ACTIVATED CARBON COMPOSITES FOR AIR SEPARATION

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ABSTRACT

Large-scale application of oxygen-blown coal gasification, currently the most effective and efficient approach for production of hydrogen from coal is still in need of a cost-effective means of enriching O_2 concentration in air. This project is focused on the development of activated carbon composites for the production of an oxygen-enriched air stream suitable for coal gasification. The method proposed exploits the molecular sieving properties of porous materials, including activated carbon. The potential to separate O_2 and N_2 from air is based on the different diffusion rates of the two molecules in the composite, which has been previously demonstrated. In addition, it has been recently discovered that the peculiar magnetic properties of O_2 molecules induce sizable magnetic field intensity. The effects were attributed to a transition in the magnetic state of O_2 molecules; i.e., from paramagnetic in the gas phase to a non-magnetic state when confined in the micropores of the adsorbent. This effect was observed only when the carbon adsorbent was pre-loaded with super-paramagnetic nanoparticles of Fe₃O₄, and was explained attributed to the local enhancement of magnetic field caused by magnetite particles supported on the microporous adsorbent.

In continuation of the development of composite materials for air separation based on molecular sieving properties and magnetic fields effects, several molecular sieve materials were tested in a flow system, and the effects of temperature, flow conditions, and magnetic fields were investigated. New carbon materials adsorbents, with and without pre-loaded super-paramagnetic nanoparticles of Fe₃O₄ were synthesized; all materials were packed in chromatographic type columns which were placed between the poles of a high intensity, water-cooled, magnet (4 Tesla). In order to verify the existence of magnetodesorption effect, separation tests were conducted by injecting controlled volumes of air in a flow of inert gas, while the magnetic field was switched on and off. Gas composition downstream the column was analyzed by gas chromatography and by mass spectrometry. Under the conditions employed, the tests confirmed that $N_2 - O_2$ separation occurred at various degrees, depending on material's intrinsic properties, temperature and flow rate. The effect of magnetic fields, reported previously for static conditions, was not confirmed in the flow system. The best separation was obtained for zeolite 13X at sub-ambient temperatures. Future directions for the project include evaluation of a combined system, comprising carbon and zeolite molecular sieves, and testing the effect of stronger magnetic fields produced by cryogenic magnets.

INTRODUCTION

Carbon fiber composite molecular sieves (CFCMS) have been developed at Oak Ridge National Laboratory (ORNL) as a novel class of adsorbent which combines tailored gas adsorption properties, good gas permeability, and electrical conductivity.^[1-3] The structure of CFCMS comprises carbon fibers (about 10 μ m in diameter) bonded at their contact points by a continuous carbon skeleton that is electrically conductive. The combination of open structure, microporosity, and electrical conductivity allows the material to be used in a regenerative, electrical swing adsorption (ESA) system.^[4-6] This is the analog of pressure swing adsorption (PSA) where regeneration of the adsorbent is facilitated by resistive heating of the carbon monolith.



Figure 1 - SEM images showing the structure of CFCMS material

Preliminary work demonstrated that CFCMS material has the potential for separating O_2 and N_2 in air. ^[7] Further findings also indicated that the efficiency of separation could be increased by better tailoring the pore size distribution to very narrow micropore sizes (< 2 nm). Based on characterization of a series of CFCMS samples, it was found that O_2 was more rapidly adsorbed on the carbon fiber than N_2 , and with higher equilibrium uptakes, provided the fiber contained a high proportion of very narrow micropores. The conclusion was that CFCMS activated under certain conditions is able to separate O_2 and N_2 from air based on kinetic effects, i.e. the difference in diffusion rates of the two molecules in the network of narrow micropores in the carbon fiber structure. Since O_2 molecules are only slightly smaller in size than N_2 molecules (their kinetic diameters are 3.62 nm and 3.80 nm), achieving selectivity requires a carbon molecular sieve with very narrow and well controlled pore size distribution in the critical range of 3-5 nm. Although controlling the pore size of carbon materials is achievable, the almost perfect separation selectivity offered by zeolites under favorable conditions is hard to match.

Addition of a magnetic field was also investigated in an attempt to further enhance the potential of molecular sieving effects. On theoretical grounds it was hypothesized that adsorption of O_2 should be influenced by magnetic fields whenever adsorption is accompanied by a transition between paramagnetic and diamagnetic states. Although the literature on magnetic field effects on adsorption and desorption is scarce, several examples show that the occurrence of magneto-adsorption-desorption effects strongly depends on the nature of solids and gases, temperature, and the intensity of magnetic fields. ^[8-12] An appropriate combination of these factors could lead to a separation process based on selective adsorption from a mixture of gases. In this context it has been reported that separation of O_2 and N_2 in high magnetic fields is in principle plausible in a steady magnetic field through adsorption of paramagnetic O_2 molecules into micropores of porous ferromagnetic solids. ^[13]

In order to verify this theoretical prediction, a series of measurements of O_2 adsorption isotherms was carried out in static, equilibrium conditions.^[14] In preliminary tests it was found that the amount of O_2 adsorbed at equilibrium was lower (by about 6% at 0.1 MPa) in presence of a weak magnetic field (0.8 Tesla) than in absence of the magnetic field. Although small, this magnetodesorption effect was sizable and observed repeatedly at room temperature for activated carbon fibers loaded with 10 wt % of superparamagnetic magnetic nanoparticles (Fe₃O₄, average size 16 nm). The results of these preliminary tests motivated continuation of the project for additional verification and confirmation of the effect at higher magnetic field intensities and in dynamic conditions; in the new series of tests reported here, a flow system, rather than static, was used to better simulate the conditions of air separation for practical applications. Tests for air separation in absence of a magnetic field were also carried out.

DISCUSSION OF CURRENT ACTIVITIES

MATERIAL PREPARATION AND CHARACTERIZATION

Carbon Fiber Composite Molecular Sieves (CFCMS)

A series of CFCMS composites were used as support for loading a dispersed magnetic phase, magnetite. The CFCMS materials were selected based on their microporous structure and BET surface area values.^[15] The surface and porosity properties were characterized by adsorption of N₂ at 77 K and by adsorption of CO₂ at 273 K.^[16] All adsorption measurements were made using the Autosorb 1 instrument (Quantachrome). Adsorption data were further processed to calculate the pore size distribution of porosity and surface areas using the density functional theory (DFT) in the non-local density approximation. ^[16-19] The DFT software available from Quantachrome was used for calculations. Figures 1 and 2 show the results of pore size distribution in form of cumulative and differential plots. The results obtained from N₂ adsorption at 77 K (lowest accessible pores ~ 10 Å) and CO₂ adsorption at 273 K (pores between 3 and 12 Å) are overlapped in both plots.



Figure 1: Cumulative pore volume distribution as a function of pore widths. Data calculated by DFT method from N₂ adsorption and CO₂ adsorption are plotted on a common scale.



Figure 2: Differential distribution of pore volume as a function of pore width. Data calculated by DFT method from N₂ adsorption and CO₂ adsorption are plotted on a common logarithmic scale.

The results presented in Table 1 include the surface area values calculated from N_2 adsorption data using the BET equation, and from the N_2 and CO_2 adsorption data using the DFT method; the pore volume calculated from N_2 and CO_2 adsorption data using the DFT method; and the average pore width calculated from the same adsorption data using the DFT method. Comparing the results reveals several differences between the three samples:

- Sample NG-4 has a relatively large surface area based on BET result (1123 m^2/g) and DFT result from N₂ adsorption (923 m^2/g). In contrast, CO₂ adsorption reveals lower surface area (600 m^2/g), suggesting blockage of pore entrance by bulky oxidized surface groups, that may have been formed during the activation process.
- Sample TIM-2 has a slightly lower BET surface area (964 m^2/g) and the DFT results for N₂ adsorption (960 m^2/g) and CO₂ adsorption (1090 m^2/g) are not much different. For this sample, both molecular probes used as adsorbate (N₂ and CO₂) give essentially the same pore volume (0.34 0.39 cc/g), which is a strong indication that the material is largely microporous. Indeed, the average pore width calculated by DFT method is 9.8 Å (based on N₂ adsorption) or 5.2 Å (based on CO₂ adsorption).
- Sample K2-B has a very small surface area value based on N₂ adsorption data (73 m²/g from BET equation; 85 m²/g from DFT method). This contrasts with the large surface area measured by CO₂ adsorption (640 m²/g based on DFT method), and is a strong evidence that the material contains very narrow micropores that cannot be accessed by N₂ at 77 K, but are accessible to CO₂ at 273 K. With an average pore width of 7.5 Å (based on N₂ adsorption) or 3.5 Å (based on CO₂ adsorption), the K2-B material is expected to have the most pronounced molecular sieving properties.

Distribution of porosity by pore width is different for the three materials. While all carbons have narrow micropores between 5 and 7 Å (Fig. 2), the corresponding volume is larger for sample TIM-2 than for

samples K2-B and NG-4. In addition, samples TIM-2 and NG-4 have large micropores in the range of 7 - 20 Å; this range of micropores is missing from sample K2-B.

	Surface area			Pore volume		Average pore width	
CFCMS	BET N2	DFT N2	DFT CO2	DFT N2	DFT CO2	DFT N2	DFT CO2
	m2/g	m2/g	m2/g	cc/g	cc/g	А	А
NG 4	1123	923	600	0.476	0.195	9.4	5.2
TIM 2	964	960	1090	0.394	0.344	9.8	5.2
K2-B	73	85	640	0.035	0.185	7.5	3.5

Table 1 Surface area and porosity characteristics of base CFCMS materials selected for tests

Magnetite-loaded CFCMS

As explained in the Introduction, previous results have shown that moderate intensity magnetic fields affect the adsorption of O_2 at equilibrium (at room temperature and maximum 0.1 MPa) on magnetite-loaded microporous activated carbon composites (CFCMS). ^[14] The effect was explained ^[20] on thermodynamic grounds after Ozeki and Sato ^[13] assuming that the magnetic state of O_2 confined in narrow micropores of activated carbon changes from paramagnetic (for free O_2) into random magnetism (for O_2 clusters confined. The decrease of magnetization manifested in presence of an applied magnetic field is expected to induce an increase in pressure (i.e., desorption) of O_2 . The effect is too weak to be observed on pure microporus adsorbents (such as CFCMS), but was observed in static conditions with magnetite-doped CFCMS, because of the local enhancement of magnetic fields in nanopores induced by superparamagnetic Fe₃O₄ nanoparticles. ^[14]

In order to verify the occurrence of this effect in dynamic conditions, a series materials loaded with Fe₃O₄ were synthesized. The carbon supports selected were two granular mesoporous carbons derived from wood (WV A-1500 and SA-30, both from MeadWestvaco) and the CFCMS sample NG-4. The cold precipitation method used previously ^[21] was employed for loading magnetite on these carbons. The method consists essentially in precipitation of Fe₃O₄ from aqueous solutions containing Fe(NO₃)₃ and FeSO₄ in the stoichimetric ratio Fe^{III}: Fe^{II} = 2:1 necesary for formation of the mixed oxide Fe^{III}₂O₃.Fe^{II}O. The precipitation agent is concentrated ammonia solution and the precipitation occurs at pH 10-11. No heating is necessary. In contrast with the ceramic method for synthesis of magnetite, which involves high temperature processes and results in a highly sintered product, the cold synthesis method preserves the nanosized colloidal particles of magnetite.

It was found that the above procedure can be used with the CFCMS sample NG-4, yielding uniformly loaded carbon particles (about 1 - 1.5 mm in size) and a clear residual solution that contained only traces of iron salts. The Fe-NG-4 sample (10 % Fe₃O₄) was obtained by this method. X-ray analysis showed the presence of a Fe₃O₄ phase with small average crystallite size. However, the precipitation method described above could not be used with the wood-derived mesoporous carbons because of their high phosphorus content. It was found that these carbons have been activated with phosphoric acid, and the residual phosphorus on their surface perturbed the precipitation reactions leading to formation of Fe₃O₄ in presence of ammonia as described above. Rather than leading to homogeneous deposits of magnetite on carbon, preferential complexation of Fe^{1II} by surface phosphorus caused deposition of Fe₂O₃ (non magnetic) and precipitation of ferrous hydroxide in the aqueous phase.

Zeolite

A granular sample of zeolite X13 was also used for air separation tests in the flow system. Before packing in a chromatographic-type column, zeolite beads (1.5 - 2 mm) were pretreated at 140 °C for dehydration.

Experimental setup

All adsorbents in form of granules (1-2 mm) were packed in chromatography-type columns (copper tube, 50 cm long, ¼ inch diameter) which were rolled into a 3-coils spiral (5 cm diameter) that was placed flat between the poles of a water-cooled electromagnet (Applied Magnetics Lan, model 2V1). The maximum field intensity between the magnet poles was 1.5 Tesla.

Before packing in chromatography-type columns, all adsorbents were pre-dried overnight at 140 °C. After packing in the column all adsorbents were conditioned overnight in a flow of He (50 cc/min) at 150 – 200 °C. After conditioning the adsorbent, the flow of He was adjusted to the desired value for air separation tests (15 – 30 cc/min). The tests were performed by injecting controlled volumes of air through a septum upstream of the adsorbent column, and analyzing the composition of the gas exiting from the column.

In a first series of tests the gas composition analysis was performed with a Hewlett Packard 5890 series II gas chromatograph (GC) equipped with thermal conductivity (TC) detector. The chromatograph was operated by Agilent ChemStation software for GC systems. It was found that the difference between thermal conductivities of oxygen and nitrogen is not large enough to afford separate identification of the components in a mixture using the TC detector. To correct that, the GC analysis of gas composition downstream the adsorbent column was replaced later by analysis by mass spectroscopy. A Vacuum Technology Inc. mass spectrometer (MS) was used, which was operated by the manufacturer's software (VTI). The instrument was connected through a flow controlling valve and capillary tube to a derivation of the main gas line downstream the adsorbent column. This latter setup was proven more efficient for separate detection of oxygen and nitrogen concentration breaking through the column after air injection, even though the two components were poorly separated on the column.

AIR SEPARATION BASED ON MOLECULAR SIEVING EFFECTS

Carbon fiber composite molecular sieves (CFCMS)

Figures 3 shows examples of mass spectrometer signals for N_2 (blue line) and O_2 (red line) recorded after the breakthrough of air following an injection of 5 cc in a flow of He (30 cc/min) through a 50 cm long column filled with CFCMS TIM-2. The tests were done a various temperatures and at constant flow rate, which was vented out at atmospheric pressure. The pressure shown on this plot (in the range of 10^{-5} Torr) is the actual pressure in the mass spectrometer chamber.



Figure 3: Example of breakthrough nitrogen and oxygen mass spectrometer signals measured for sample TIM-2 after injection of 5 mL air at various temperatures. Flow rate of He carrier was 30 cc/min.

The result of Fig. 3 shows that the peaks corresponding to N_2 and O_2 exiting the column were overlapped. The area under each peak corresponds to the concentration of that component in the mixture and reflects the air composition (roughly 21 vol % O_2 ; 79 vol % N_2 , traces of Ar, CO_2 H₂O etc). In order to better evaluate the effectiveness of $N_2 - O_2$ separation, the mass signals for N_2 and O_2 were normalized to the peak maximum of each component; the results for four different temperatures are shown in Fig. 4.



Figure 4: Normalized mass spectrometer signals for N_2 (blue) and O_2 (red) breaking through a 50 cm column packed with CFCMS TIM-2 material after injection of air at various temperatures. The flow rate of carrier gas (He) was 30 cc/min.

This test result shows that (1) separation of $N_2 - O_2$ is possible, with the N_2 peak being delayed with respect to the O_2 peak; (2) the delay of N_2 peak increases as the temperature is lowered below room temperature; (3) the efficiency of separation was not high enough in the test conditions selected, but could be improved by adjusting the flow rate, length of column, and the size of injection.

Figure 5 shows the results of a similar test with the CFCMS material K2-B.



Figure 5: Normalized mass spectrometer signals for N₂ (blue) and O₂ (red) breaking through a 50 cm column packed with CFCMS K2-B material after injection of air at various temperatures. The flow rate of He was 15 cc/min.

A common trend observed by comparing the two sets of results is that lowering the temperature induces a gradual delay of the nitrogen peak with respect to the oxygen peak. The effect of temperature is greater for the TIM-2 material CFCMS sample for which the N₂ peak is delayed more at lower temperatures. As shown in Fig. 2, TIM-2 has the largest volume of micropores in the 5 - 7 Å range (which is the range of pore widths efficient for molecular sieving separation of O₂ and N₂); larger micropores and small mesopores are also present. For this material there is little separation at room temperature, but separation is enhanced at lower temperatures. In contrast, although separation occurs even at room temperature on the CFCMS K2-B material, lowering the temperature causes the overlap of the two peaks. This material has a very narrow porosity distribution, with narrow micropores are absent. The results of Table 1 also show that the access of N₂ to the internal porosity is more restricted on this material. The combination of

these factors determine the occurrence of $N_2 - O_2$ separation at room temperature on the K2-B sample, which is taught to be caused by its narrow pore distribution. The reason why separation selectivity is gradually lost at low temperatures (Fig. 5) is probably blockage of narrowest pores by selective adsorption of gas molecules at low temperature.

Zeolite

Zeolite 13X, with internal channel openings between 7.5 - 9 Å, is an industry standard for air separation and gas purification based on molecular sieving. ^[22,23] A sample of zeolite 13X (beads, 1.5-2 mm) was used in this project as a reference material. A temperature effect similar to that shown in Fig. 3 for CFCMS, though more clearly defined, was observed for this zeolite (Fig. 6). Although the column length used in this test was shorter (30 cm), air separation was more efficient, and improved greatly with lowering the temperature. However, common trend, also observed with CFCMS materials, was seen: lowering the temperature causes a delay of the nitrogen peak with respect to the oxygen one. This observation shows that the separation mechanism is the same for all three materials, i.e. it is based on kinetic separation during diffusion in narrow nanopores.



Figure 6: Mass spectrometer signals showing breakthrough of N₂ (blue) and O₂ (red) on zeolite 13X after injection of 5 mL air at various temperatures on. Flow rate of He carrier was 30 cc/min.

INFLUENCE OF MAGNETIC FIELDS ON DYNAMIC AIR SEPARATION

All tests on effects of magnetic fields on dynamic air separation were carried out using gas chromatographic (GC) analysis; the instrument was equipped with a thermoconductivity detector and the carrier gas was He. Various materials, packed in chromatography-type columns as described above, were investigated in dynamic conditions; this included CFCMS samples TIM-2 and K2-B, magnetite-loaded CFCMS (Fe-NG-4), and zeolite X13 for reference. The flow conditions and the size of air injection were varied in an attempt to improve the separation. All tests were done at room temperature and atmospheric pressure.

When carbon materials (pure or loaded with Fe_3O_4) were used, the delay of the N₂ peak was not large enough as to afford separate detection of O₂ and N₂ peaks using the TC detector of the GC system. However, when a zeolite packed column was used, the nitrogen peak was delayed more compared with the oxygen one, and two distinct peaks were clearly identified by the TC detector (Fig. 7). For this system, several tests were done in an attempt to detect magnetic field effects. The zeolite column was placed between the poles of the water-cooled magnet, and identical air injections were done repeatedly with the magnetic field switched on and off. No differences were observed in the breakthrough times or the magnitude of the two peaks (Fig. 7). Based on this result (and on many others, not reported here) the effect of magnetic fields on dynamic air separation on zeolite 13X was not confirmed.



Figure 7: Example of air separation on zeolite 13X at room temperature (red – oxygen; blue – nitrogen). The flow rate of he carrier was 30 cc/min and the injection size was 0.1 cc. No detectable differences were seen between the cases with and without magnetic field (1.5 Tesla) applied.

CONCLUSIONS

The findings reported here confirm that CFCMS materials have the potential for separating O_2 and N_2 from air on the basis of the different diffusion rates of the two molecules in the composite. The results show that in dynamic conditions, nitrogen diffusion through the extended pore system of microporous adsorbents is slower than that of oxygen, and become seven slower as the temperature is lowered. Carbon fiber molecular sieves have potential for N₂ - O₂ separation, provided better tailored porosity is developed, with pore widths comparable to molecular dimensions. This depends on identifying and demonstrating alternative techniques of activation, and using new carbon precursors (such as lignine) that are known to produce extensive networks of uniformly narrow micropores. The effect of magnetic fields on air separation was not confirmed at room temperature for zeolite 13X and for Fe₃O₄-loaded CFCMS in dynamic conditions. At this moment it is unclear whether this conclusion can be generalized, given the fact that this finding in dynamic conditions contradicts previous results obtained in static (equilibrium) conditions. These data require more sophisticated measurements, using much higher intensity magnetic fields, to verify the preliminary results and determine whether magnetic fields can be exploited for separation of O2 and N2 from air. Furthermore, as current results show, efficient air separation is achievable on zeolite. There is a large potential for further development by combining air separation on zeolite with second-stage purification or oxygen enrichment on carbon molecular sieves, notably CFCMS, to exploit the electrical swing adsorption capability of these materials.

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