



# Development of Graphite Fiber Reinforced Magnesium Alloys for Lightweight Mirror Substrates and Zero CTE Metering Structures

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# Development of Graphite Fiber Reinforced Magnesium Alloys for Lightweight Mirror Substrates and Zero CTE Metering Structures: Outline

- ◆ Why Magnesium?
- ◆ Why Mg/Gr?
- ◆ Continuous Mg/Gr Composites
- ◆ Discontinuous Mg/Gr Composites
- ◆ Calculated and measured properties
- ◆ Applications and Manufacturing Methods
  - Mirrors- Discontinuous fiber preforms
  - Optical bench, metering structures
- ◆ Conclusions

# Why Magnesium?

- ◆ Low density: AZ31 alloy: 1.85 g/cc  
(AZ31= Mg 3Al-1Zn)
- ◆ Same specific stiffness as Al, Ti, Fe
- ◆ Compatible with high performance graphite fibers
- ◆ Electrically conductive
- ◆ High thermal conductivity
- ◆ No outgassing in space and orbital environments

# Why Mg/Gr Composites?

- ◆ Low density ~2 g/cc
- ◆ Higher stiffness than Be >70 msi
- ◆ Higher specific stiffness than Be ~1.5x
- ◆ Low CTE/high thermal stability
- ◆ Potentially lower mass structures than Be
- ◆ Low cost
- ◆ Matrix is ductile...No crazing or microcracking
- ◆ Non-toxic

# Constituent Materials

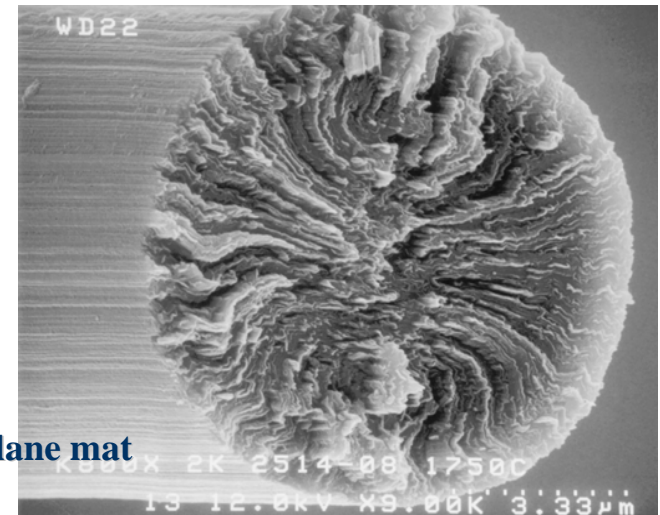
Matrix Alloys	Composition	E (gPa)	CTE (ppm/K)	TC (W/mK)
<b>Mg- AZ-31</b>	<b>Mg- 3 Al- 0.2 Mn- 1 Zn</b>	<b>45</b>	<b>“ 26</b>	<b>96</b>
Mg-AZ-91D	Mg- 9 Al- .13 Mn- 1 Zn	45 (6.5 msi)	26	72
Mg- ZC-61	Mg- 6 Zn- 3 Cu- 0.5 Mn	45	“ 26	122
Al 413-HP	Al-12.5 Si- 0.3Mg	71	“ 21	167

## Graphite Fiber Reinforcement

	<u>E<sub>1</sub> gPa (msi)</u>	<u>E<sub>2</sub> gPa (msi)</u>	<u>CTE ppm/K</u>	<u>TC-W/mK</u>
*CKD-x:	862 (125)	6.9 (1)	-1.5	550
*CKA-x:	965 (140)	6.9 (1)	-1.5	950
**K63B12	862 (125)	6.9 (1)	-1.5	440
**K63712	641 (93)	6.9 (1)	~1.5	140

\*Cytec graphite fibers: 1” chopped, processed into a random, in plane mat

\*\*Mitsubishi continuous fibers



# Discontinuous Reinforced Composites

- ◆ Low cost
- ◆ Continuous control of volume fraction reinforcement
- ◆ Easy to machine into mirror substrates or electronic thermal management packages
- ◆ Near quasi-isotropic properties

# Modified Schapery's Equation for Composite CTE Prediction

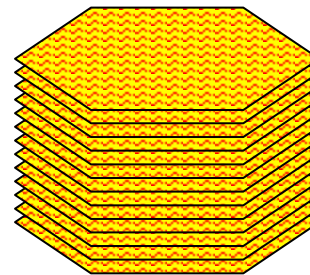
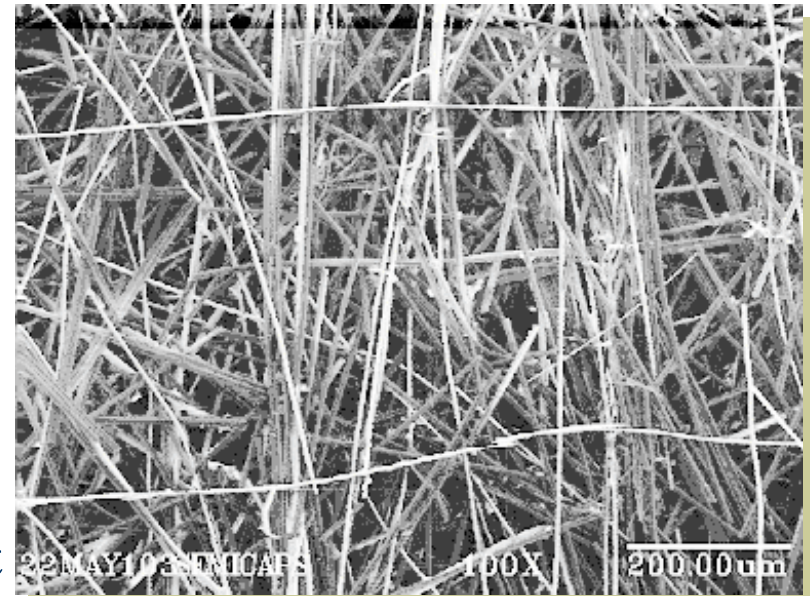
$$\alpha_{cx} = \frac{\alpha_m E_m v_m + \alpha_{fa} E_{fa} \frac{v_f}{2} + \alpha_{ft} E_{ft} \frac{v_f}{2}}{E_m v_m + E_{fa} \frac{v_f}{2} + E_{ft} \frac{v_f}{2}}$$

## Where:

- $\alpha_{cx}$  = coefficient of thermal expansion of composite  
 $E$  = Young's modulus of elasticity  
 $v$  = volume fraction reinforcement

## Subscripts:

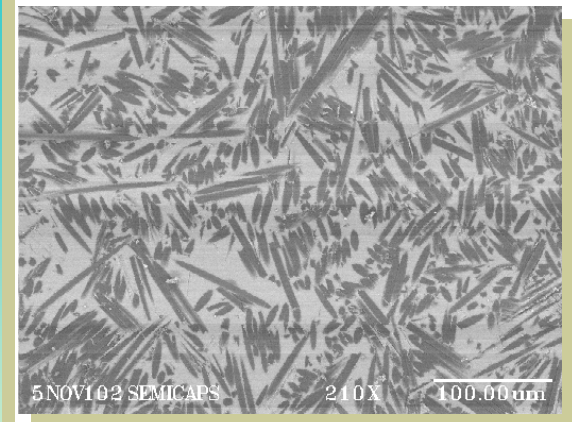
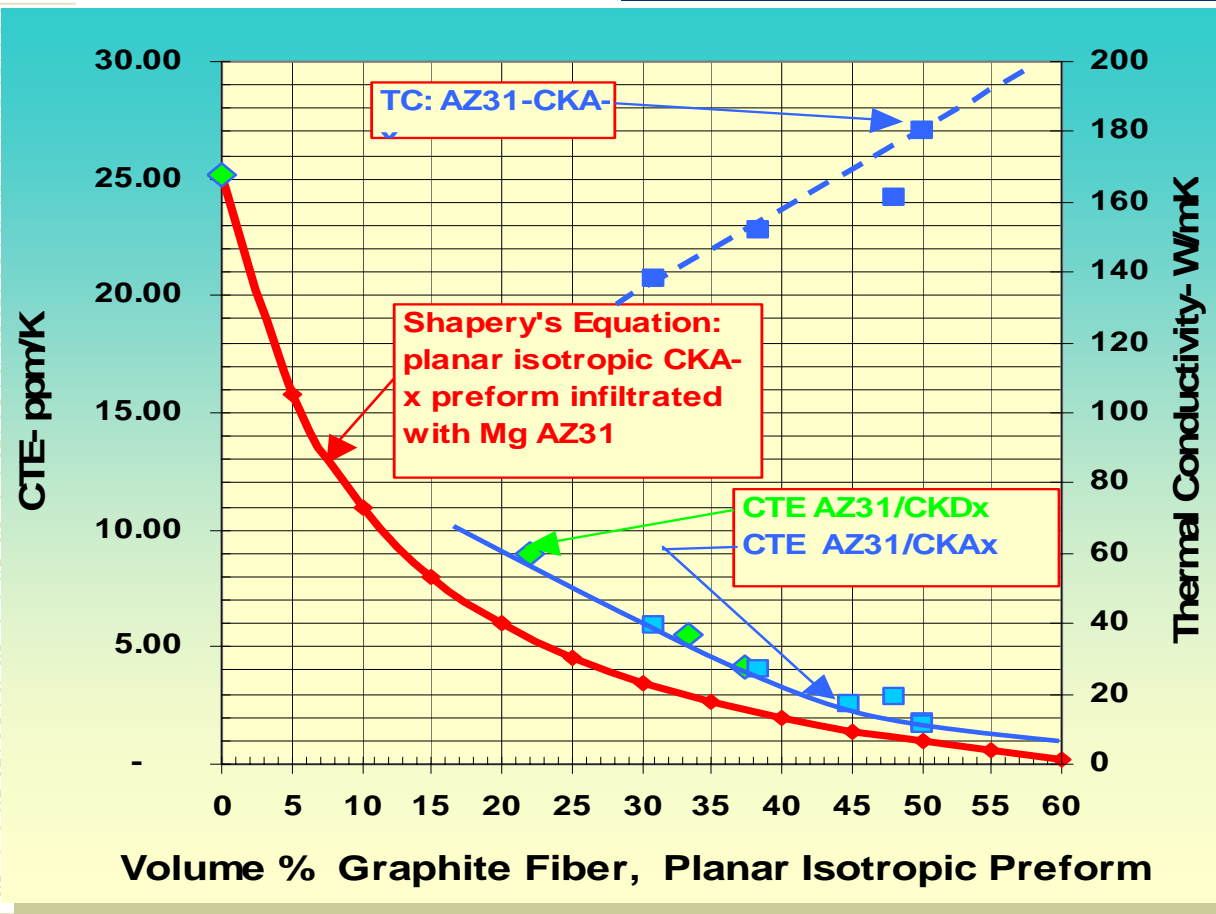
- $m$  = matrix  
 $f$  = fiber  
 $fa$  = fiber axial  
 $ft$  = fiber transverse



Planar-isotropic architecture results from discontinuous fibers randomly distributed in-plane



# Variation of CTE (and Thermal Conductivity) with v/o Compared to Shapery's Equation (and Rule of Mixtures Calculations)

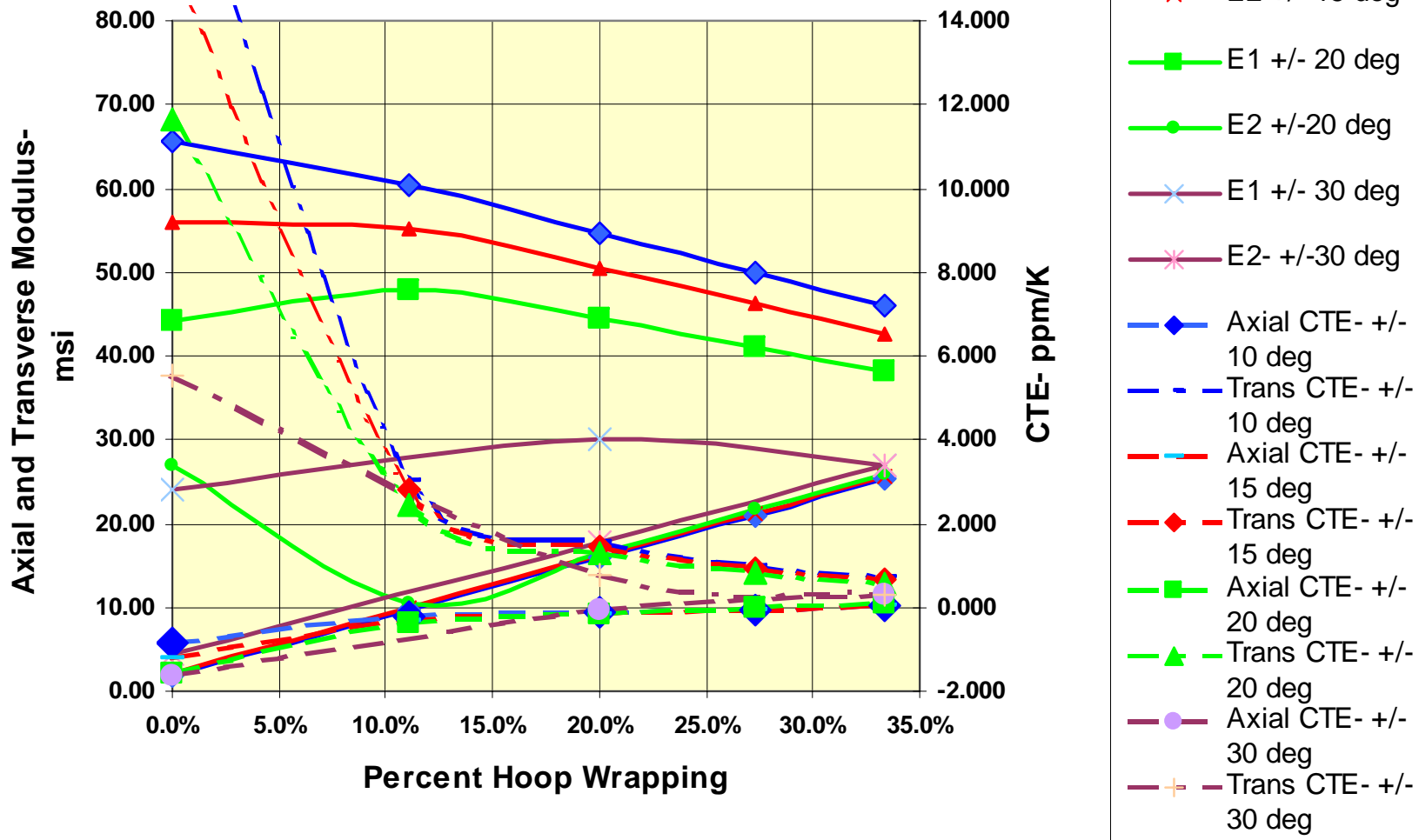


# Continuously Reinforced Angle-Ply Graphite/Mg Composites

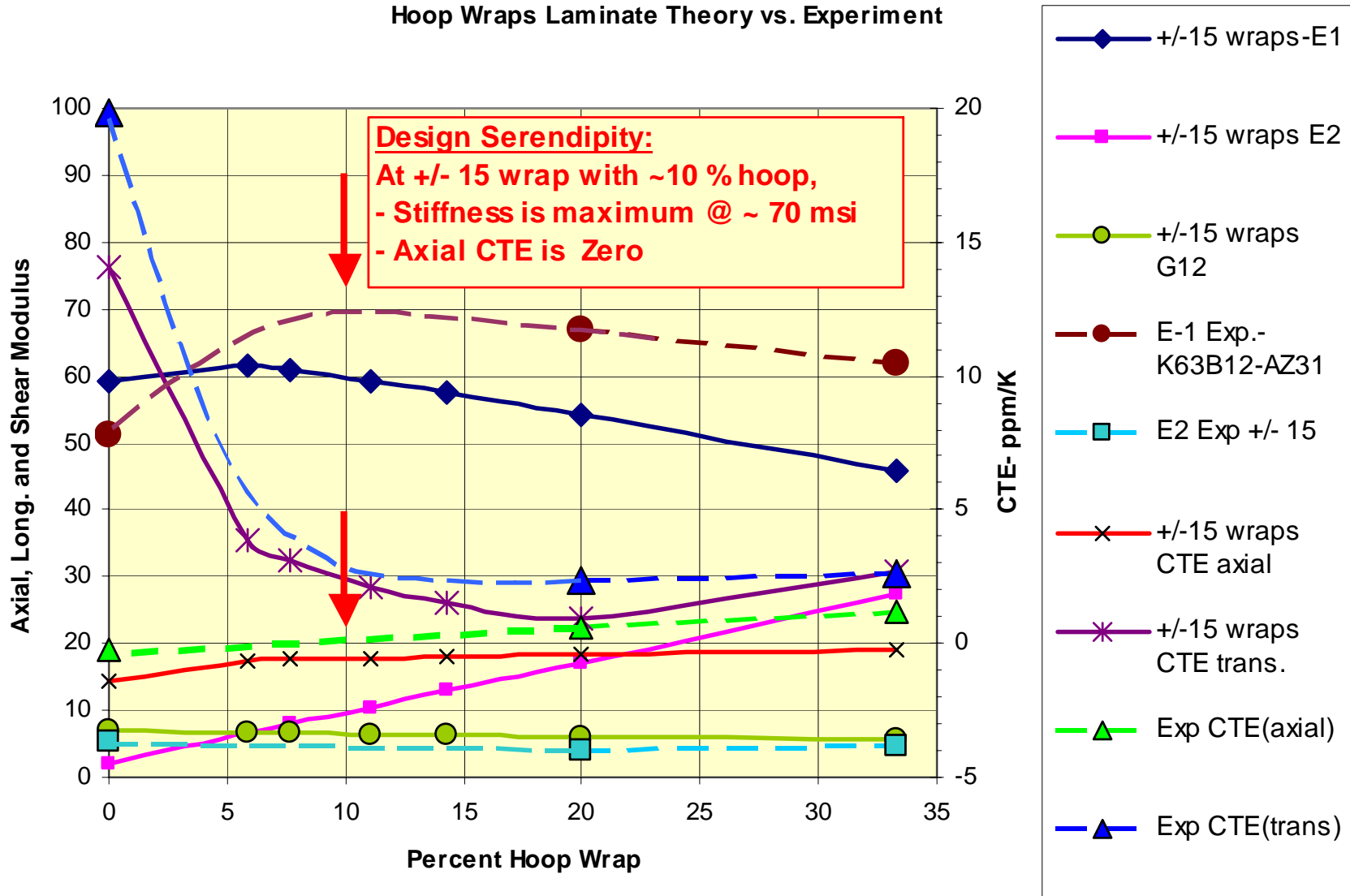
- ◆ High stiffness
- ◆ Low density
- ◆ High strength
- ◆ Low CTE in axial and transverse directions
- ◆ Zero CTE in axial direction easy to achieve
- ◆ Higher specific stiffness than Be
- ◆ Non toxic
- ◆ Lower Cost than Be
- ◆ Electrically conductive/High CTE

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## Laminate Theory Calculations of Properties of Mg AZ31 Reinforced with 55 v/o K63B12 as a Function of Wrap Angle and Percent Hoop Wrap



Variation of Stiffness and CTE of Mg-K63B12 +/- 15 Deg Wraps as a Function of Percent Hoop Wraps Laminate Theory vs. Experiment



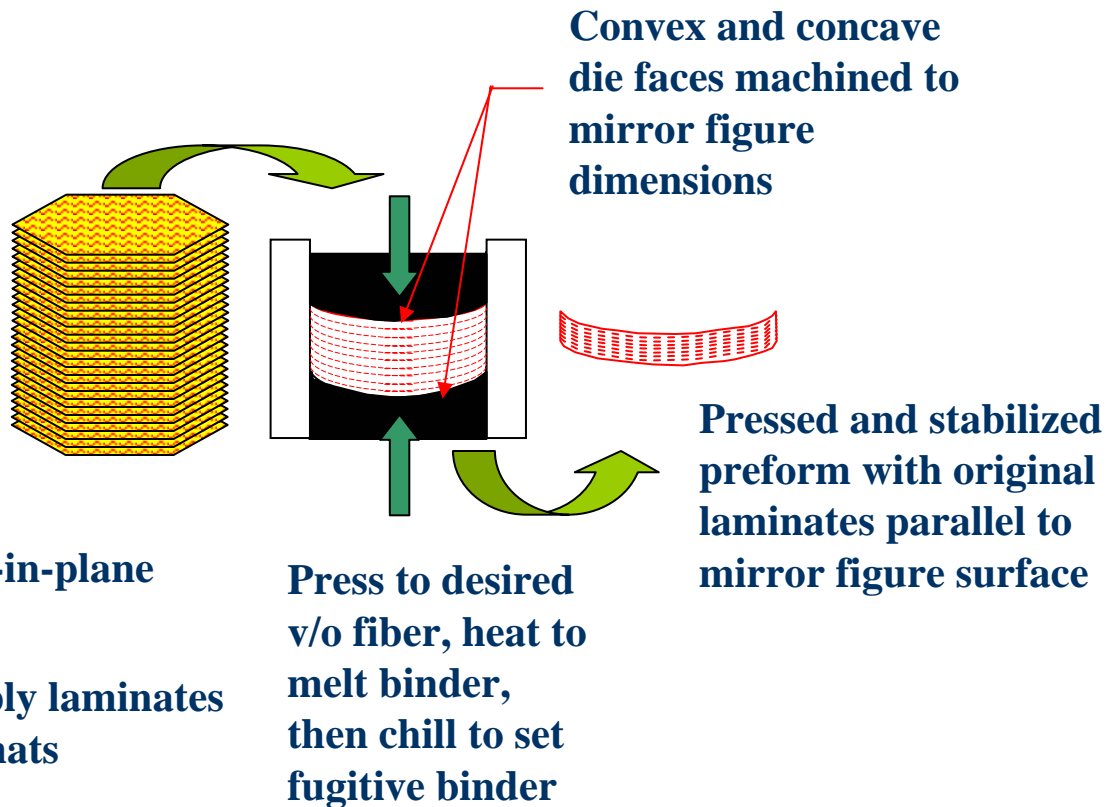
# Fabrication of Continuous and Discontinuously Reinforced Mg/Gr Composite Structures

- ◆ 1) Discontinuously Reinforced Composites:
  - Randomly dispersed in-plane paper-like preform
  - Compress with binder to desired v/o
  - Set binder and load into confining mold
  - Debinder in mold, pressure infiltration cast with Mg and directional solidify for porosity free composite.

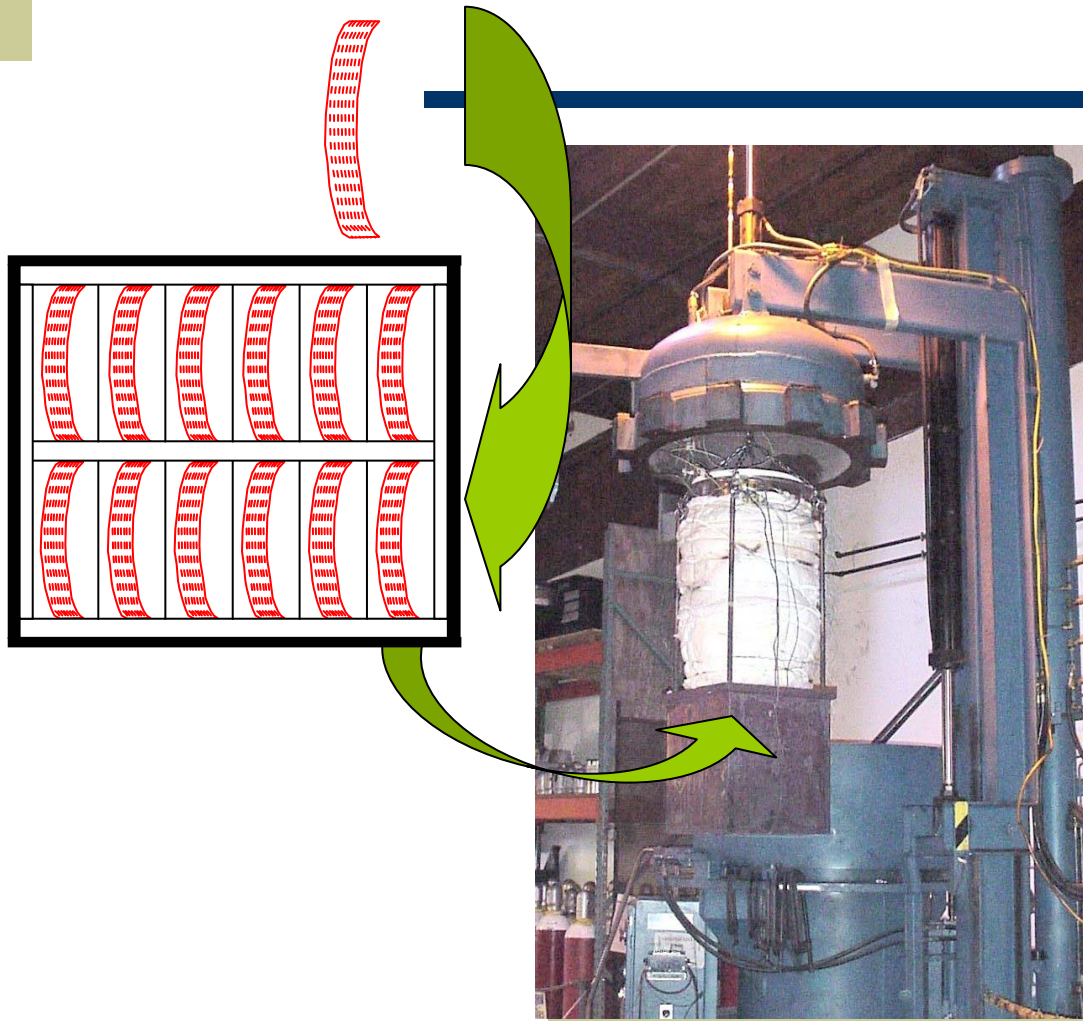
# Discontinuous or Continuous Angle Ply Graphite, (or Hybridized) Fiber Preform Manufacture

Preforms composed of

- Paper mats of random-in-plane lamina or
- Quasi-isotropic angle ply laminates hybridized with paper mats



# APIC<sup>tm</sup> Facility



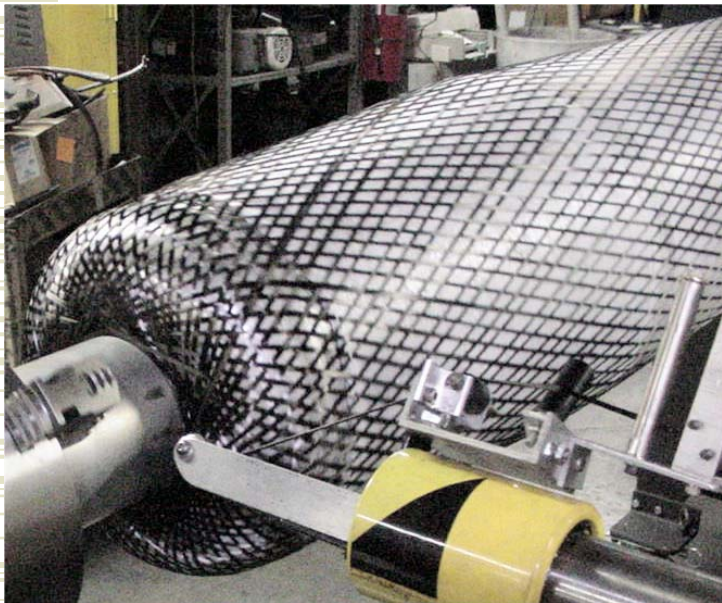
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# Fabrication of Continuous and Discontinuously Reinforced Mg/Gr Composite Structures at MMCC

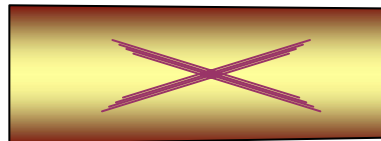
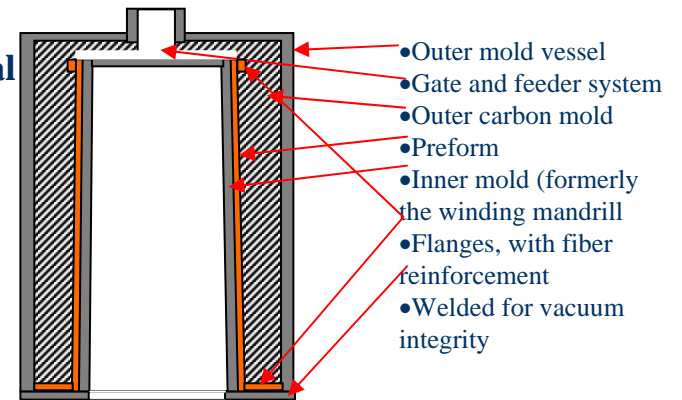
- ◆ 2) Continuously Reinforced Composites
  - Filament wind to desired ply pattern
  - Remove from winding mandrel for panel preform
  - Leave on winding mandrel for tubular structures
  - Load into casting mold, debinder and pressure infiltrate with molten Mg, directionally solidify for porosity free casting.

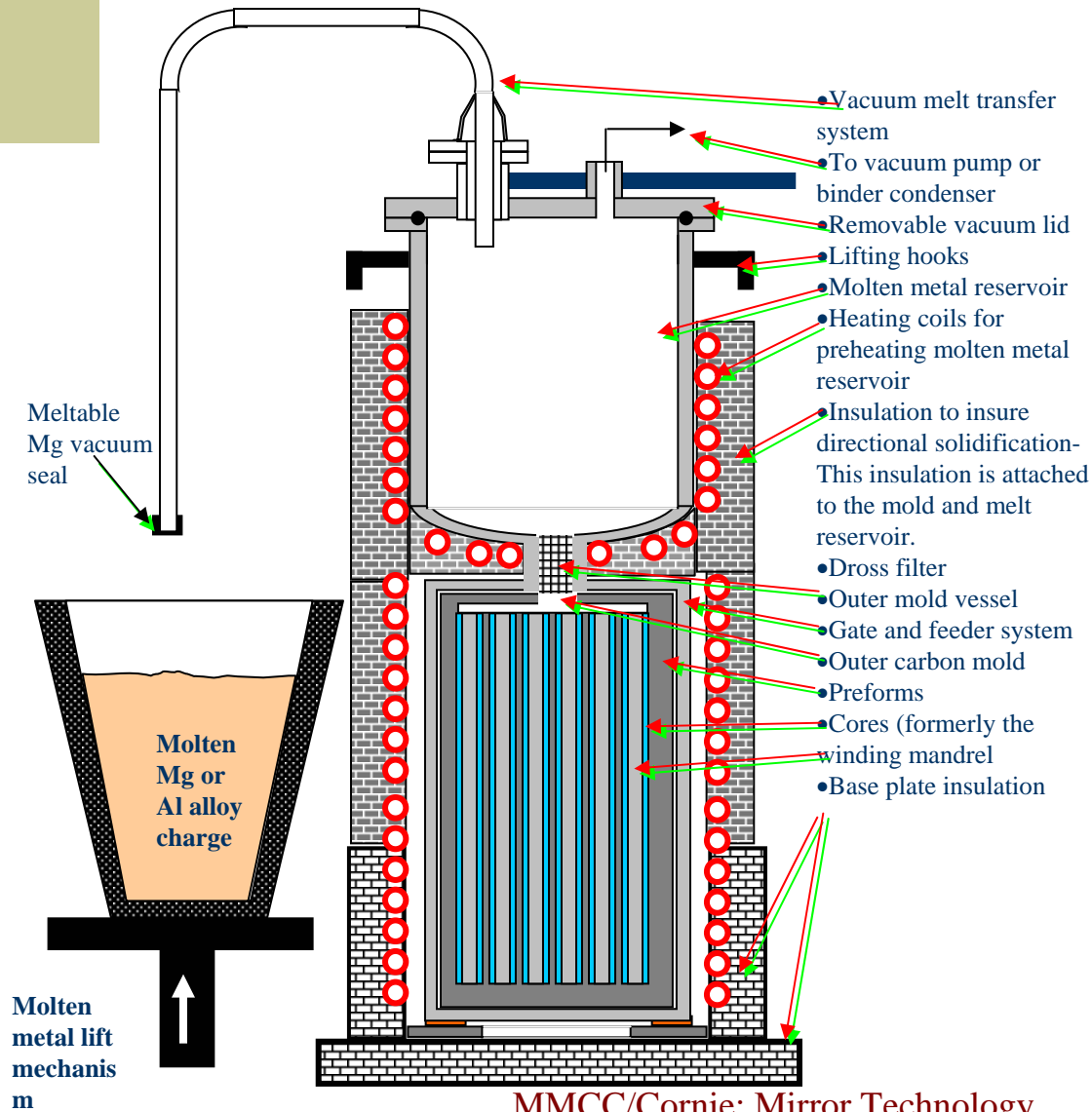


# Tubular Structures Manufacturing Steps:



- Manufacture steel wrapping mandrel/casting core to ID dimensions (compensated for thermal contraction)
- Fit with end plugs and attach to shaft
- Wrap mandrel with Saffil or chopped glass paper
- Wind with desired angle and hoop architecture with fugitive binder





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# Optical Bench/ Metering Structures

- ◆ Filament wound Mg-Gr can be designed to Zero thermal expansion in axial direction with transverse CTE  $\sim 2$  ppm/K
- ◆ Specific Stiffness 1.5 x Be and 9.5 x Invar
- ◆ Egg crate designs are possible with filament wound preforms sliced from the mandrel and laid out flat and cast as panels which are subsequently cut into egg crate patterns

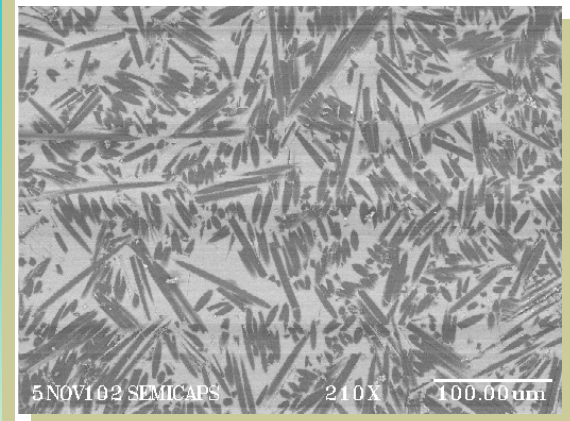
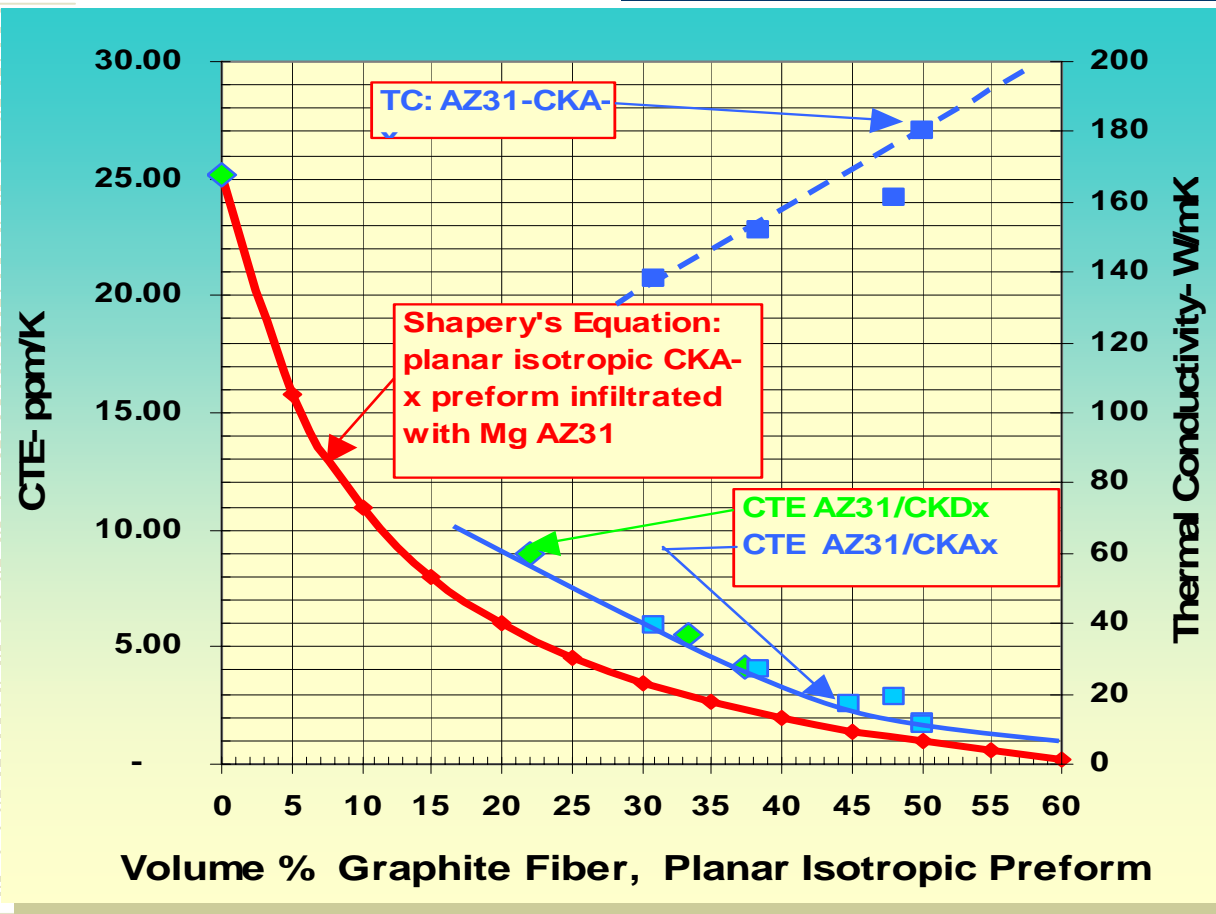
# Metering Structures: Summary

- ◆ Planar isotropic CTE  $\sim 1$  ppm produced for MetGraf Mg mirror substructures. CTE can be specified and controlled by varying volume fraction reinforcement.
- ◆ Structures can be filament wound to produce a specific stiffness  $\sim 1.5 \times \text{Be}$  and CTE = zero
- ◆ Fabrication technology for Mg/Gr is mature and robust

# Lightweight Mirror Development

- CVD design to match the reflective membrane
- Comparative specific stiffness and thermal stability of candidate mirror and membrane materials
- Dimensional stability after thermal cycling
- Thermal seasoning
- CTE matching to Si and SiC membranes
- Membrane attachment and lightweight machining

# Variation of CTE (and Thermal Conductivity) with v/o Compared to Shapery's Equation (and Rule of Mixtures Calculations)

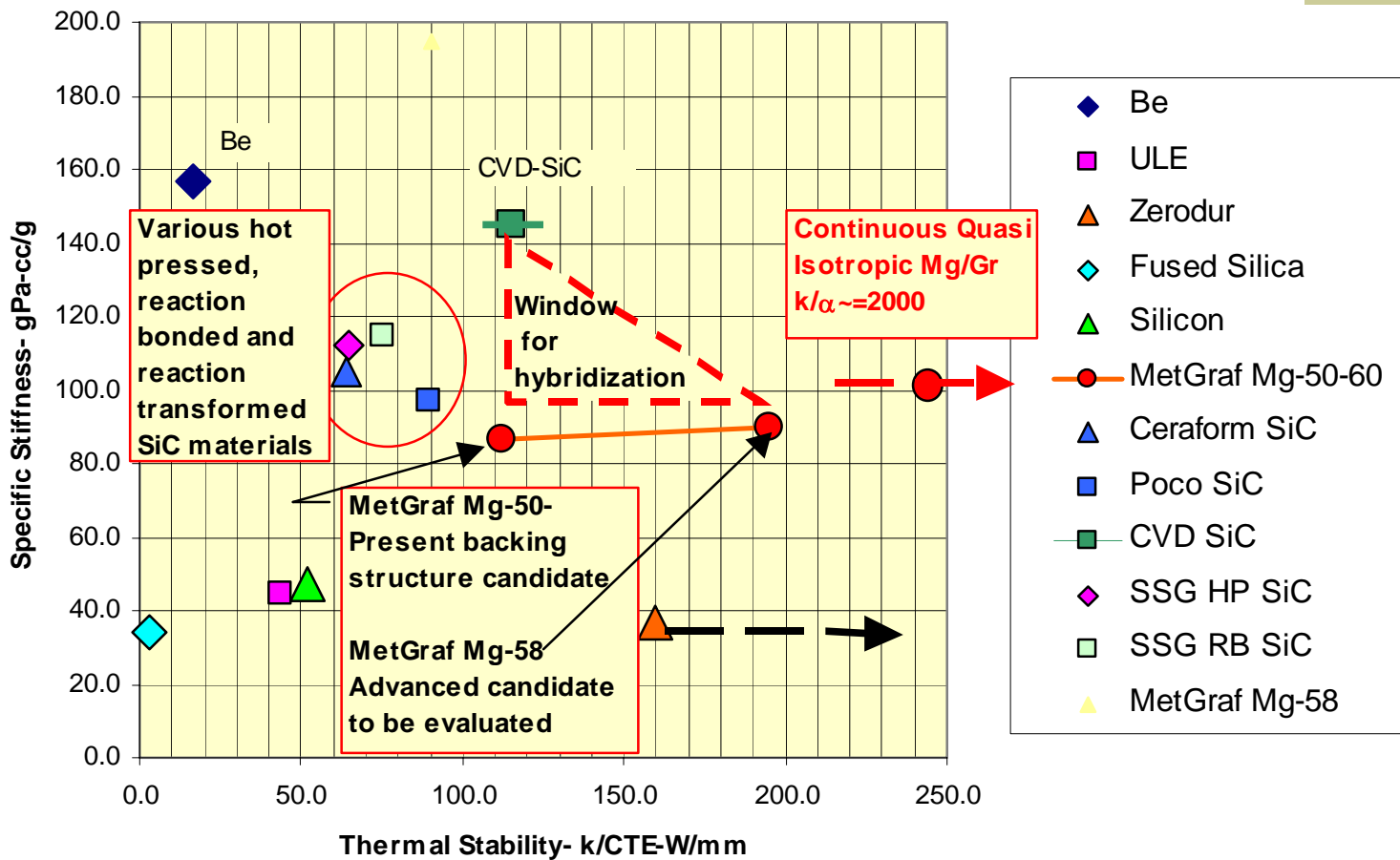


# Properties of Candidate Space Mirror Materials

Material \ Property	Modulus E-gPa	density g/cc	Specific Stiffness gPa-cc/g	CTE ppm/K	Thermal Cond. W/mK	Thermal Stability K/CTE- W/mm	Reference
Mg	44.8	1.77	25.3	26	122	4.7	ASM Metals Hbk
Be	289.6	1.85	156.5	11.6	190	16.4	"
Al	69.0	2.70	25.5	22.7	170	7.5	"
ULE	98.0	2.21	44.3	0.03	1.31	43.7	Xinetics- NMSFC w orkshop 5/01
Zerodur	92.0	2.50	36.8	0.01	1.6	160.0	Poco- NMSFC w orkshop 5/01
Fused Silica	72.0	2.10	34.3	0.5	1.5	3.0	Poco- NMSFC w orkshop 5/01
Silicon	110.0	2.33	47.2	2.4	125	52.1	ASM Metals Hbk
<b>Gr/Mg P-I Phase I</b>	<b>172.0</b>	<b>1.98</b>	<b>86.9</b>	<b>1.67</b>	<b>180</b>	<b>107.8</b>	<b>MMCC Ph. I</b>
<b>Gr/Al Phase I</b>	<b>180.0</b>	<b>2.38</b>	<b>75.6</b>	<b>1.5</b>	<b>220</b>	<b>146.7</b>	<b>MMCC Ph. I</b>
Ceraform SiC	310.0	2.95	105.1	2.44	156	63.9	Xinetics- NMSFC w orkshop 5/01
Poco SiC	248.0	2.55	97.3	1.9	170	89.5	Poco- NMSFC w orkshop 5/01
<b>CVD SiC</b>	<b>448.0</b>	<b>3.21</b>	<b>139.6</b>	<b>2.6</b>	<b>240</b>	<b>92.3</b>	
SSG HP SiC			112.0			65.0	SSG- NMSFC w orkshop 5/01
SSG RB SiC			115.0			75.0	"
Mg/Gr Quasi Isotropic	<b>203.0</b>	<b>2.00</b>	<b>101.6</b>	<b>0.45</b>	<b>200</b>	<b>444.4</b>	JC Lam. Theory + ROM for K
Gr/Al Quasi isotropic	<b>210.0</b>	<b>2.37</b>	<b>88.6</b>	<b>1.14</b>	<b>220</b>	<b>193.0</b>	"

# Performance Comparison for Mirror Candidate Materials

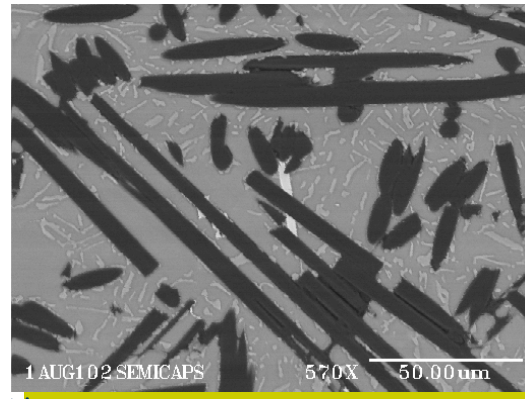
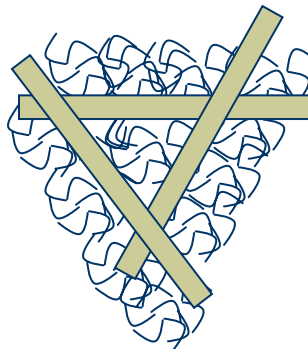
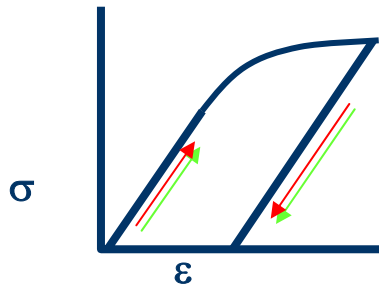
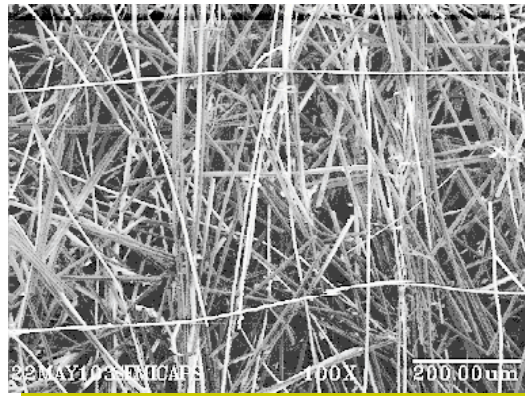
Specific Properties of Candidate Mirror and Hybrid Mirror Materials



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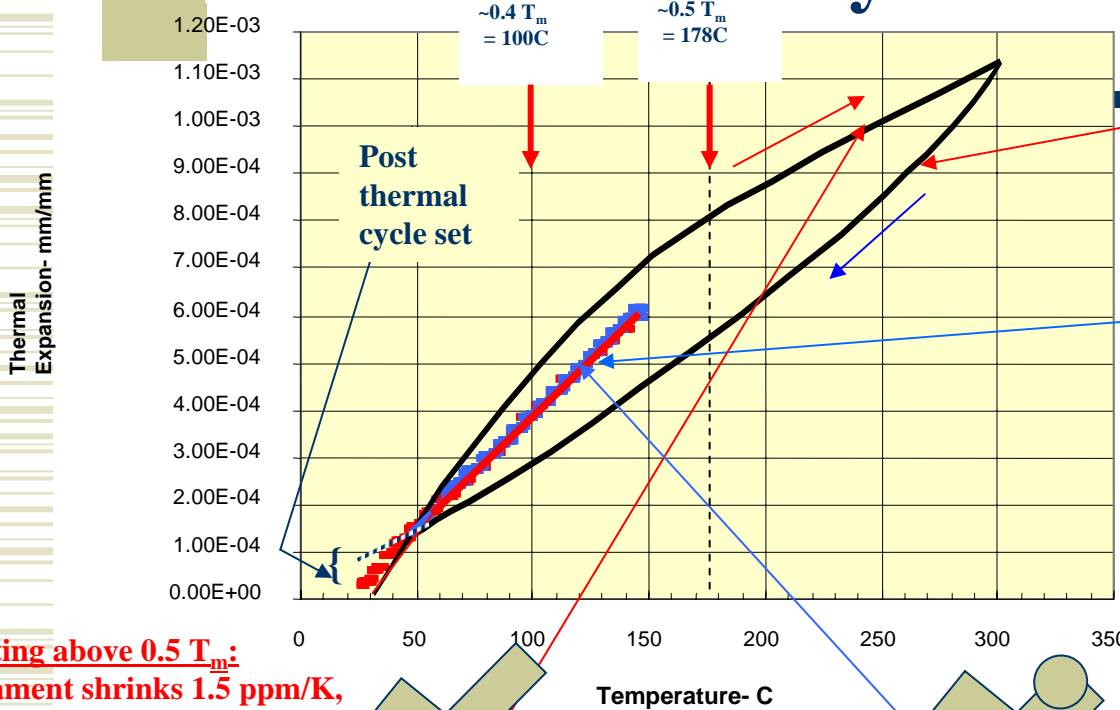


# Cold working: Proposed method of producing dimensional stability



Straining composite to beyond micro-yield leads to work hardening of matrix by creating dislocations

# Illustration that differential expansion induced cold work can eliminate hysteresis



**First post cast cycle:**  
25-300-45C

**Second Cycle:**  
Reheating 25 C to 145C-

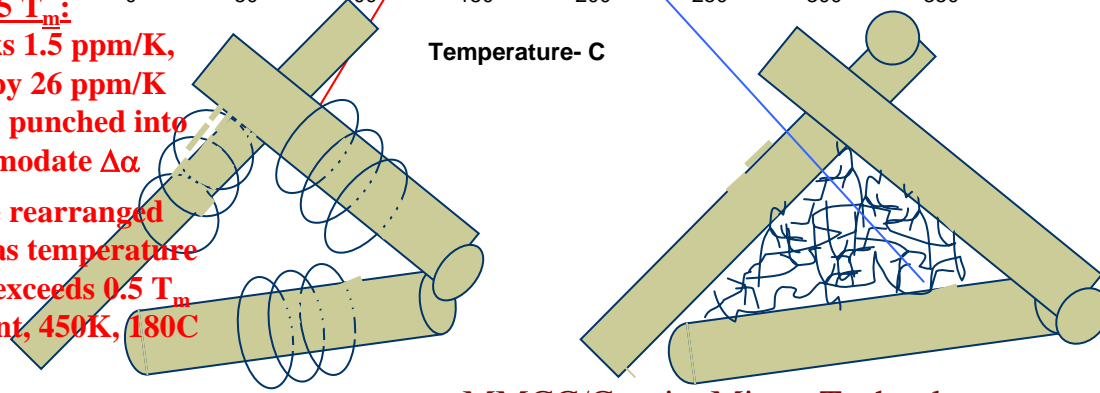
Cooling from 145C to 40C

Cooling below  $0.5T_m$  and thermal cycling below  $0.5T_m$ :

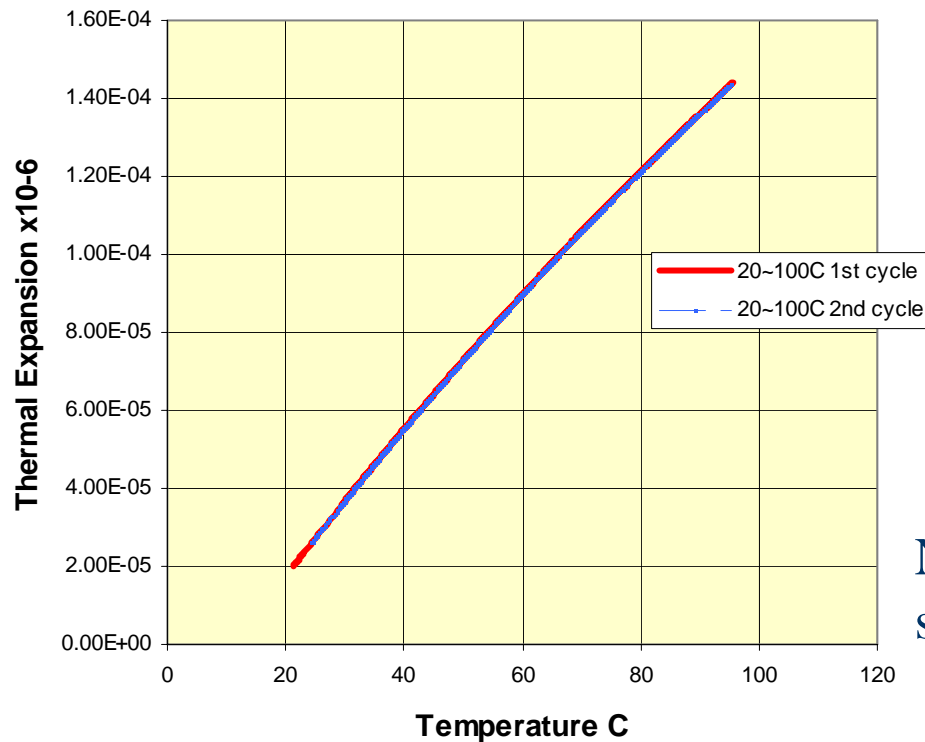
- Filament expands 1.5 ppm/K, matrix shrinks 26 ppm/K
- Dislocations are punched to accommodate  $\Delta \alpha$  of 27.5 ppm/K
- Dislocations are created faster than they are annihilated as temperature decreases below  $0.5T_m$ , leading to work hardening matrix
- Further thermal cycling below  $0.5T_m$  will rearrange dislocations into a stable substructure
- No distortions or "set" after repeated thermal cycling at application temperatures

Heating above  $0.5T_m$ :

- Filament shrinks 1.5 ppm/K, matrix expands by 26 ppm/K
- Dislocations are punched into matrix to accommodate  $\Delta \alpha$
- Dislocations are rearranged (or annihilated) as temperature approaches and exceeds  $0.5T_m$  (half melting point, 450K, 180C)

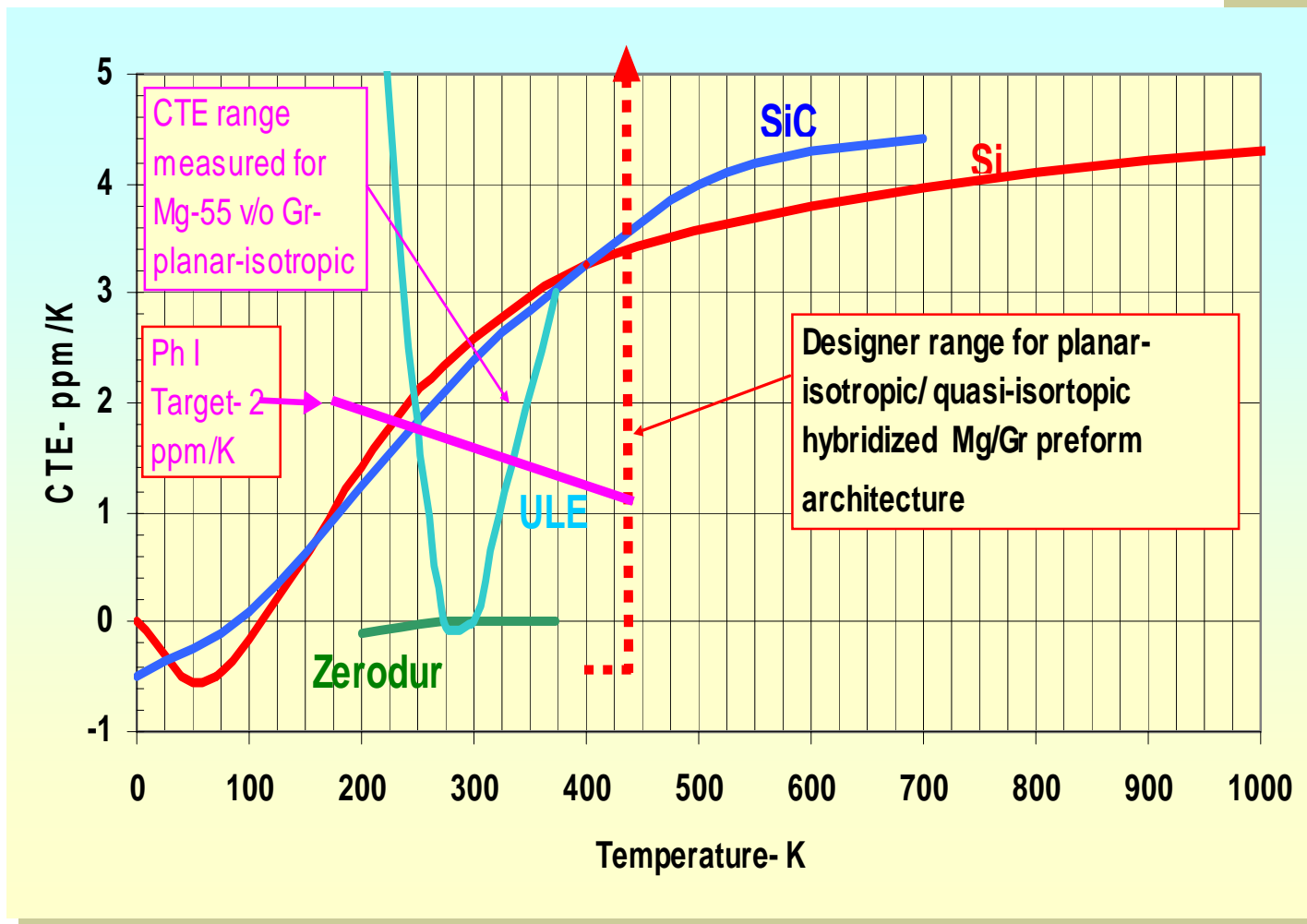


# Two thermal expansion cycles of MetGraf Mg 50 after two thermal seasoning cycles from RT → 125C → -75C → RT.



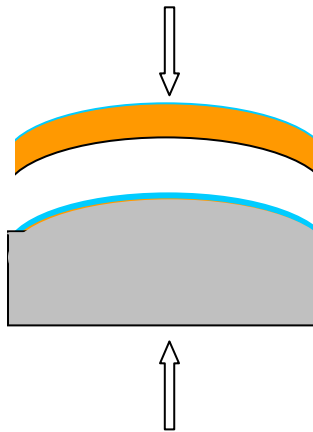
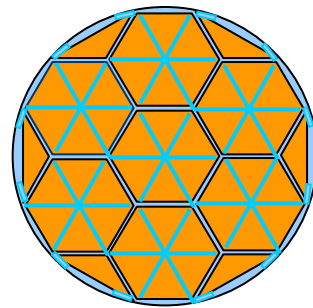
Note lack of thermal set or hysteresis.

# CTE of Silicon and SiC As a Function of Temperature

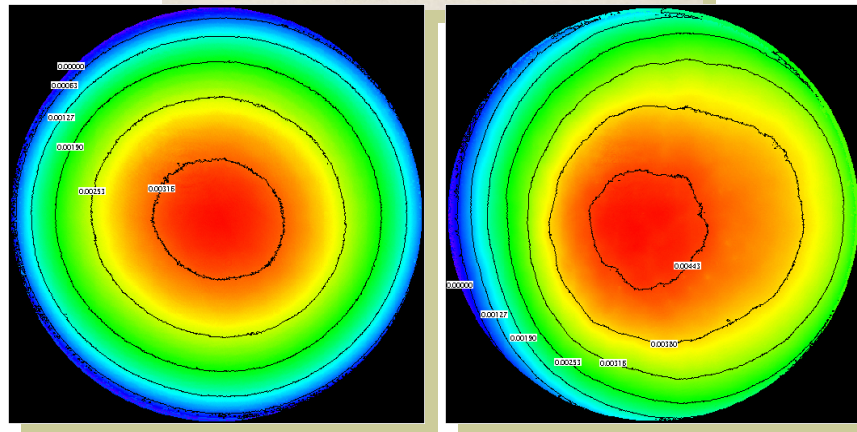


# Three Approaches to Lightweight Low-Cost Mirrors

- 1) Si coat Metgraf-Mg, then polish
- 2) Replicate on master (nanolaminate or polymer) and transfer to MetGraf Mg
- 3) Adhesive or solder bond surface membrane of Si or SiC and polish

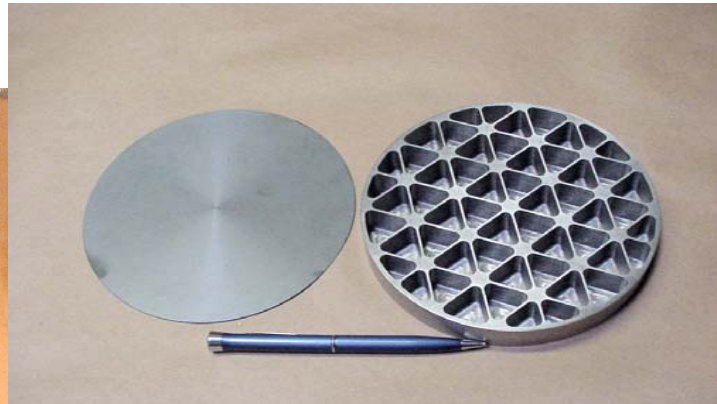
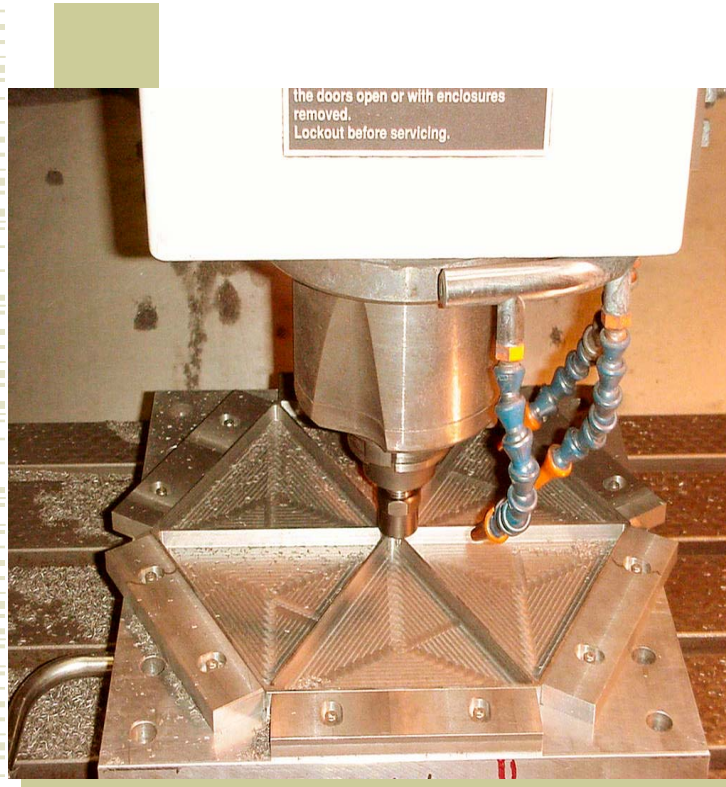


# Polished Si Coated MetGraf Al and Mg



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# Lightweighting and Hybridizing SiC with MetGraf Mg



**Plate 1, 310 mm dia. X 10 mm thick.**

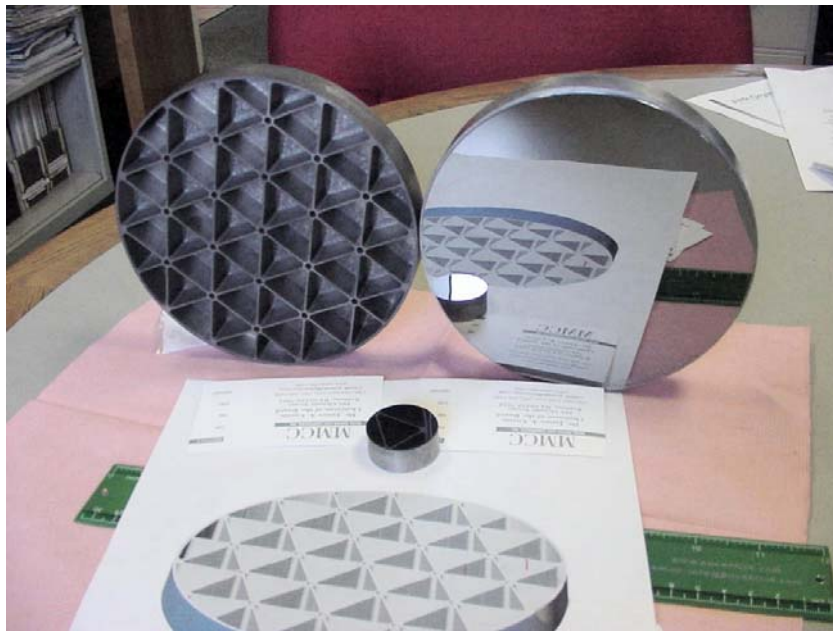
- Total machining time=3 hr
- Face thickness = 2 mm, rib thickness + 2 mm
- Mass = 306 g
- Areal density = 5.04 kg/m<sup>2</sup>

**Plate 2: 152 mm dia x 20 mm thick.**

- **Total machining time = 1 hr.**
- Face thickness =1 mm, Total thickness = 20 mm, Outer rim =2 mm, Inner ribs = 1.0 mm,.
- Mg/Gr = 211 g, Areal density in Mg/Gr = 11.56 kg/m<sup>2</sup>
- Areal density with 0.5 mm SiC attached = 13.29.
- Further lightweighting (thinner outer rim, machined nodes, 0.5 mm face would reduce Areal Density to ~10 kg/m<sup>2</sup>

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# Attachment of Si and SiC Membranes to Lightweight MetGraf Mg Substrates with Pre and Post Polishing Options



Pre-polished Si membrane bonded to lightweight substrate. Areal density =  $9.7 \text{ kg/m}^2$



SiC membrane bonded to Mg/Gr lightweight substrate. Part to be Si coated and polished



# Summary

1. Attachment of Si or SiC membranes to MetGraf Mg substrate is a viable approach
2. Low temperature Si deposition of MetGraf Mg substrate is also an acceptable approach
3. Si membrane hybrids have resulted in less than 10 kg/m<sup>2</sup> areal density
4. MetGraf Mg can be designed to CTE match a wide range of membrane surfaces
5. Thermal seasoning treatments can result in thermal stability over a wide range of operating temperatures