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Fuel cell related materials using in situ neutron diffraction

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Neutrons for Materials Science and Engineering - ASM Educational Symposium

Overview

Ceramic membranes

- Fuel cell electrodes and electrolytes
- Water dissociation
- Gas separation

Materials properties and role of neutron diffraction

- Bulk phase composition
- Mixed ionic/electronic conductivity cation and oxygen vacancies
- pO_2 gradients gas mixtures to control pO_2 on both sides of membranes
- Kinetics time-resolved studies

Examples with current instrumentation

- SFC2 Sr₂(Fe,Co)₃O_{6.5+ δ}
- LSFC $(La,Sr)_1(Fe,Co)_2O_{6-\delta}$
- CY20 $Ce_{0.8}Y_{0.2}O_{1.9-\delta}$

Future prospects

- Higher neutron flux more detail from each point, shorter runs
- Higher spatial resolution
- New analysis capabilities



Mixed-conducting ceramic membranes

Technologically important membranes with unique properties

- High oxygen/hydrogen conductivity along with electronic conductivity
- Long-term structural stability under steep pO₂ gradients
- Typically perovskite-based oxides with oxygen vacancies
- Typical dimensions: 1-3 mm thick (future applications require 1-100 μm)

Cross-cutting research opportunities

- Understand bulk and surface ionic transport in insulating and electronically-conducting materials
- Learn to tailor the properties of materials
- Achieve chemical and thermal stability and surface catalytic properties while maintaining the required transport

Applications

- Solid-oxide fuel cells: e.g., $Ce_{0.8}Y_{0.2}O_{1.9-\delta}$, (La,Sr)(Fe,Co)O_{3- δ}
- Gas separation: e.g., Sr(Fe,Co)O_{3- δ}, (La,Sr)₃(Fe,Co)₂O_{3- δ}
- Hydrogen production: e.g., $Ba(Ce_{0.7}Zr_{0.2}Yb_{0.1})O_{3-\delta}$



Solid oxide fuel cells and water dissociation

Solid oxide fuel cells

- Multi-component assemblies that generate electricity from chemical dissociation
- Wide variety of applications and materials

Water dissociation

- Shifts water decomposition reaction toward dissociation $H_2O = H_2 + \frac{1}{2}O_2$
- e.g., $Ba(Ce_{0.7}Zr_{0.2}Yb_{0.1})O_{3-\delta}$ proton conductor





SOFC components

Complex electrochemistry

- Electrodes "painted" on electrolyte

Cathode

- Reduction, dissociation of O_2
- Operates at comparatively high pO2
- e.g., (La,Sr)(Fe,Co)O_{3- δ}

Anode

- Oxidation, dissociation of e.g. H_2O or H_2
- Operates at low pO₂
- e.g., Ni or Ni/stabilized zirconia

Electrolyte

- Ionic only or O^{2-}/H^+ mixed conduction
- Nanoscale connectivity critical
- Triple phase boundary ion, electron, gas conduction: need to avoid "blockages"
- e.g., Sm-, Gd-, Y-doped ceria







Gas separation

Typical application is conversion of CH_4 to syngas (CO + H_2)

- Membrane activated simply by passing gases across opposing surfaces
- $-pO_2$ gradient generated: $pO_2 = 10^{-0.5}$ (air), 10^{-18} (CH₄)
- Oxygen ions permeate through membrane to react with CH₄
- Single-phase, e.g., Sr(Fe,Co)O_{3- δ} and multi-phase, e.g., Sr₂(Fe,Co)₃O_{6.5- δ}





Membrane properties and role of neutron diffraction

Phase composition

- Most membranes have complex chemical composition
- Some membranes are multi-phase; may vary with pO₂
- Neutrons: bulk measurement tracks composition, phase separation, decomposition in situ

Oxygen vacancy concentration and distribution

- Control conducting properties



- Mechanical stress lattice parameter changes with cation reduction, gradient across membrane could jeopardize mechanical stability
- Neutrons: oxygen is strong scatterer defect location, concentration and ordering
- Neutrons: in situ lattice parameters and peak shapes resolve issues related to stress

Hydrogen / deuterium

- **Neutrons:** $b_{H} = -3.7$, $b_{D} = 6.7$, $b_{O} = 5.8$, $b_{Fe} = 9.5$, $b_{Sr} = 7.0$ fm

Surface oxygen exchange and bulk chemical diffusion

- Neutrons: Time-resolved variation in lattice parameter

Oxygen / hydrogen flux

- Measure gas conversion





Powder diffractometer (GPPD at IPNS)

d-range: 0.3 - 18 Å

Raw (vs. time-of-flight)



Built in 1981 - x70 intensity increase @ d = 2Å, 2θ = 145° (POWGEN3 ~50x current GPPD)

L-CH₄ moderator for cool spectrum Bandwidth chopper to control λ -range Supermirror neutron guide (85 x 22 mm) to boost flux Extended detector range for summing

Time-focused (vs. d)

Datasets in 5-mins to 6 hrs





Controlling atmosphere at high temperature





Phase composition: e.g., Sr₂Fe₂CoO_{6+x} (SFC2)

Candidate for syngas production

- Multi-phase mixed conductor with phase composition dependent on $\ensuremath{\text{pO}_2}$
- Chemical composition changes within each phase
- 15 60 minute runs; slow kinetics





Lattice strain across SFC2 membrane

Perovskite lattice expansion

– From Fe^{4+} to Fe^{3+} and Co^{3+} to Co^{2+} reduction

Possible mechanical strain in gradient

- Large changes for perovskite, layered phase
- Total, weighted by volume fractions, intermediate

Lattice parameters in gradient

- Minimal lattice expansion in gradient

Lattice Expansion during Reduction











SFC2: Effect of pO₂ gradient



Phase stability:

- Layered phase stable in oxidizing conditions
- Presumably important to mixed conducting properties
- Stability extended to lower pO₂ in gradient

Perovskite phase lattice parameter:

- Expansion from Fe⁴⁺ to Fe³⁺ and Co³⁺ to Co²⁺ reduction
- Larger changes in Fe-rich perovskite (with and without gradient)



^a B. J. Mitchell et al., *MRS Bulletin*, 35, 491-501 (2000). ^b Y. Li et al., *J. Am. Ceram. Soc.*, 88 (5), 1244-1252 (2005).



Oxygen vacancies: $La_{0.6}Sr_{2.4}Fe_{1.5}Co_{0.5}O_{7-\delta}$ (LSFC)

Ruddlesden-Popper (RP) series

 $-A_{n+1}B_nX_{3n+1} = AX + nABX_3$

n=2 member in Sr-La-Fe-Co-O system

- $(La_{0.2}Sr_{0.8})_3(Fe_{0.75}Co_{0.25})_2O_x$
- − *I*4*/mmm, a* = 3.87, *c* = 20.18 Å

Lattice expansion

- Fe⁴⁺, Co³⁺ in air, Fe³⁺, Co³⁺ in reduced
- Lattice parameter changes anisotropic; magnitudes high, ±9 x 10⁻³

Oxygen vacancies

- Primarily on one oxygen site
- Variation: δ = 0.3-0.7



Y. Li et al., Solid State Ionics, in press (2007).





1.96

1.94

1.98

2.00

2.02

d spacing (Å)

2,04

2.06

2.08

2.10

- Measured as function of time (30 min. increments) and pO_2 (10^{-0.5} to 10⁻²⁰)



LSFC: **Response to low pO**₂





LSFC: **Response in pO₂ gradient**





LSFC: Calculated pO₂ profile across membrane

Peak from static measurement used as template to reproduce peak profiles in dynamic mode

- From static measurements, each pO₂ value corresponds to unique peak position
- Projected peak profiles for two possible pO_2 gradients not at all representative of actual profile
- Calculated profile suggests very strong gradient at reducing surface

Proprietary coatings typically used on surface





LSFC: *Microstructure of LSFC after neutron experiment*

Membrane intact after experiment - ~10 hrs in gradient Micrographs of air and methane surfaces

- Methane (reducing) surface degraded





Air side

Methane side



Kinetics: Solid-oxide fuel cell electrolyte: Ce_{0.8}Y_{0.2}O_{1.9-δ} (CY20)

Well-studied material

- Intrinsic vacancies from Y-doping; Ce4+, Y3+
- Extrinsic vacancies generated under reduction; some Ce4+ to Ce3+
- Ionic conductor in oxidizing environment
- Electronic conductor in reducing
- Some degree of mixed conduction in between

Behavior under reduction

- No lattice parameter change down to $log(pO_2) = -11$
- Expansion at log(pO2) = -14, -18

Gradient

- -0.5 |CY20| -18: shift to air value
- -5.3 |CY20| -18: near -18 value

Switch-like behavior

- Minor change on oxidizing side
- In and out of mixed-conducting regime?







Kinetics of CY20 reduction

Time-resolved lattice parameter evolution

- Based on linear relationship between lattice parameter and oxygen vacancy concentration
- Parameter plotted is fraction of progress from start to finish

Kinetics parameters

- K_{ex}: surface oxygen exchange constant
- D_{chem}: bulk diffusion coefficient

Static measurements

- Traditional relaxation
- (I): log(pO₂) = -11 to -14
- (II): log(pO₂) = -14 to -18

Gradient measurements

- Not traditional relaxation
- Model not known, although behavior similar
- (III): log(pO₂) = -0.5 |CY20| -18.0
- (IV): log(pO₂) = -5.3 |CY20| -18.0

Trends

- $\rm K_{ex}$ doesn't change; $\rm D_{chem}$ increases with increasing $\rm pO_2$



Li, Maxey, Richardson, J. Am. Ceram. Soc. (2007) in print.



pO₂ gradient progression in CY20





Future prospects in neutron powder diffraction

Higher flux

- More detail from each point, shorter runs
- Local structure changes
- Nanocrystalline components
- Chemical kinetics

Higher spatial resolution

- Multi-component systems
- Mapping across gradients
- Directly probe interfaces

Electrode/Membrane Design

Very challenging. Electrodes need to support several percolation networks: electronic, ionic, fuel/oxidizer/product access.



From BES Workshop on Basic Research for Hydrogen Production, Storage, and Use





New analysis capabilities - e.g. MMM

Modified Maximum entropy with Monte carlo (MMM)

- Developed by Ryoji Kiyanagi (IPNS-ANL)

Methodology

- Utilizes integrated intensities of reflections
- Density is described as continuous distribution, not discrete atomic positions
- Nuclear density is adjusted to minimize constraint equation
- Find most disordered representation consistent with data

Application to ceramic membranes

- Oxygen and/or cation vacancies and oxygen migration
- Local coordination often deviates significantly from octahedral
- Up to 5-10% of oxygens de-localized







Summary

Current neutron powder diffraction instrumentation can probe:

- Phase composition
- Lattice expansion / strain
- Oxygen / hydrogen vacancy concentration
- Diffusion kinetics

Future instrumentation may probe:

- 1-100 µm thick membranes
- Structural relaxation kinetics local structure changes
- Distributions within thin membranes spatially resolved

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