICF



- Variations in acceleration cause asymmetry => reduce yield
- Beryllium: polycrystalline, anisotropic

Be phase diagram



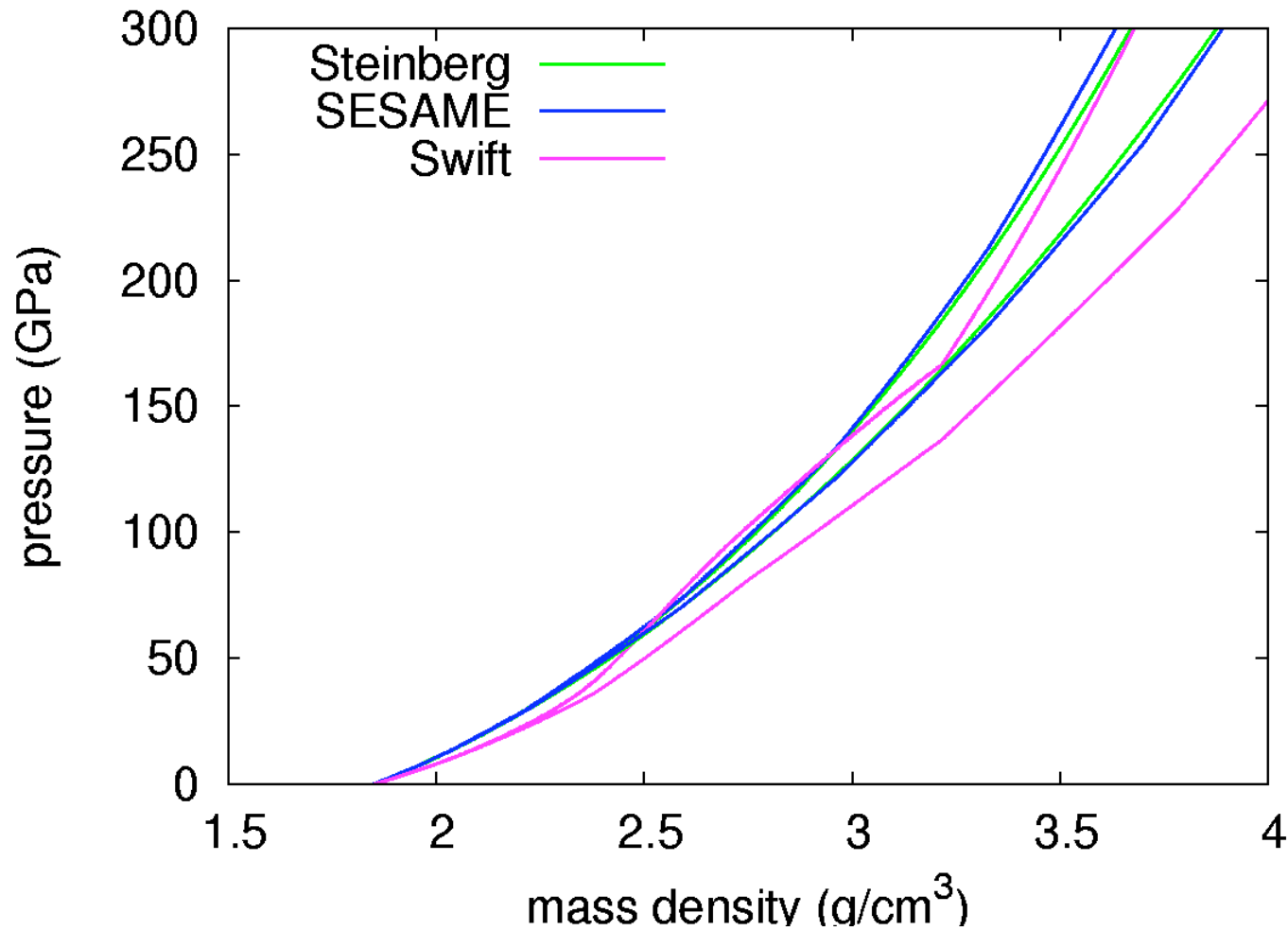
Temperatures are poorly-known (shown: Steinberg)

Isotherm: static loading e.g. diamond anvil cell.

Isentrope: quasistatic, $-p.dv$ work, no heat loss.

Hugoniot: kinetic energy to heat to entropy.

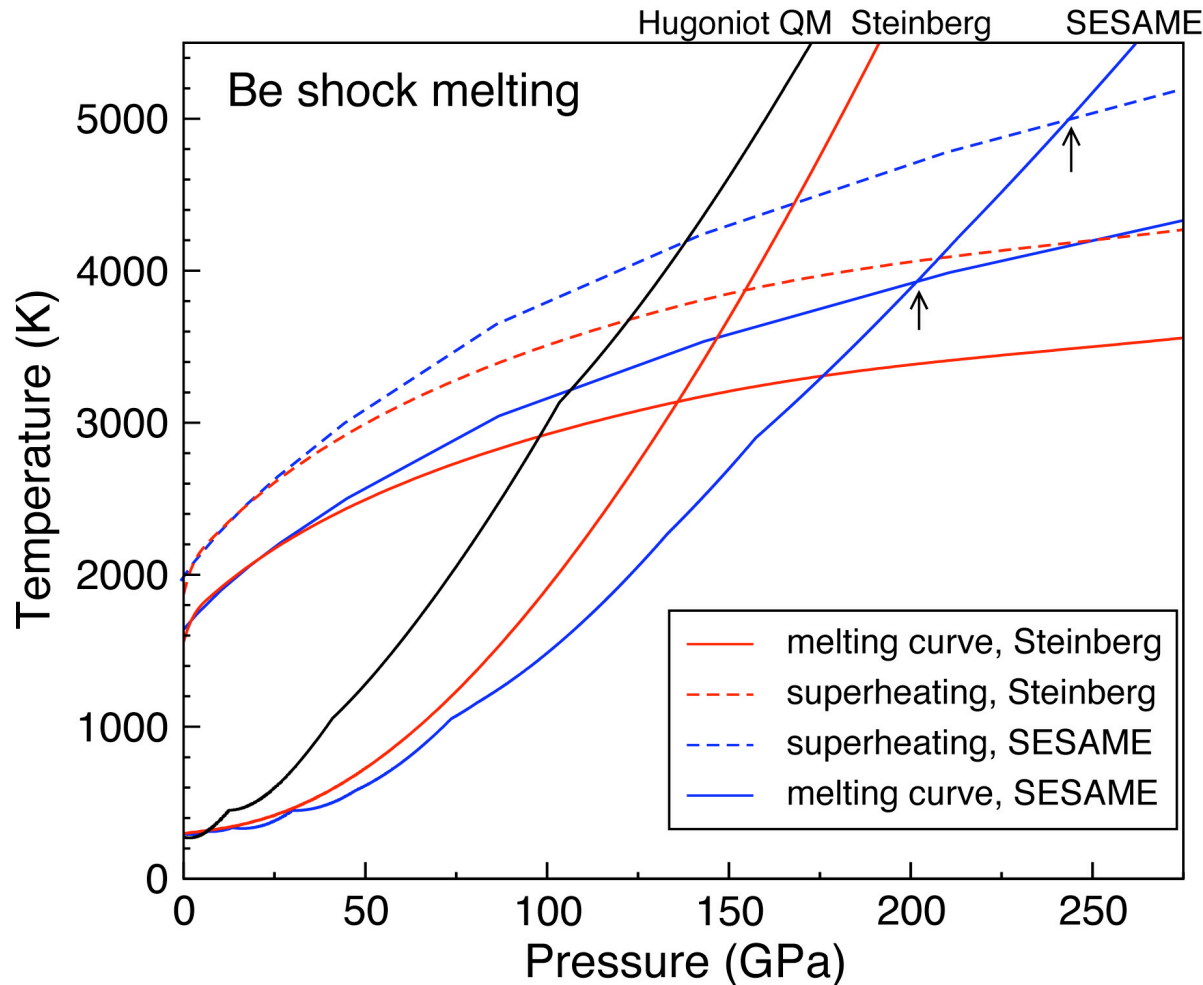
Alternative EOS for Be



Shock Hugoniots are similar up to several hundred GPa (reasonably consistent in Z flyer impact shots to 200 GPa)

STP isentropes vary significantly: a potential discriminant.

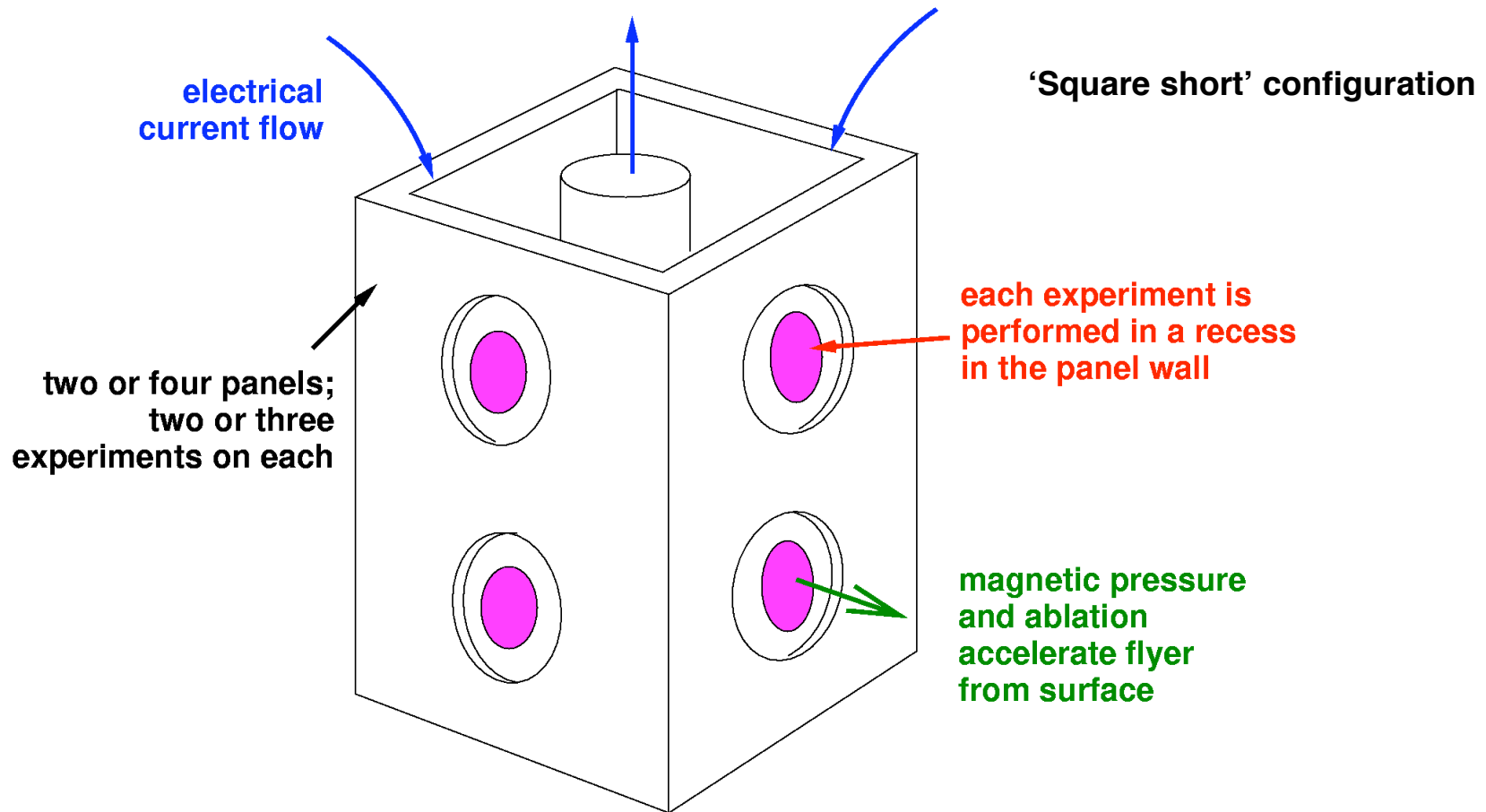
Uncertainty in thermal EOS and dynamics of melting



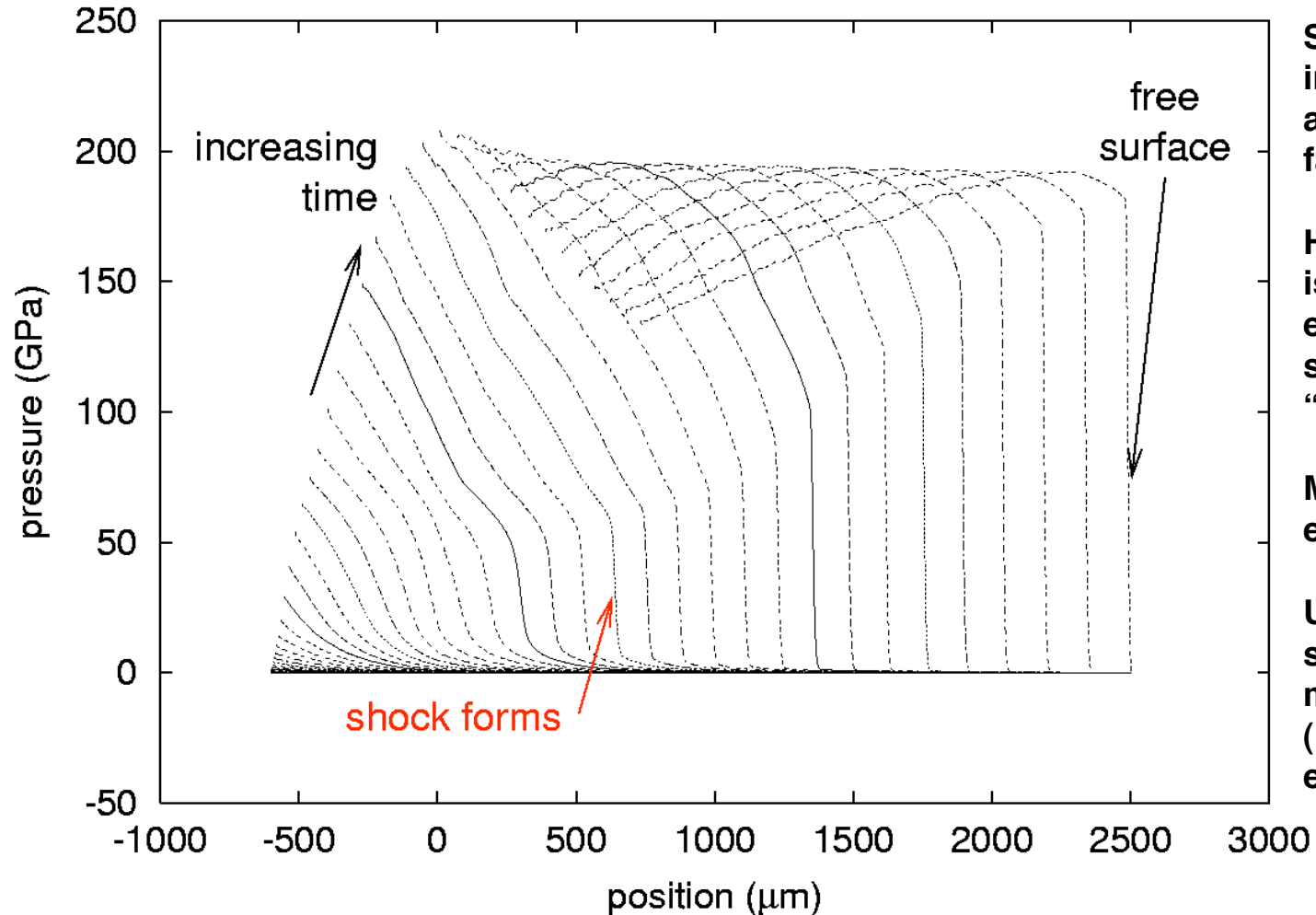
Unless experiments measure melt under precise NIF loading conditions, we need to resolve equilibrium melting curve for Be *and* understand any dynamic effects.

Dynamic melting theory: Luo et al.

Dynamic materials experiments at Z



(Quasi-) isentropic compression in smooth wave



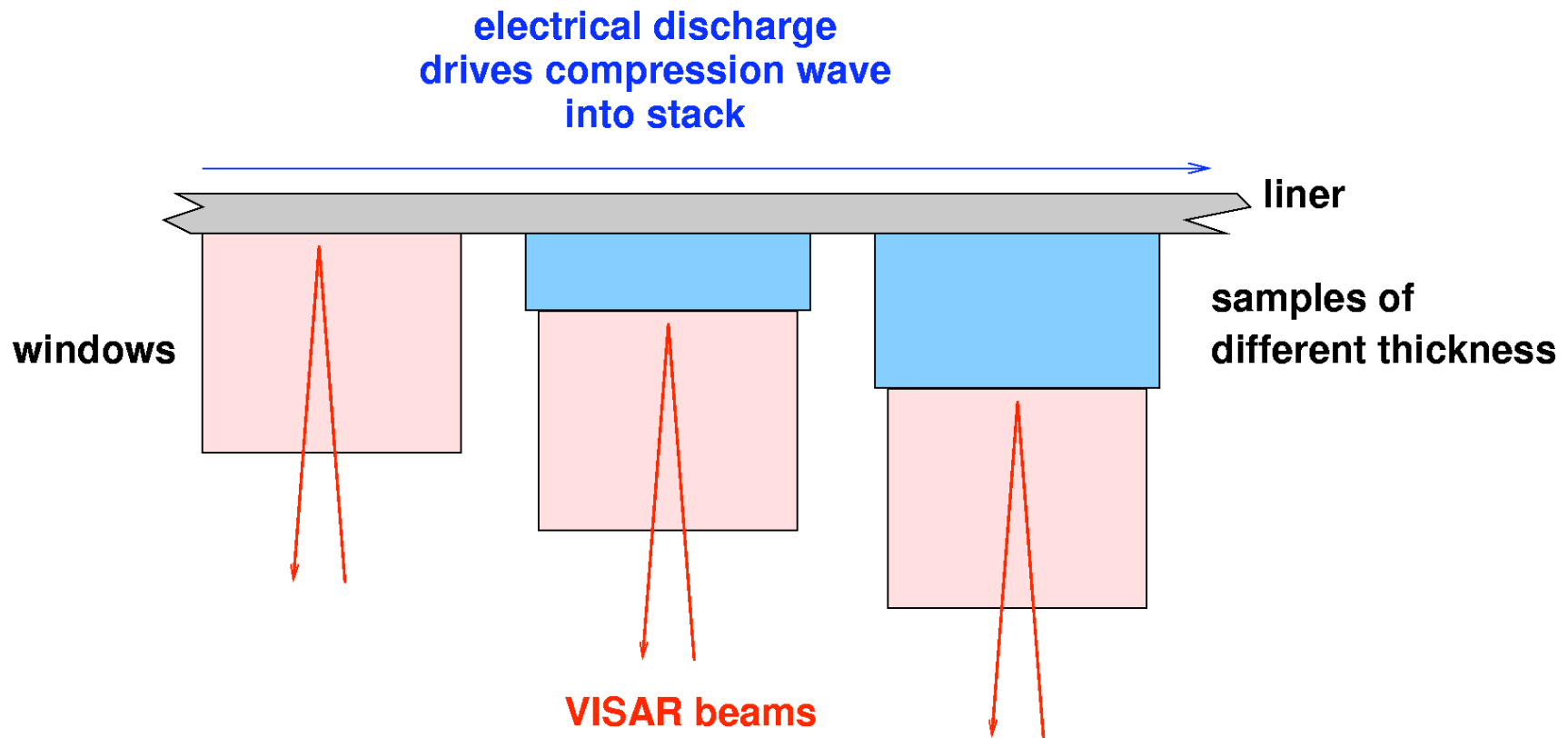
Smoothly increasing load applied to one face of sample.

Heat conduction is slow, so elements of sample are "insulated".

Measure wave evolution.

Uniaxial wave, so plastic flow may occur (increases entropy).

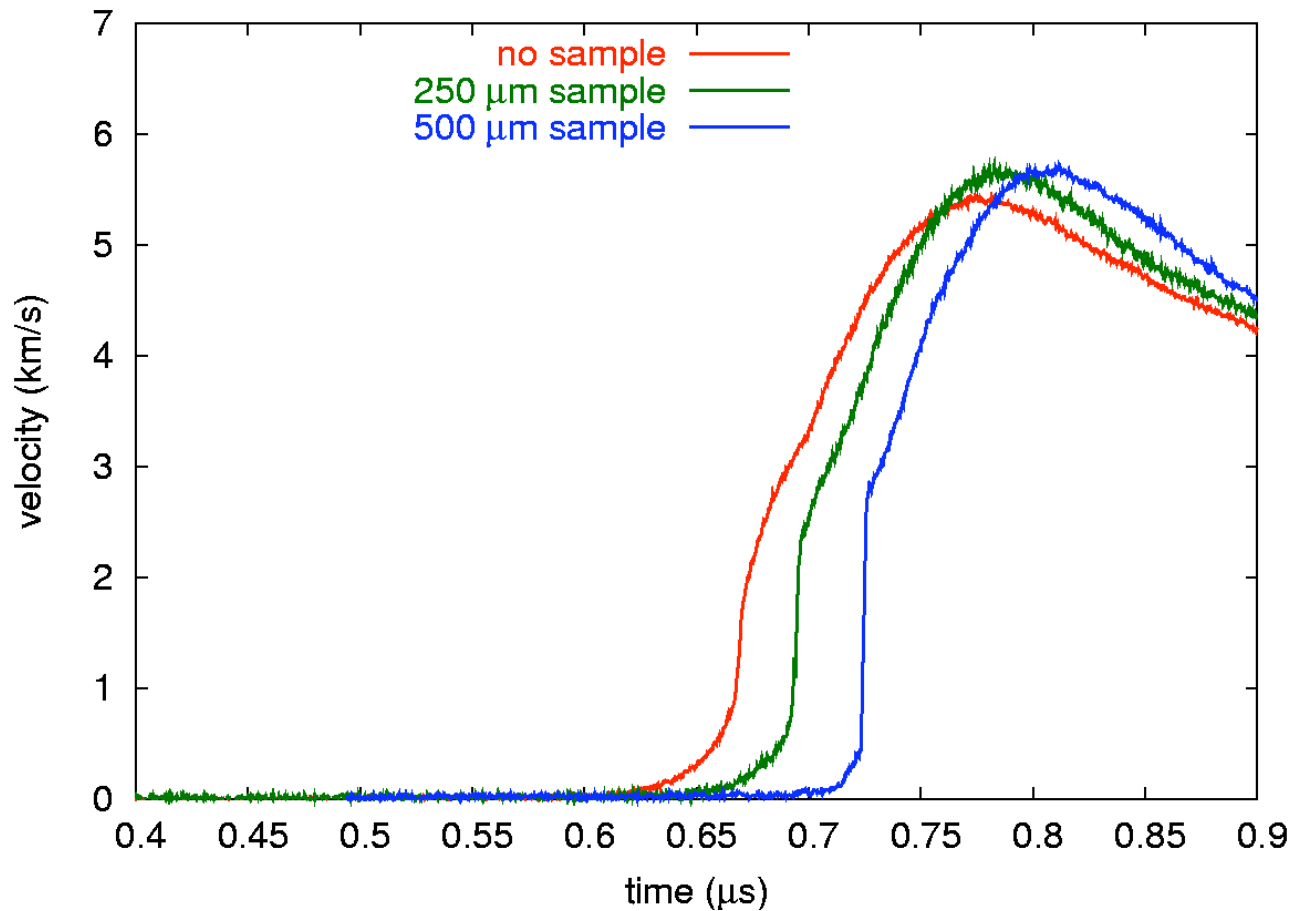
Quasi-isentropic compression at Z



Vertical scale exaggerated

Observe evolution of compression wave for different sample thicknesses.

Velocimetry shows evolution of ramp wave



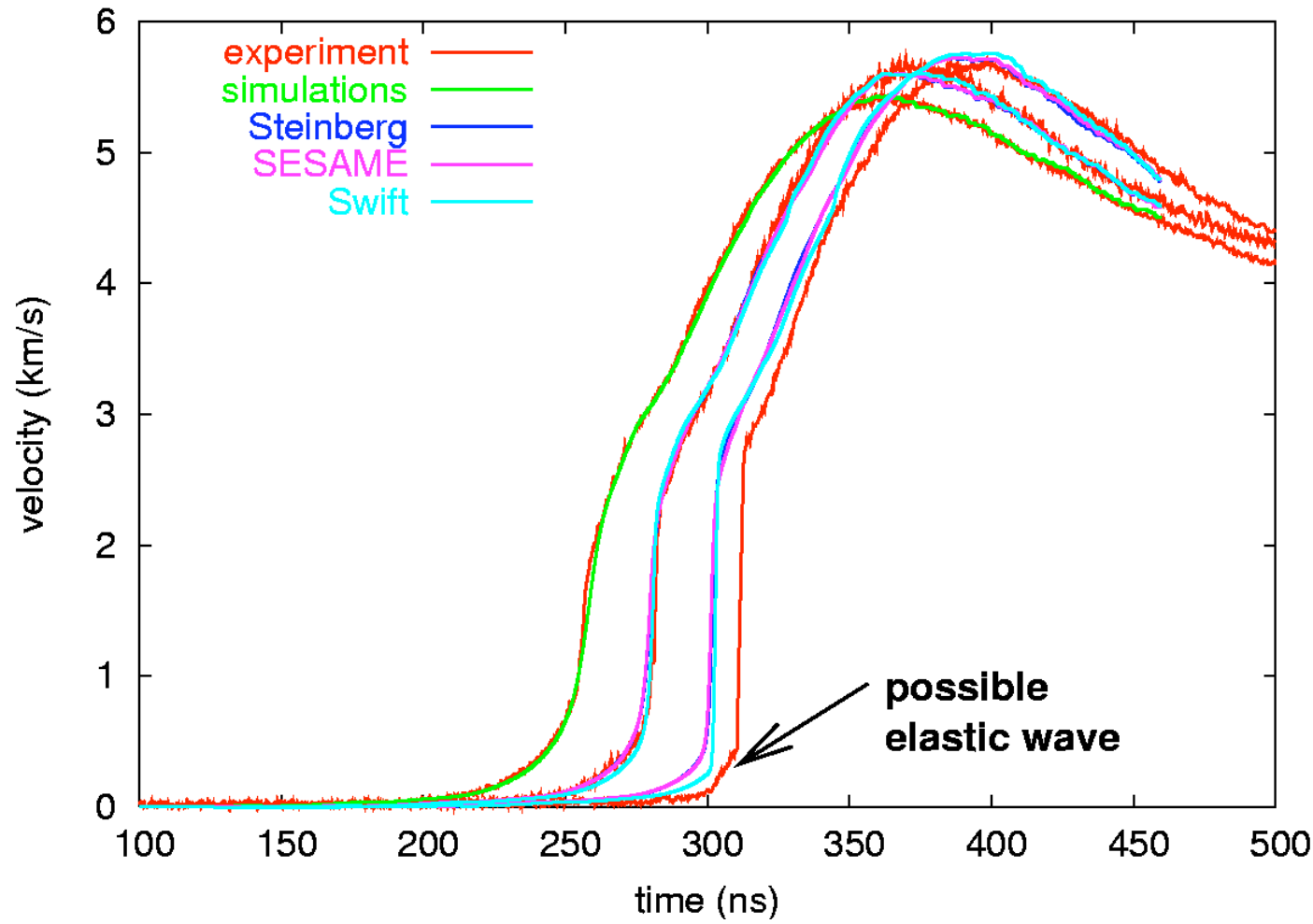
Z shot 843

Be samples
LiF windows

Principal isentrope to
~200 GPa

*Swift, Paisley, and
Knudson,
Proc APS SCCM'03*

Performance of EOS for Z ICE



Forward simulations:
Lagrangian finite-
difference, artificial
viscosity.

Pressure drive adjusted
to reproduce Al velocity
history.

Very little difference in
predicted velocity
histories for isentropic
compression (and all
reproduce the data well).

Some difference during
build-up to shock, but
strength also matters.

Material dynamics with lasers

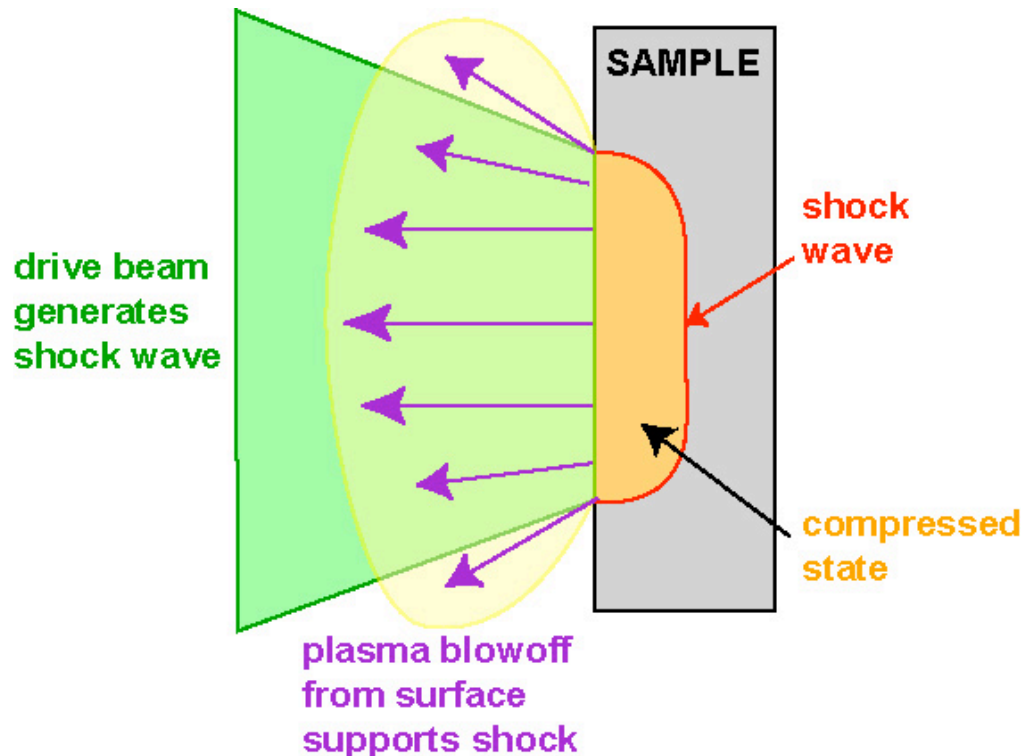
Pros:

- “Easy” synchronization of diagnostics
- Small samples (so easier to obtain crystals hence bridge length scales; hazardous / expensive materials)
- Less “collateral energy and momentum”: easier to recover samples
- Flexibility in applied load

Cons:

- Difficult to generate constant pressure by ablation
- Potential preheat
- Time / length scales shorter than for many shock applications
- Can be difficult to obtain adequate samples

Loading by laser ablation



Pulse: 0.2 to 3.6 ns, 527 nm
2.5 GW/cm² to 1 TW/cm² (5 mm dia spot)
Sample thickness: tens to hundreds of μm

Shocks in elements:
Swift, Tierney, Kopp, and Gammel,
Phys. Rev. E, 69, 036406 (2004)

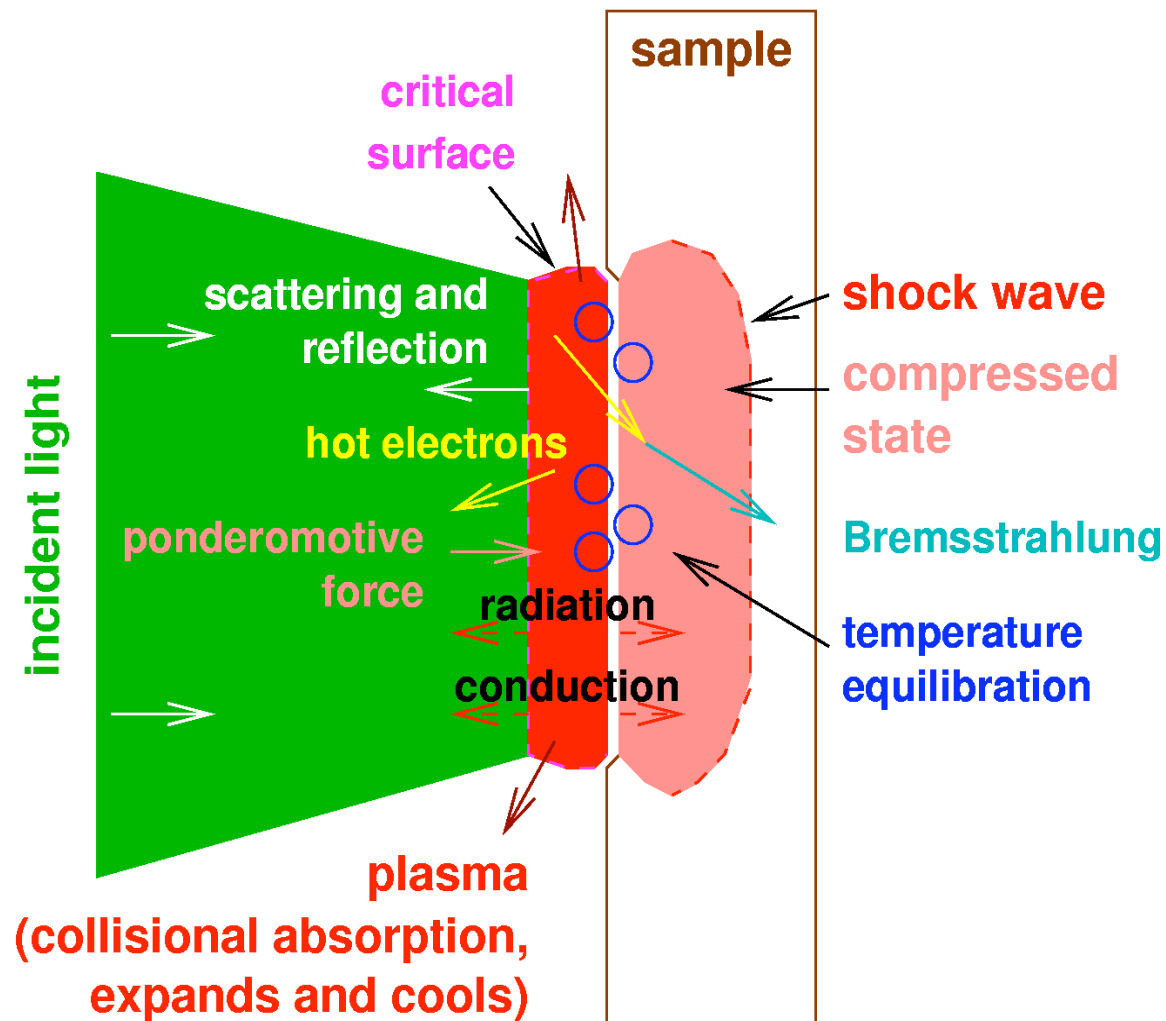
Compounds/alloys:
Swift, Gammel, and Clegg,
Phys. Rev. E, 69, 056401 (2004)

Quasi-isentropic compression ("LICE"):
Swift and Johnson,
Phys. Rev. E (submitted)

Conference papers:
HDP5, APS SCCM'03;
APS DPP '02 and '03

Useful for time-dependent phenomena, such as plasticity and phase changes, and for variable loading histories

Laser-matter interactions



Not all these processes matter in our regime (~0.1-100 GPa, 10-1000 μm , 0.1-1000 ns).

LASNEX and HYADES were used to assess importance in 1D.

Minimal reasonable model:

- 3T hydrodynamics,
- laser transport and deposition,
- heat conduction,
- Thomas-Fermi ionization,
- gray radiation diffusion.

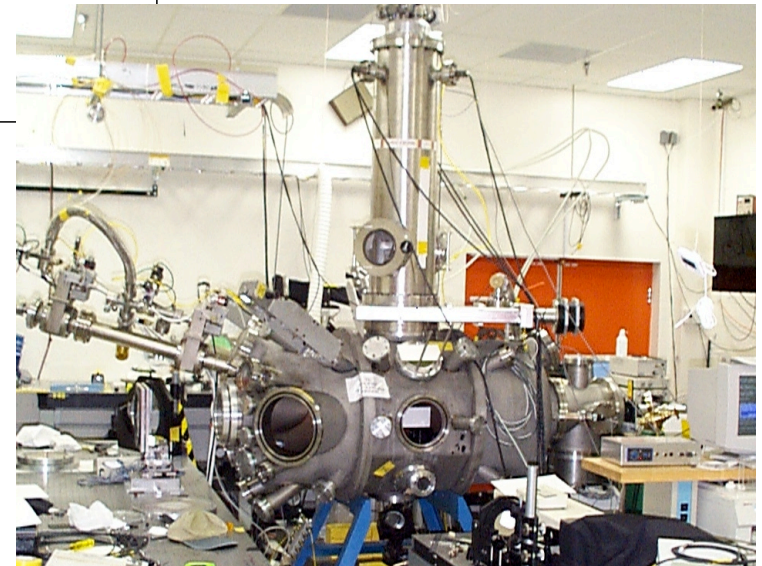
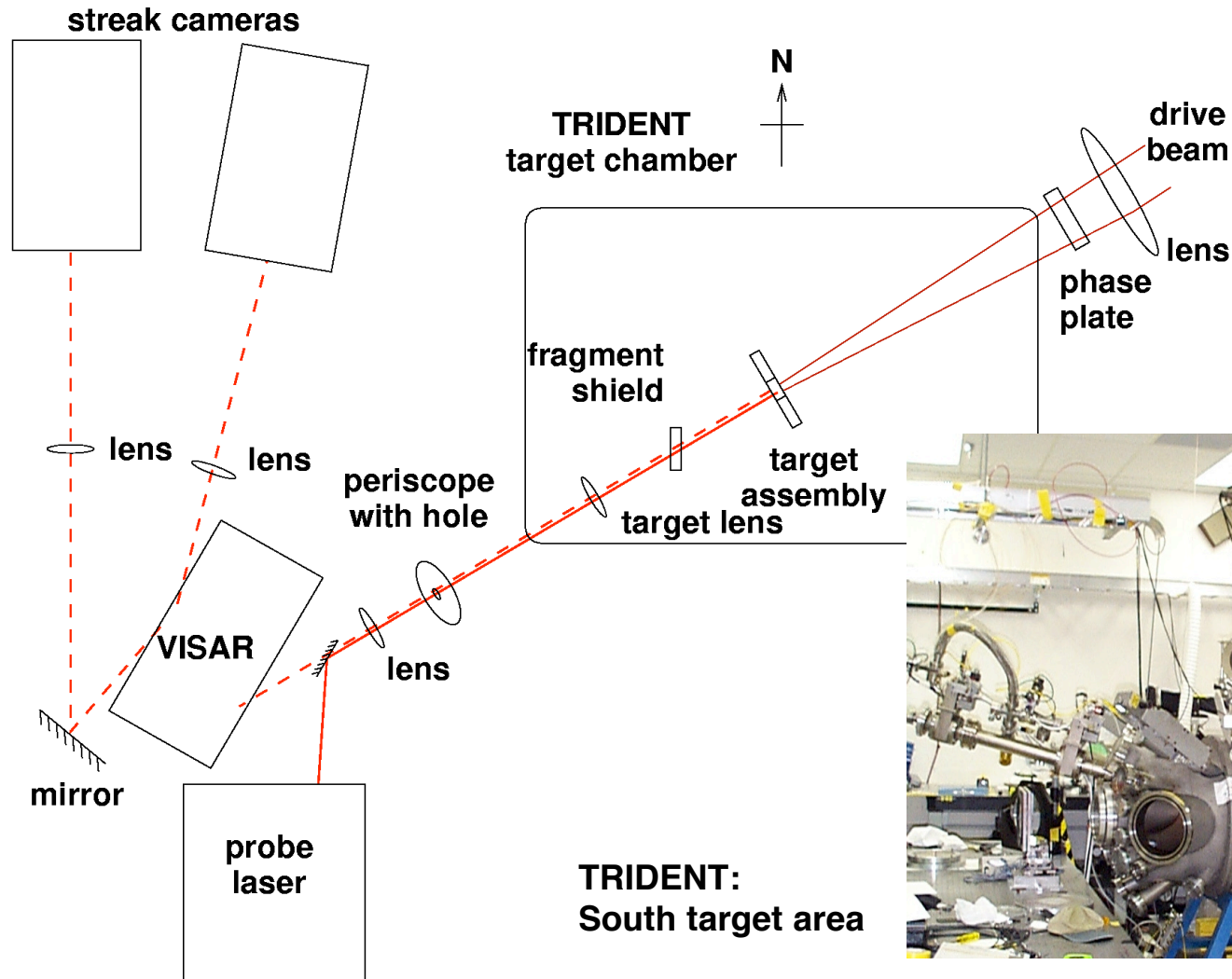
Tested for elements from Be to Au and intermetallic compounds, shock pressures to ~40 GPa

TRIDENT laser facility



- 3 beams, Nd:glass
- ~ 1 to 1000 J/beam $\pm \sim 1$ J
- pulse ~ 100 fs to 3 μ s
- 1054 / 527 / 351 nm
- spot size:
 - ~ 50 μ m to 50 mm
- controlled intensity history
- “nanosecond” mode:
 - 13 elements of 180 ps
 $\Rightarrow \sim 2.5$ ns
 - can stack two pulses
 $\Rightarrow \sim 5$ ns
 - computer control of shape (before amps)
- up to ~ 8 shots / day, or ~ 25 of below 40 J
- spatial smoothing
- misc. diagnostics

Typical experimental layout

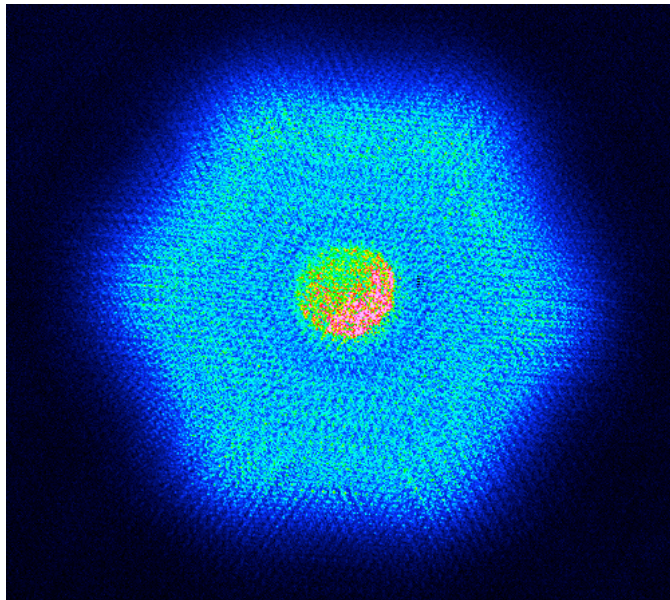


Spatial distribution of laser irradiance

Beams: super-Gaussian, can focus to $\sim 100 \mu\text{m}$. Want top-hat with few mm dia.
Use diffractive optical elements; imperfect (undiffracted \Rightarrow hotspot); don't have for all spot sizes.
Defocusing used to remove hotspots or adjust size; introduces some long- λ variation.

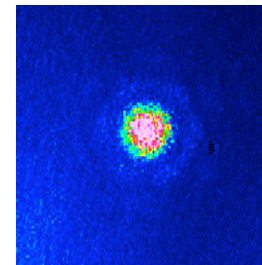
Fresnel zone plate

design = 4 mm; defocus to 5 mm

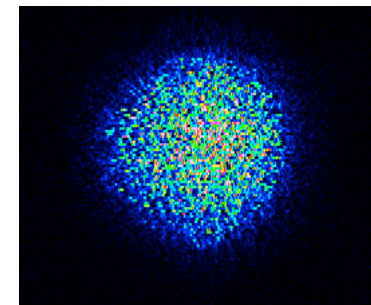


Random phase plate

design = 0.6 mm; defocus to 1-2 mm

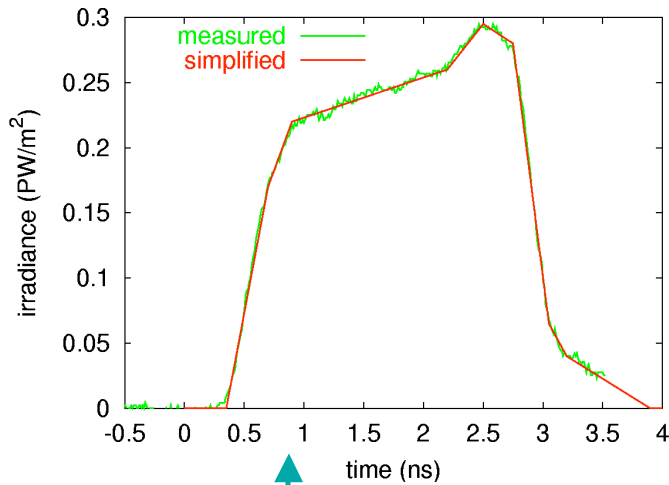


1 mm



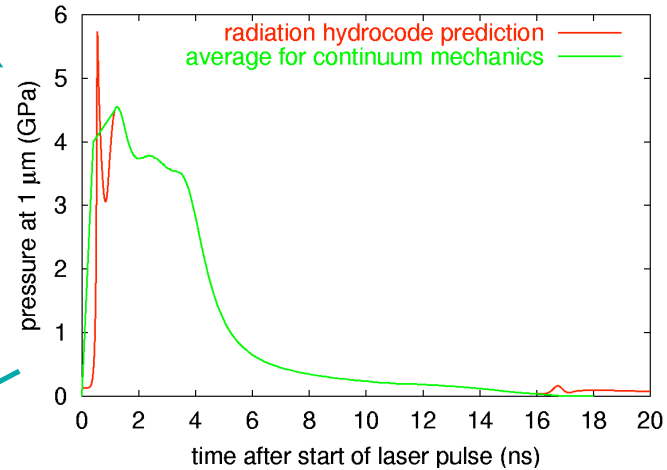
*Line-imaging velocimetry has shown no evidence of spatially-varying drive over central $\sim 50\%$.
Some variation was observed in displacement interferometry (i.e. integrated) after tens of ns.*

Time-dependent response: rad-hydro and continuum mechanics

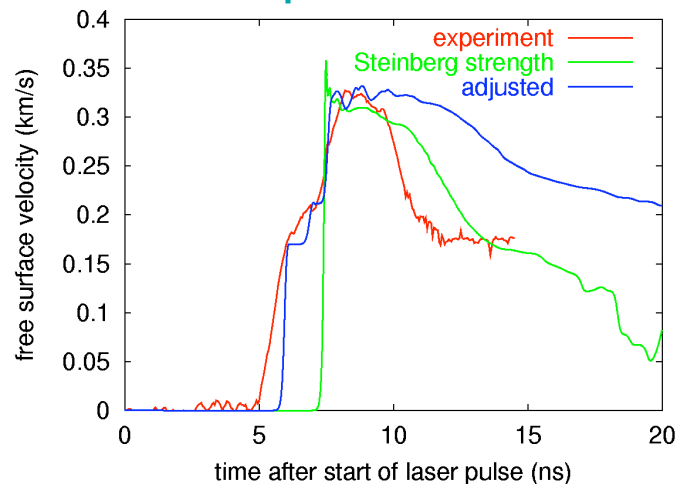


Measured/planned irradiance history

Rad-hydro, simple/no strength (HYADES)



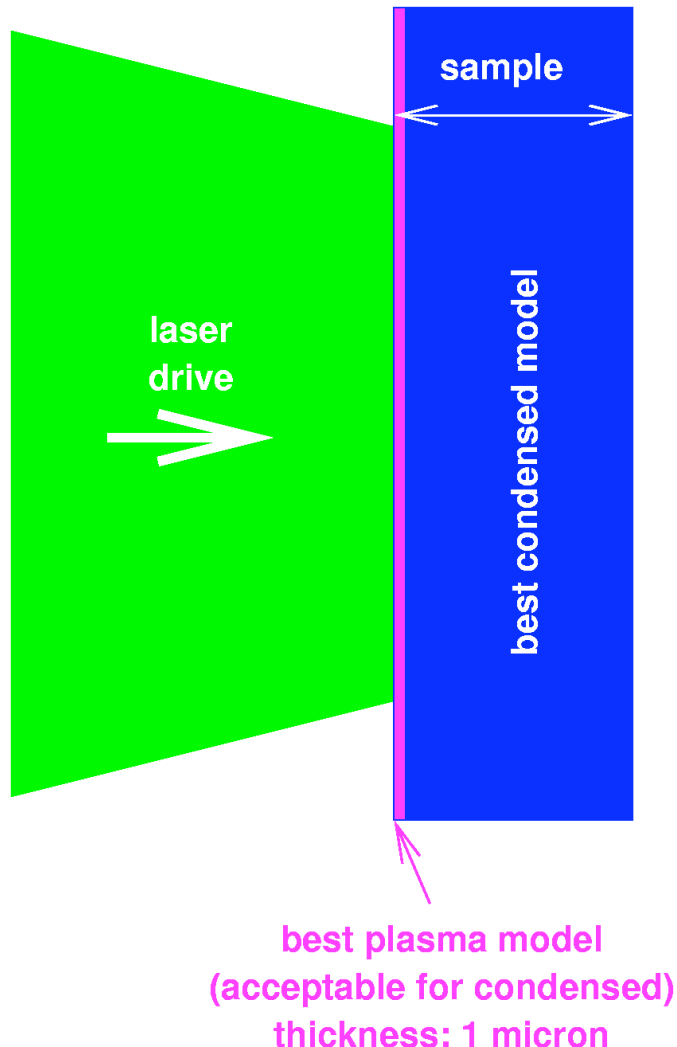
Stress history predicted just inside sample



Surface velocity history (etc),
cf experiment

Continuum mech,
strength etc
(LAGC1D/LAGC)

Radiation hydrodynamics / continuum mechanics simulations

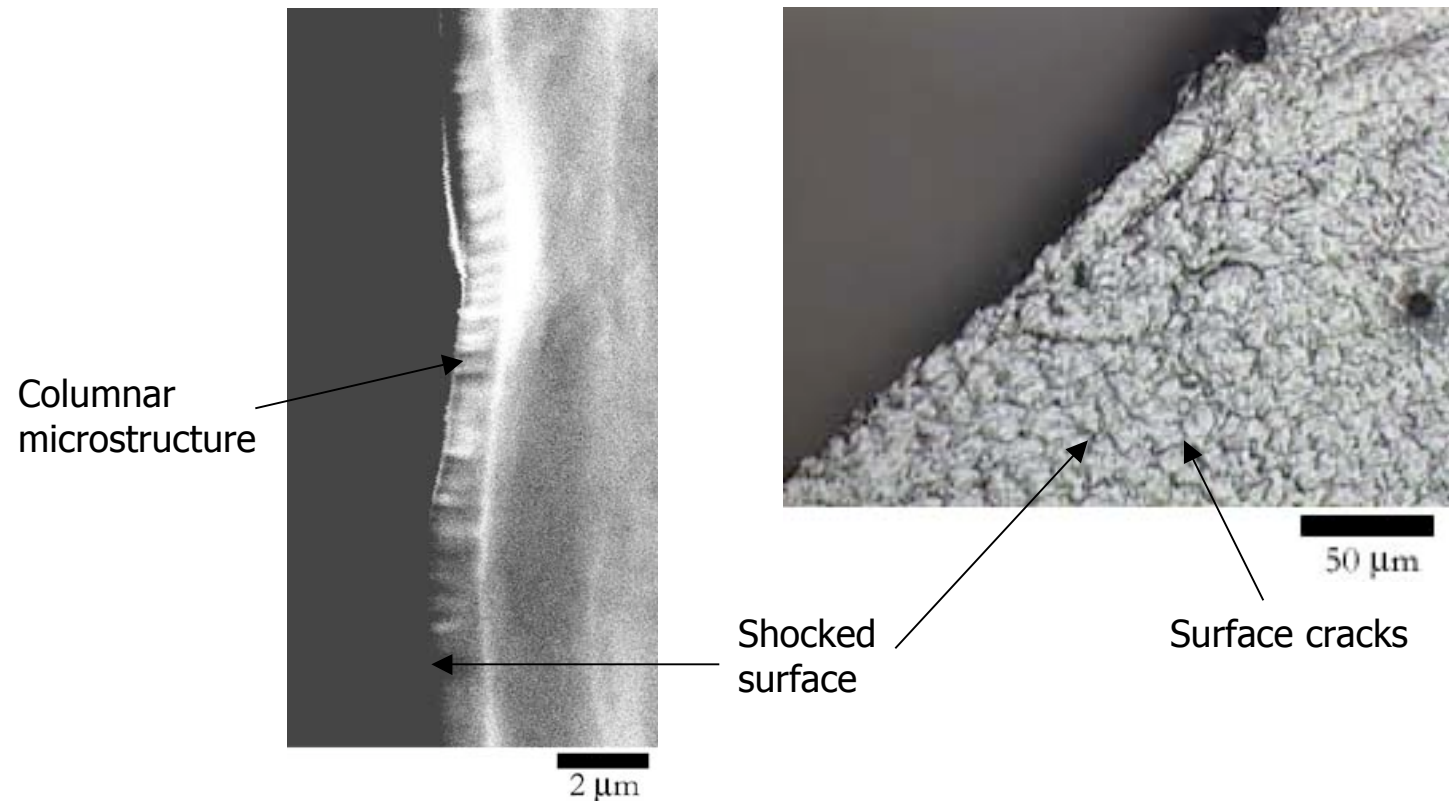


Usual situation:
reasonable condensed model (e.g. QM calculations + shock wave data);
no validated model for (possibly mixed-species) plasma.

**Various models used in ablation layer;
best condensed model elsewhere.**

**“Plasma” model should give reasonable
initial density and compressibility in
condensed phase.**

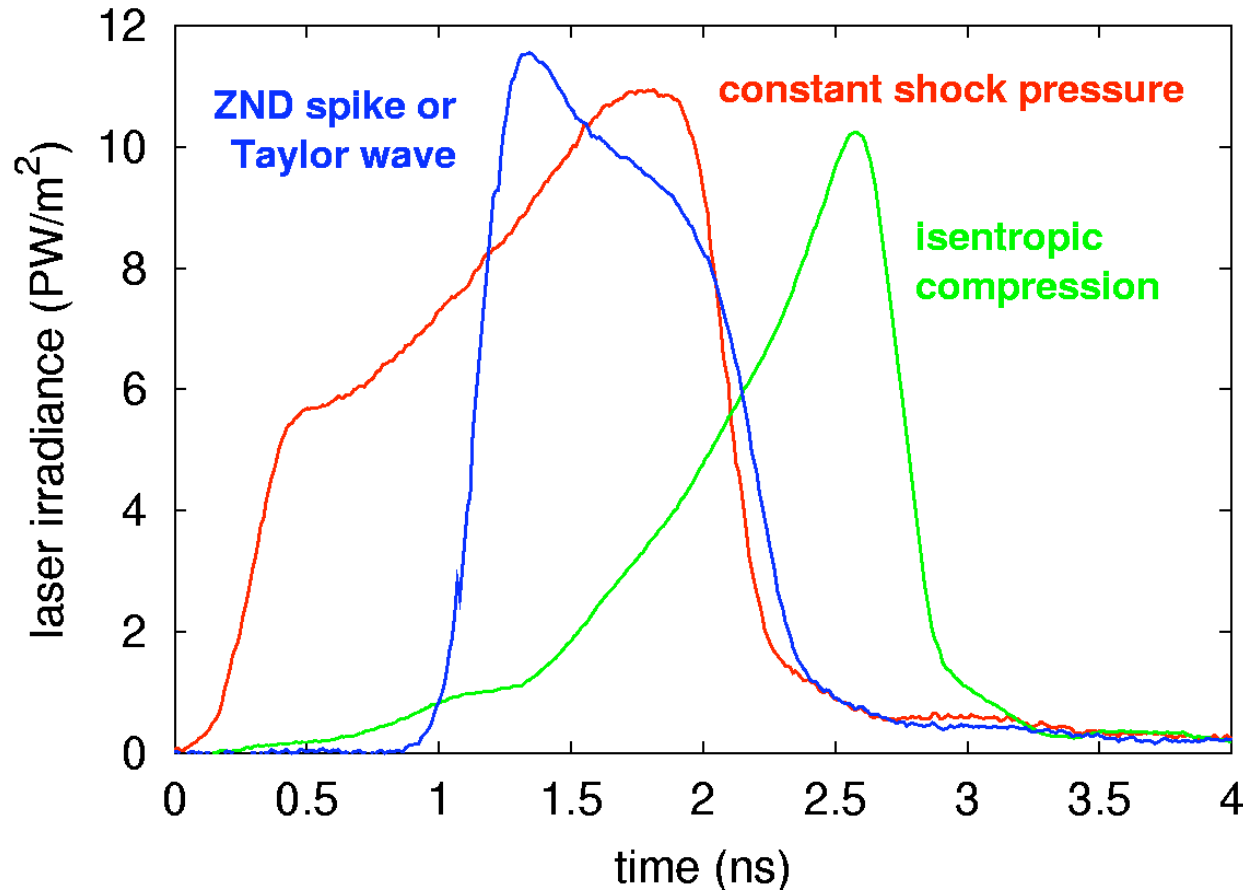
Ablative loading: phenomena near drive surface



Postshot recovered samples (here NiAl crystal: Loomis and Peralta).

~ 1 ns drive to ~ 10 GPa: ~ 1 micron layer shows evidence of recrystallization, presumably following melting; microstructure in bulk is unaffected.

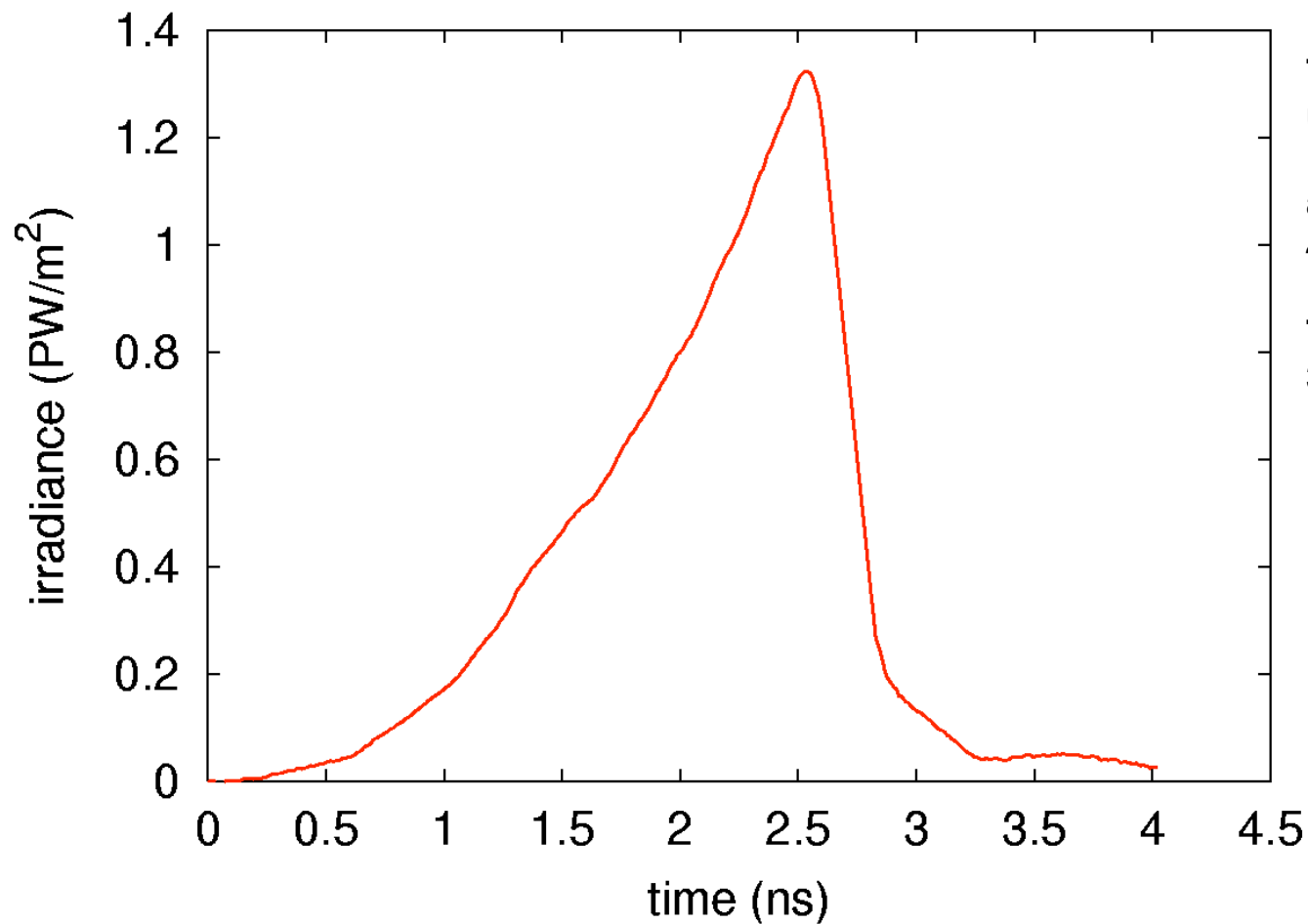
Finite risetimes: “always isentropic”



Heating and breakdown of condensed sample (poorly / arbitrarily modeled) sharpens foot.

Few tens of ps rise: shocks almost always develop from ramp (demonstrated to occur e.g. Be crystal data).

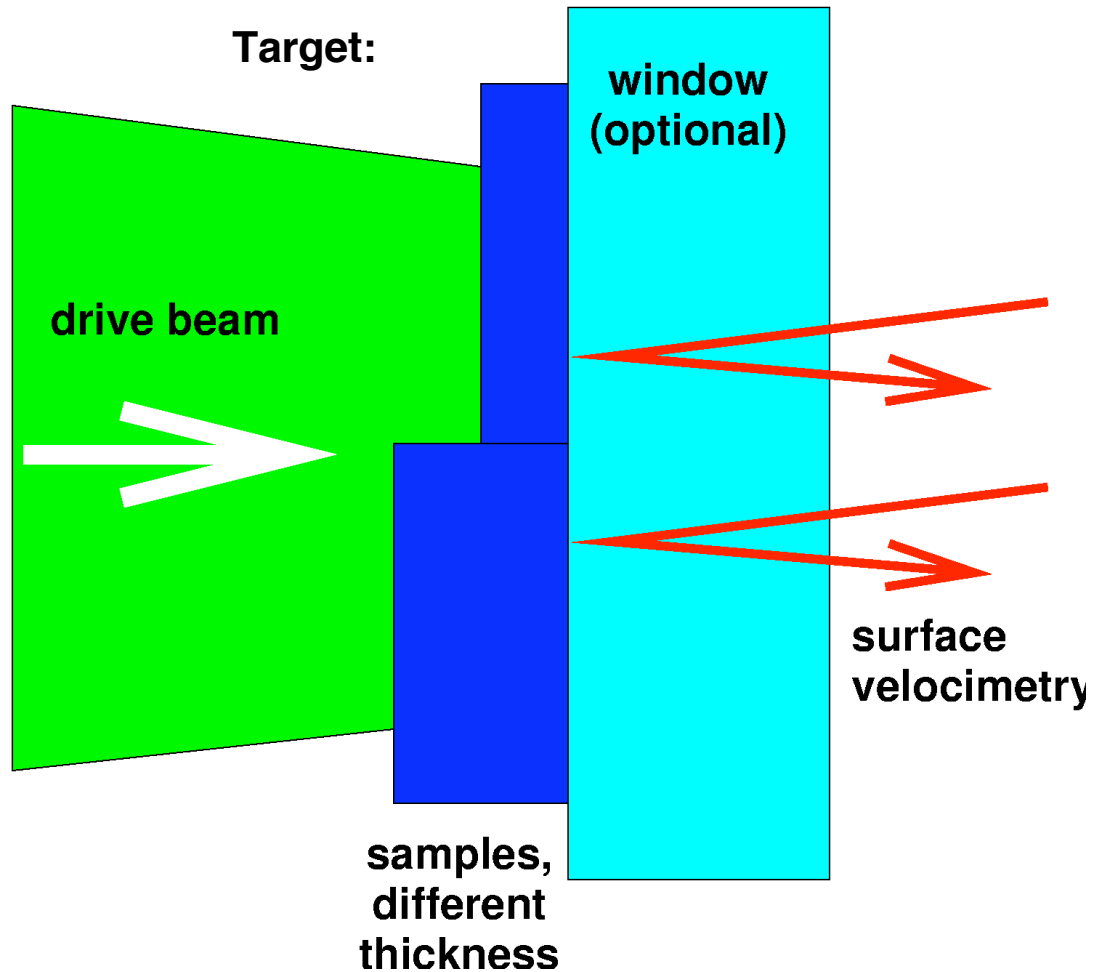
TRIDENT can deliver smoothly ramped laser pulses



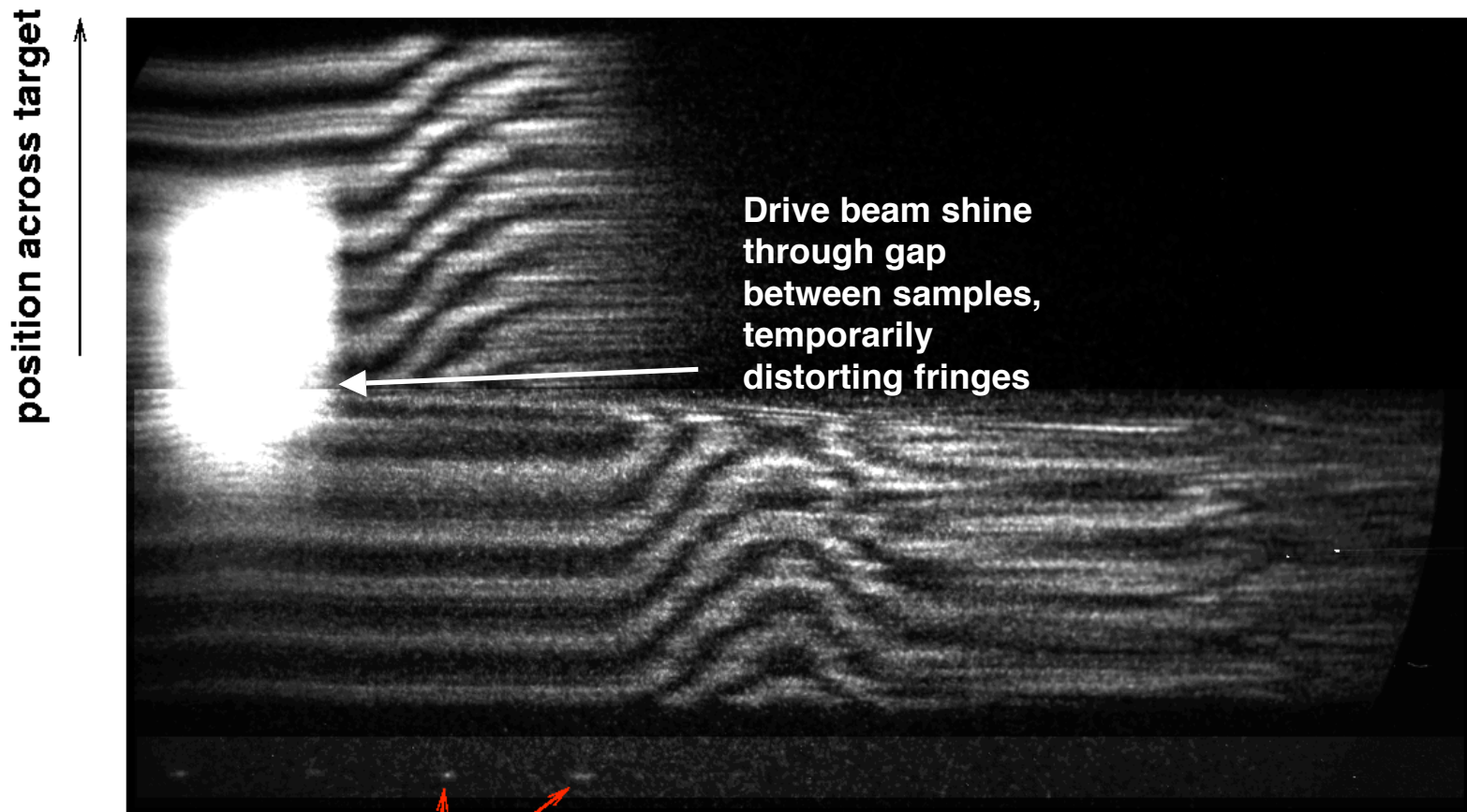
**TRIDENT in ns mode:
up to 13 elements of
180 ps each;
amplitudes
~independent.**

**TRIDENT shot 15018:
33 J**

Simultaneous irradiation of multiple samples



Example LICE line VISAR record



position across target

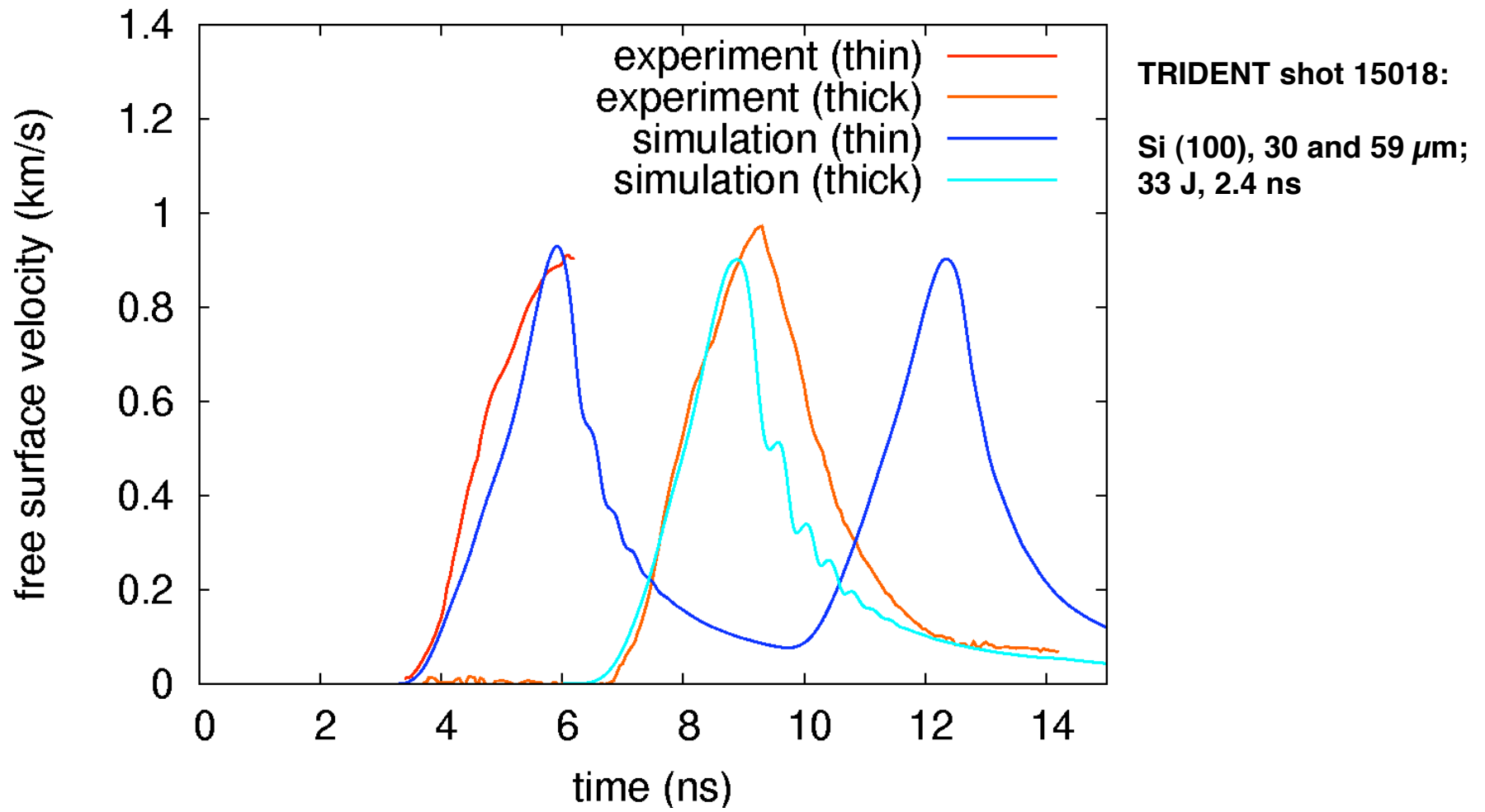
Drive beam shine through gap between samples, temporarily distorting fringes

timing markers (2 ns apart)

time

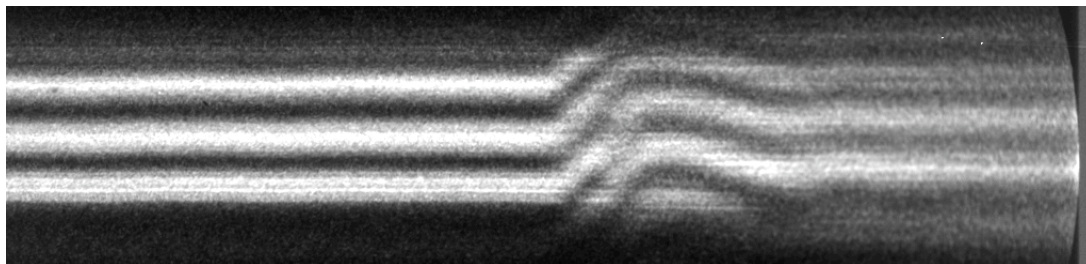
TRIDENT shot 15018:
Si (100), 30 and 59 μm ; 33 J, 2.4 ns
425 m/s/fringe, 2 mm field

Rad-hydro simulations of LICE data

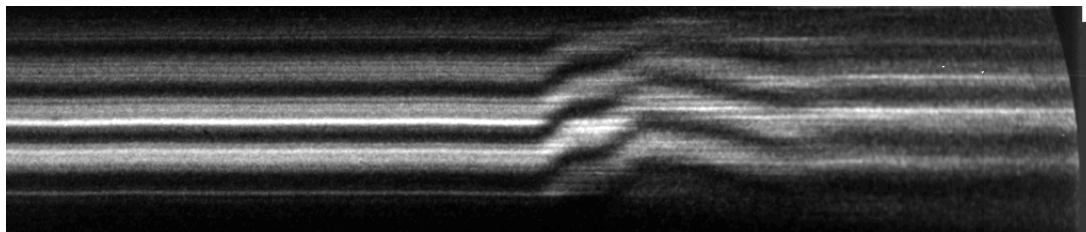


Shock and quasi-isentropic data for Zr

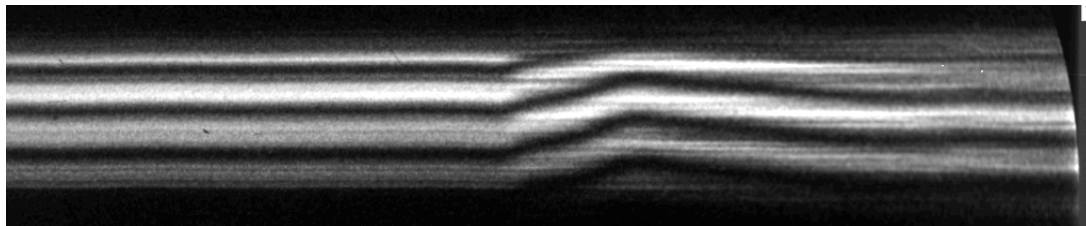
Foils, line-VISAR records, 800 m/s/fringe sensitivity, recording period of 20 ns, different delays:



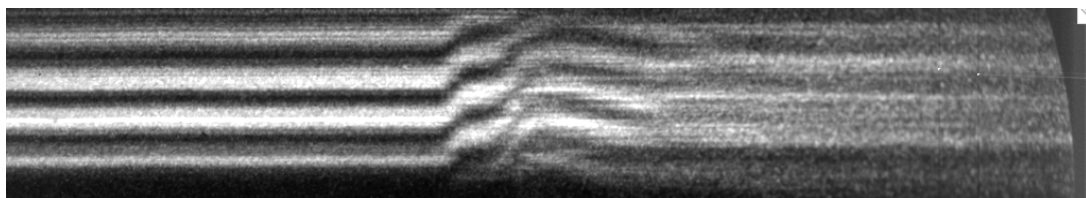
Shock (shot 17130):
26 μm
102 J, 2.5 ns
 ~ 17 GPa



LICE (shot 17137):
26 μm
48 J, 2.5 ns



LICE (shot 17138):
61 μm
49 J, 2.5 ns

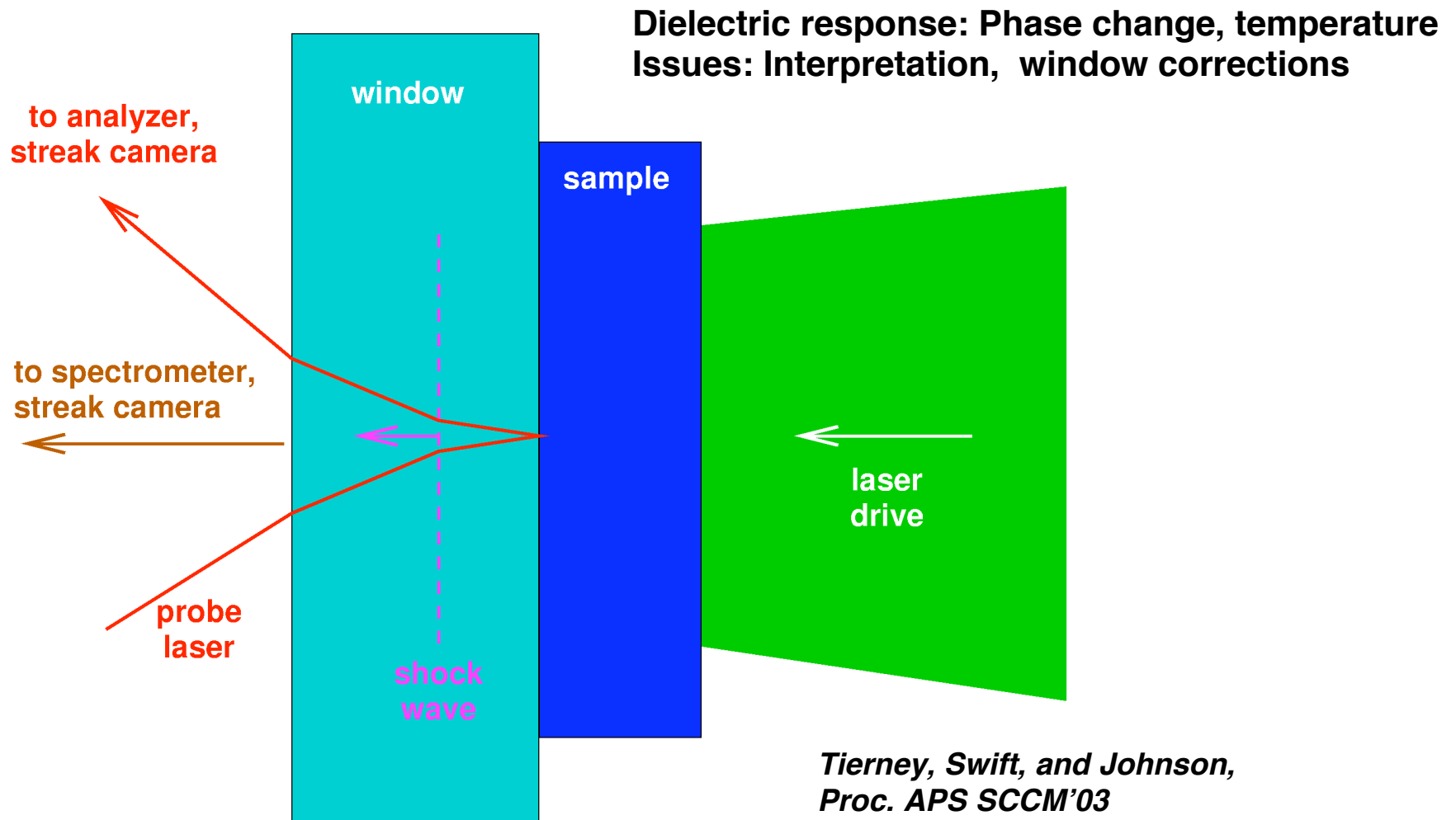


LICE (shot 17139):
26 μm
108 J, 2.5 ns

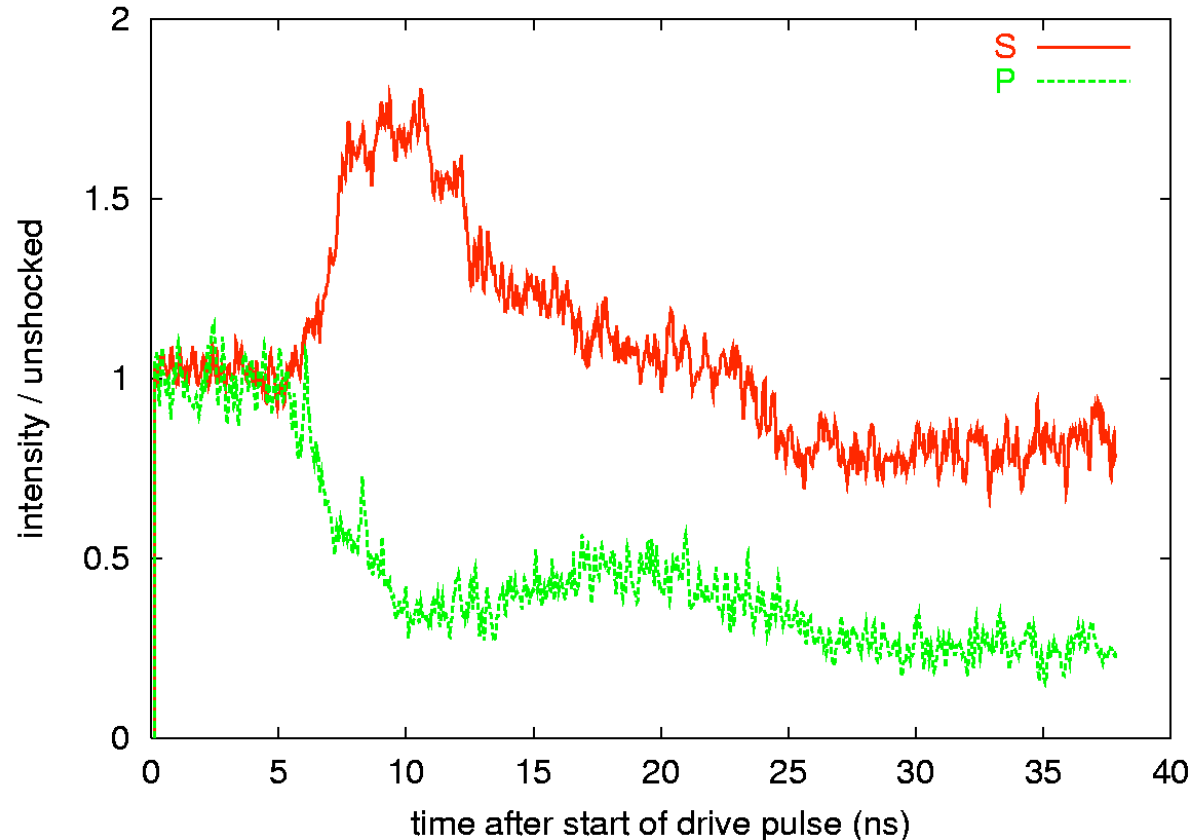
Diagnostics for material response

- Surface velocimetry: EOS, elastic constants, flow stress, phase changes (through volume change)
- Emission spectroscopy: temperature, hence EOS, phase changes
- Electron-photon and ion-photon interactions: density of states (EOS components), phase changes - e.g. ellipsometry, Raman spectroscopy
- In-situ x-ray diffraction: compression (EOS), lattice deformation (phase changes, elastic and plastic strain)
- Imaging velocimetry and displacement interferometry: polycrystal / grain boundary effects
- Sample recovery and micrography

Dynamic ellipsometry and emission spectrometry



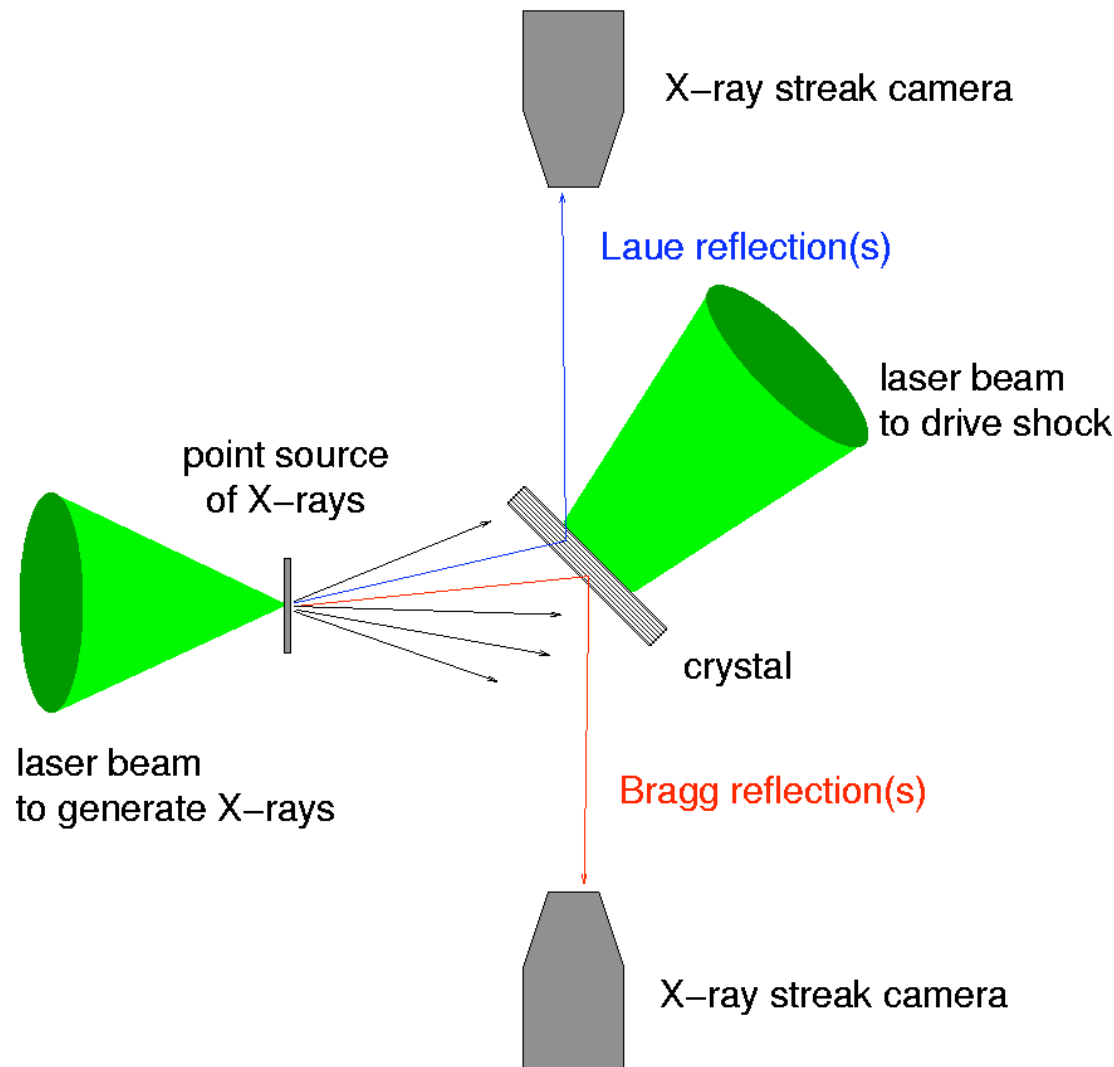
Dynamic ellipsometry



- Indicates phase change at sample surface.
- Scoping experiments performed on Si; melt demonstration on Sn.
- Window characterization shots performed using W/LiF (W not expected to melt or exhibit solid-solid changes up to several hundred GPa)
- Series of shots fired on Be foils: strong signal at relatively low pressures. BeO / band structure?

TRIDENT shot 14972: Sn with LiF release window.
Pressure on release = 28 GPa.
(Signals unchanged on release to lower pressures.)

Transient X-ray diffraction

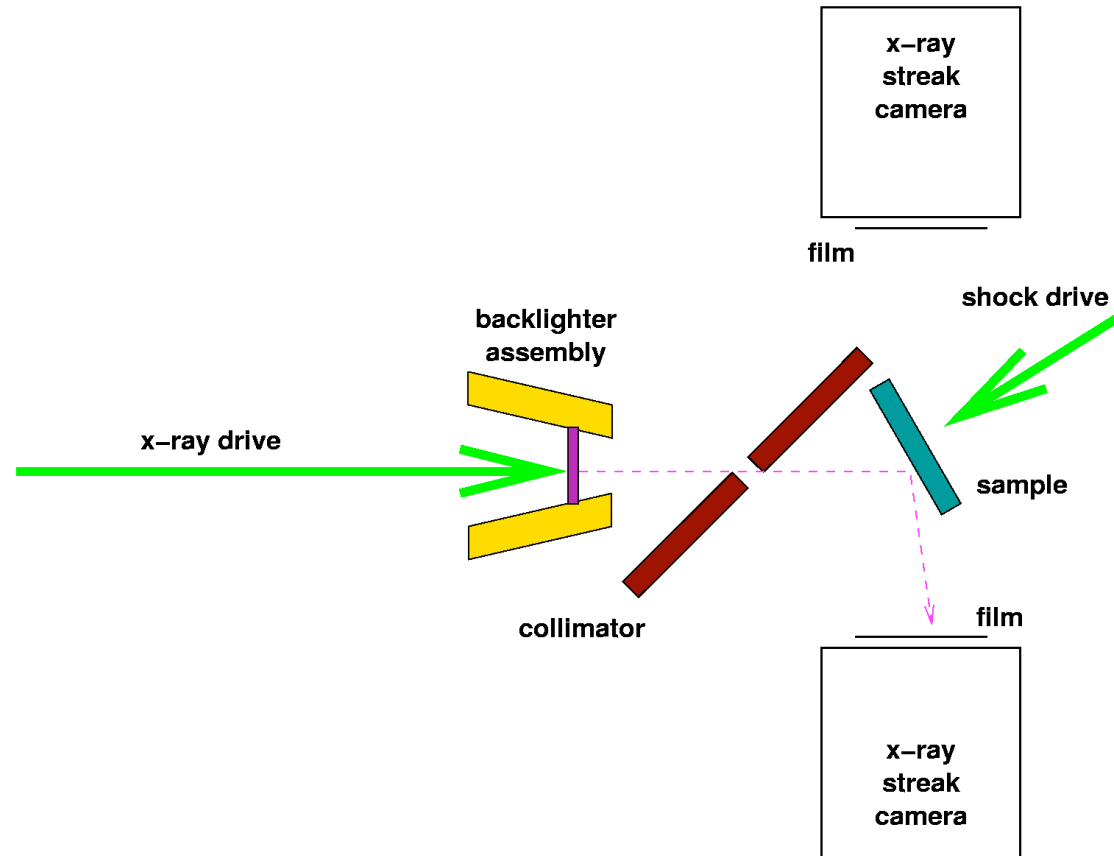
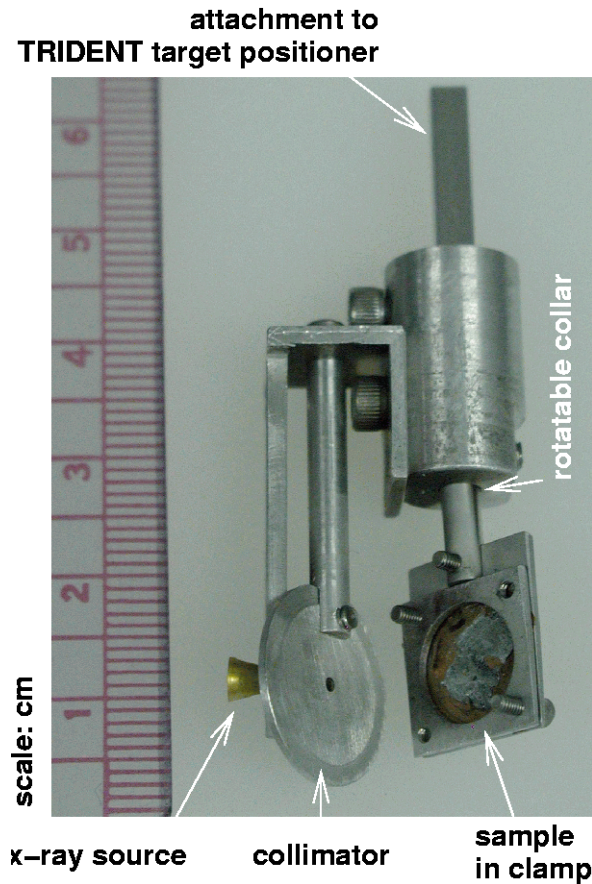


X-rays penetrate
=> sample different states
through thickness

Crystal + diverging X-rays:
diffracting point moves
across surface with
compression.

TRIDENT: max photon
energy
~ 6 keV or ~ 2 Å,
~1 ns duration

Polycrystal TXD

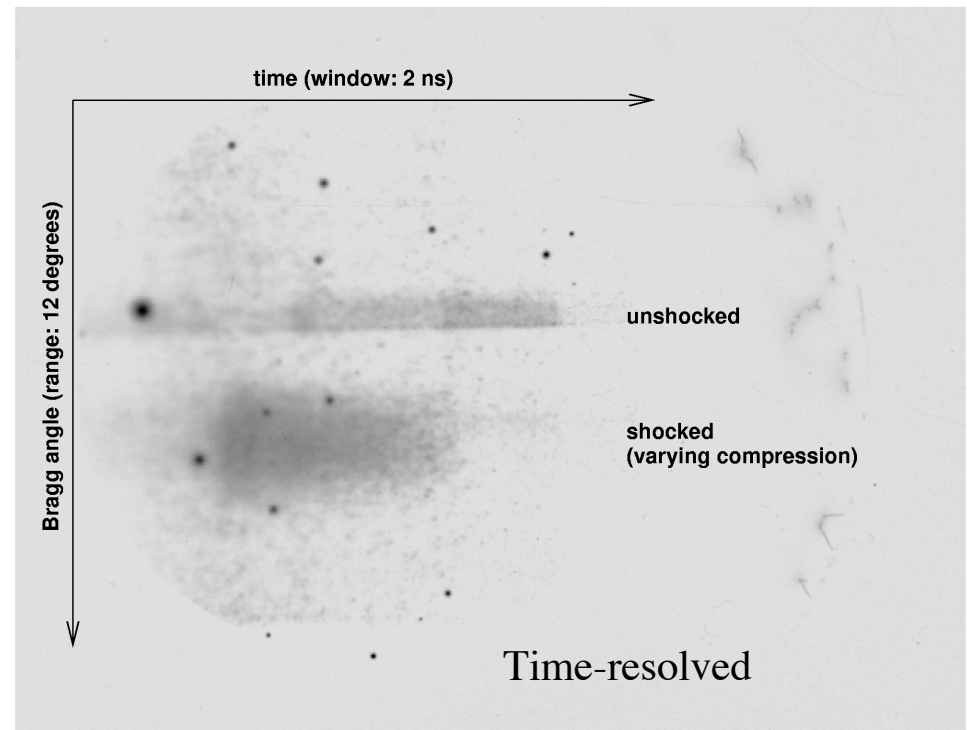
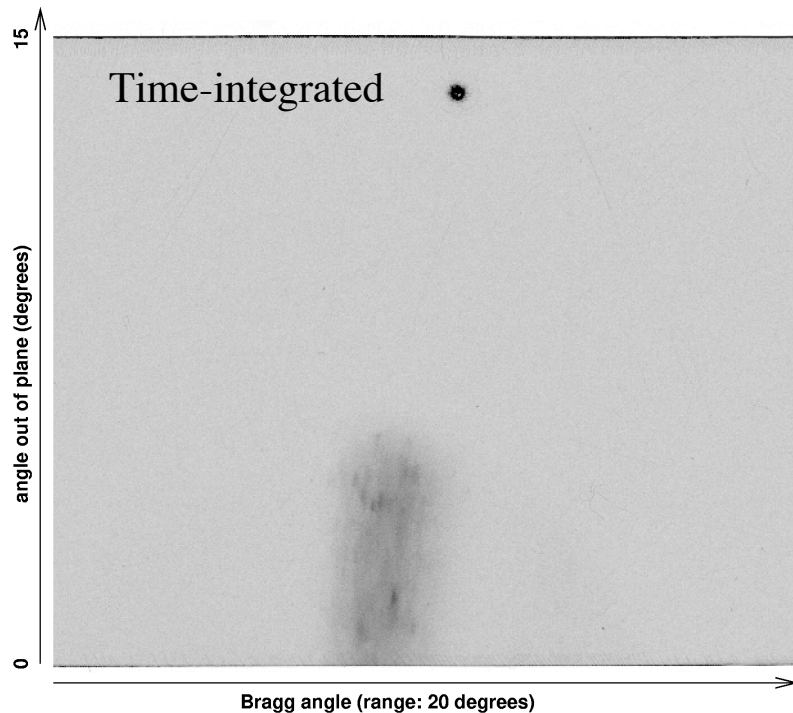


Desirable for TXD at high pressures, requiring smaller drive spot.

May be necessary for response of representative Be-Cu material: fabrication method may matter.

Also desirable to follow phase changes: daughter phase may be polycrystalline.

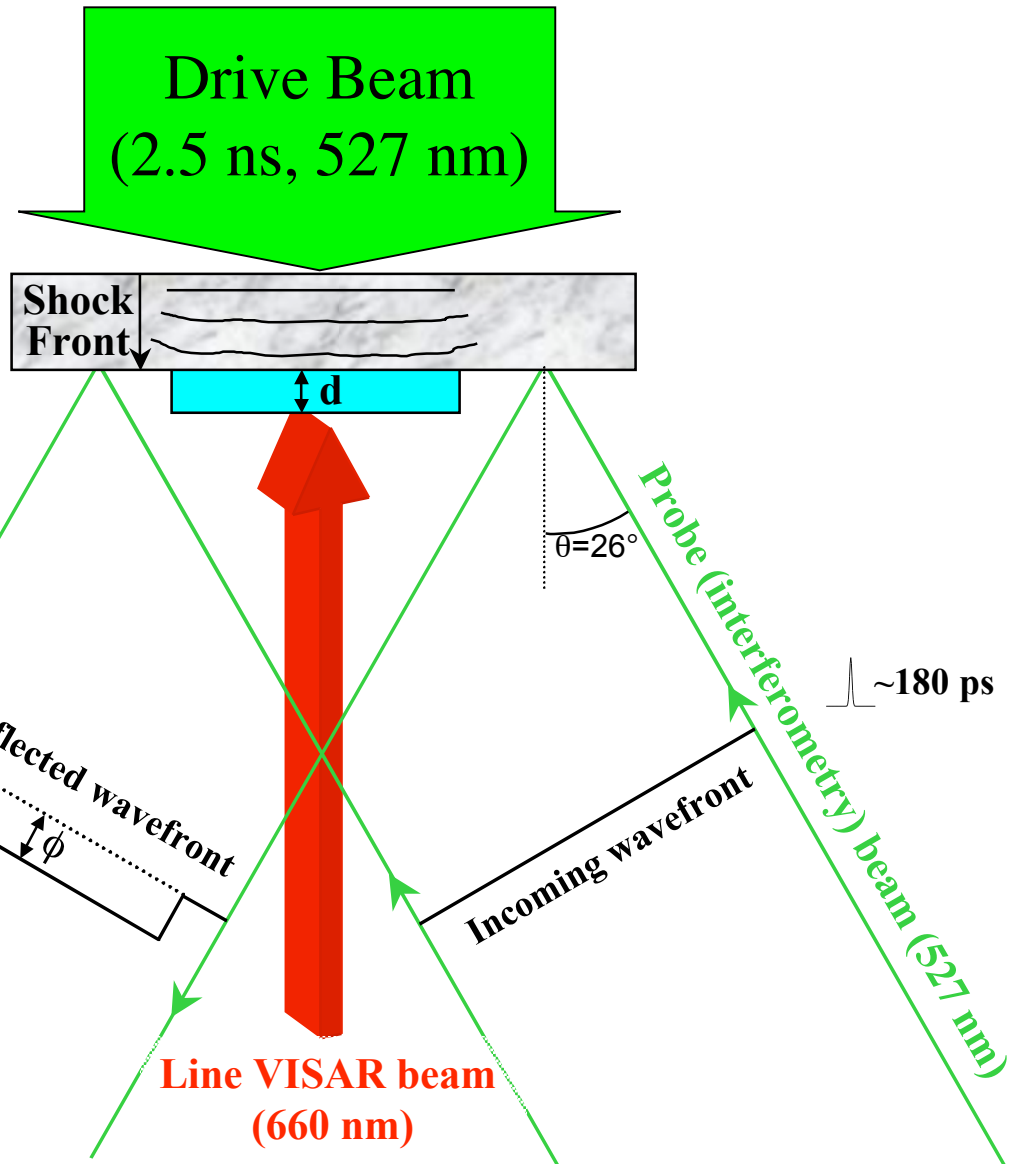
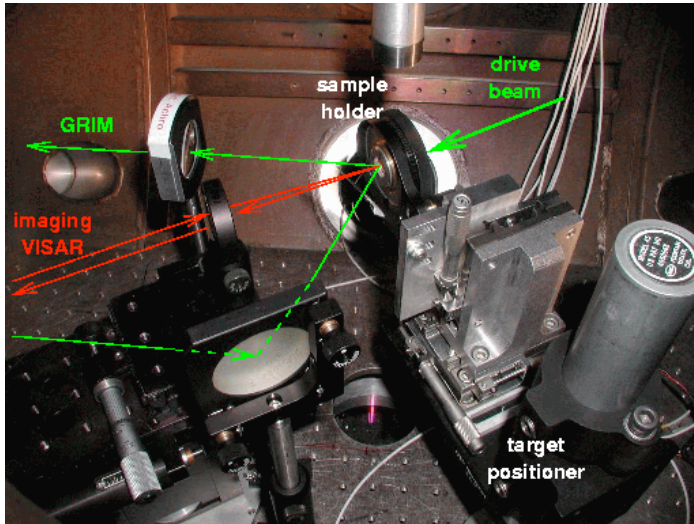
Polycrystal TXD: results



TRIDENT shot 15002: Be foil, 11 GPa shock.

Shocked signal \Rightarrow 7.1 to 18.4 GPa isotropic, or flow stress = 3.4 to 8.1 GPa (uniaxial)
c.f. observed flow stress in foils \sim 2.5 to 6 GPa from velocimetry (thickness-dependent)

Imaging displacement interferometry

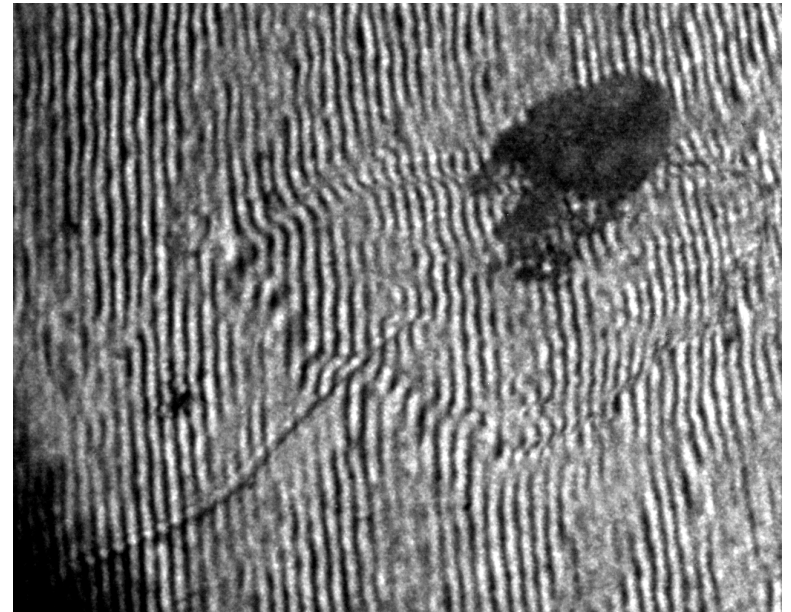
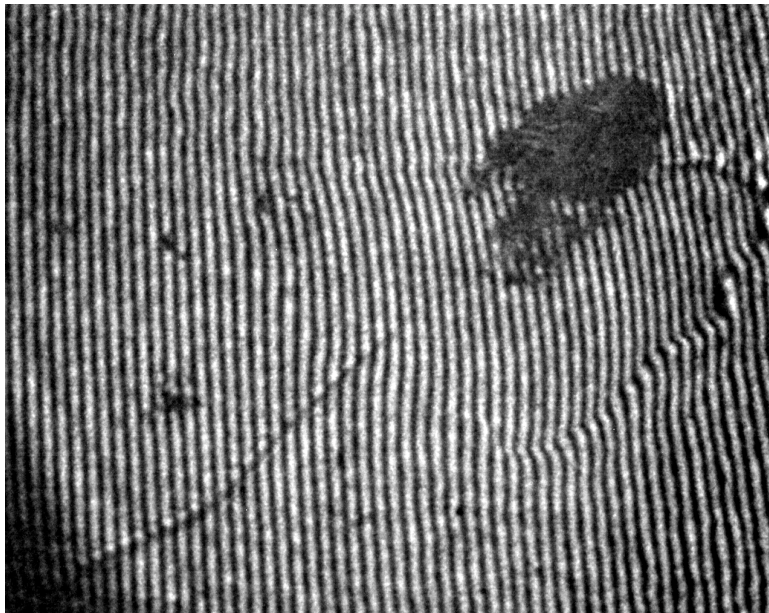


Displacement-phase relationship:

$$d = \phi\lambda / 4\pi \cos \theta$$

*S.R.Greenfield et al,
Proc APS SCCM '03*

Imaging Michelson displacement interferometry



Shown: NiAl bicrystal, $\sim 1 \times 1$ mm, frames 6 ns apart after shock breakout.
Fringes show relative surface relief: included grain is rising out of surface.

Greenfield, Koskelo, Swift, Proc. APS SCCM '03.

Conclusions

- Versatile loading history from TRIDENT: shock / multiple shock / decaying shock / ramp wave
- Diagnostics demonstrated for EOS, flow stress, phase changes, heterogeneous response, electronic and lattice processes
- Diagnostics include imaging surface velocity and displacement, emission and reflection spectrometry, x-ray diffraction
- Published EOS reproduce mechanical data on Be Hugoniot and isentrope
- Discrimination between EOS (thus Lindemann melt predictions): need more temperature data

Acknowledgments

Project scientists:

George Kyrala, Dennis Paisley, Jim Cobble, Tom Tierney, Sheng Luo (P-24)
Aaron Koskelo, Scott Greenfield (C-ADI), Ken McClellan, Darrin Byler (MST-8)
Dan Thoma, Jason Cooley (MST-6), Doran Greening (X-7), Scott Bardenhagen (T-14),
Roger Kopp, Nels Hoffman (X-1), Paul Bradley, Doug Wilson (X-2),
Pedro Peralta, Eric Loomis (ASU), Marcus Knudson (SNL),
Hector Lorenzana, Bruce Remington (LLNL)

Trident staff and P-24 support:

Sam Letzring, Randy Johnson, Bob Gibson, Tom Hurry, Fred Archuleta,
Tom Ortiz, Nathan Okamoto, Bernie Carpenter, Scott Evans, Tom Sedillo

Target fabrication and characterization:

Ron Perea, Bob Day, Art Nobile, Bob Springer (MST-7), John Bingert (MST-6)

Funding and project support:

Allan Hauer, Nels Hoffman, Cris Barnes, Steve Batha (TNX)
STB - LDRD-DR, Bruce Remington (NIF Materials IET)