Joining and Processing Issues for Ferritic/Martensitic Steels

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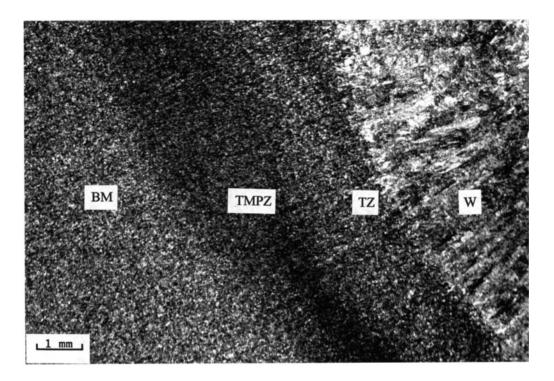


Ferritic/Martensitic Steel Welds: Necessary for Fusion Structure

- •High-Cr ferritic/martensitic (F/M) steel welding processes are established.
 - -Steels can be welded by conventional techniques— SMAW, GTAW, GMAW, TIG, SAW, etc.
 - -Electron beam and laser welding also developed
 - -F/M steels are more difficult to weld than austenitic stainless steels
 - •Pre-heat and post-weld heat treatment required
 - •Complicated microstructure of F/M weldment affects properties and failure mechanisms



Weldment Structure of a Ferritic-Martensitic Steel is Complicated

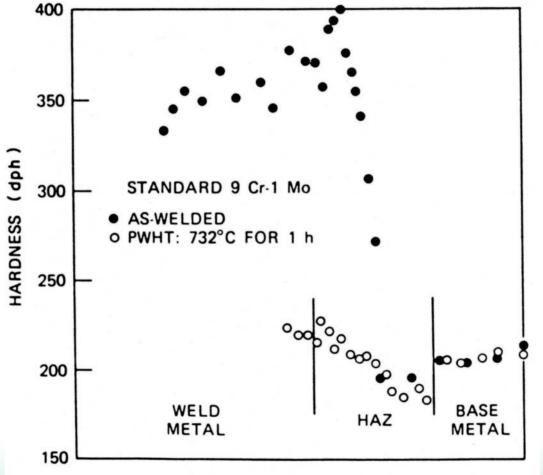


BM: Base Metal, TMPZ: Tempered Zone

TZ: Transformation Zone, W: Weld Metal

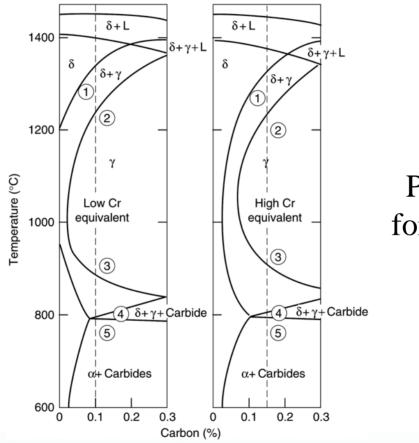


Hardness Varies with the Microstructure of Weldment





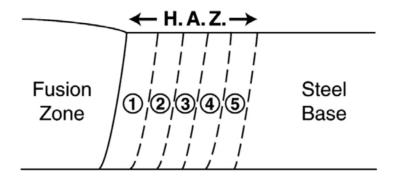
Microstructural Zones Are Determined by Heating-Cooling Process



Phase Diagram for High-Cr Steel



Five Different Microstructural Regions Form in the Heat-Affected Zone



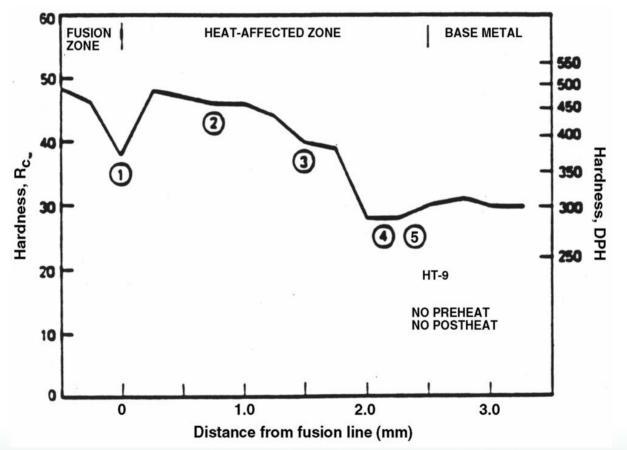
Fusion Zone (FZ): $T > T_m$

Heat - Affected - Zone (HAZ) [as-welded]:

Region 1	$T_m > T > T_{\gamma\delta}$	$\gamma + \delta \rightarrow Martensite + \delta$
Region 2	$T_{\gamma\delta} > T > Ac_3$	Coarse grained $\gamma \rightarrow$ Martensite
Region 3	$T_{\gamma\delta} > T > Ac_3$	Fine grained $\gamma \rightarrow$ Martensite
Region 4	$Ac_3 > T > Ac_1$	$\gamma \rightarrow$ Martensite + Overtempered Martensite
Region 5	$Ac_1 > T > T_T$	Overtempered Martensite

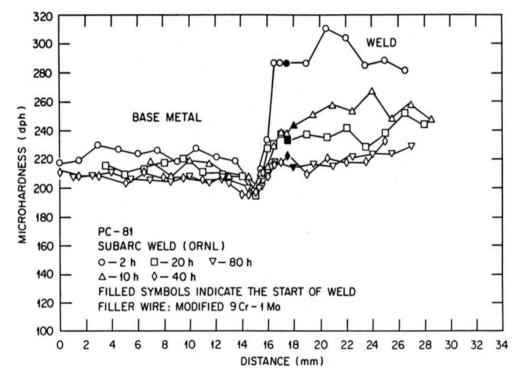
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Hardness Across Weldment Determined by Microstructure



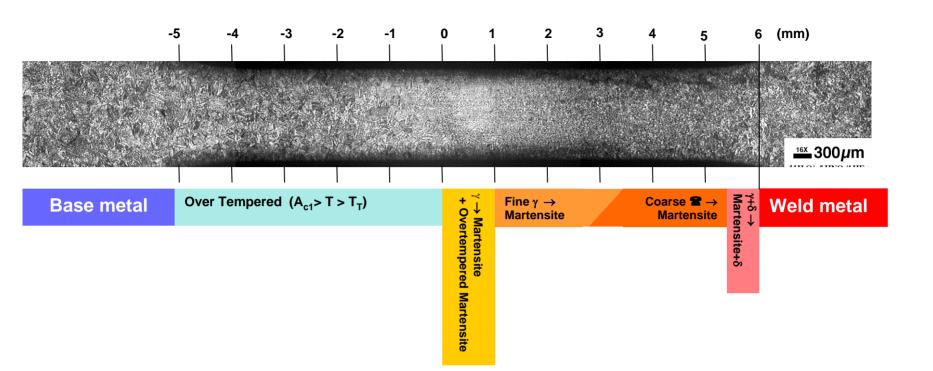


PWHT Tends to Smooth Out Hardness (Strength) Differences



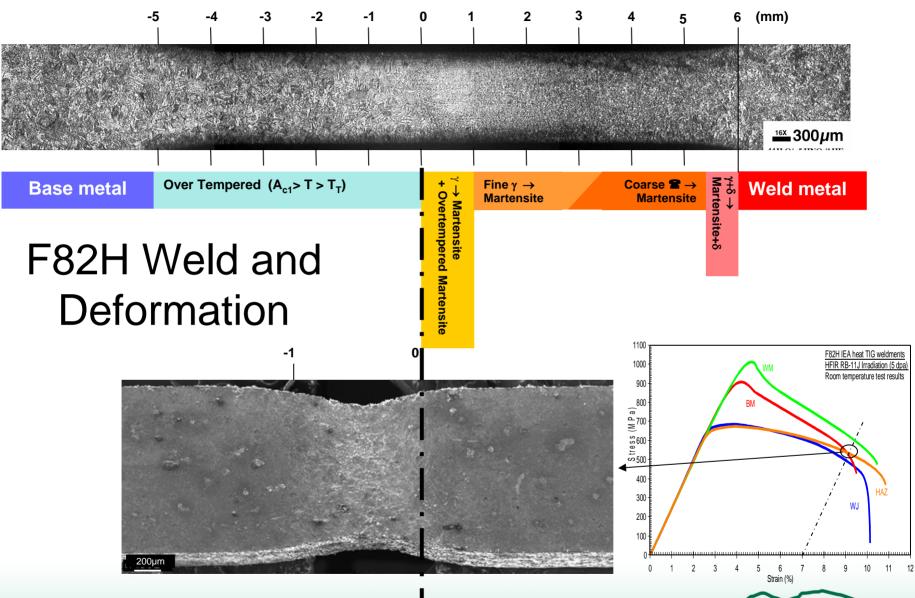
•Dip in hardness of HAZ remains after PWHT





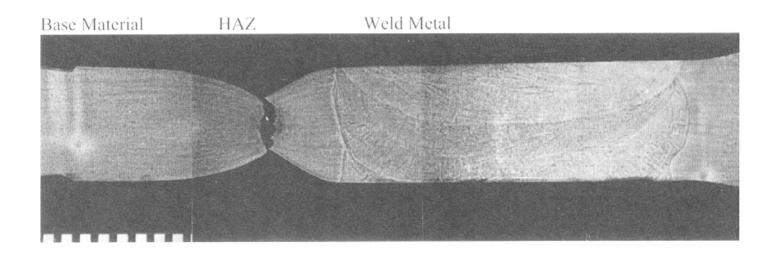
F82H Weldment







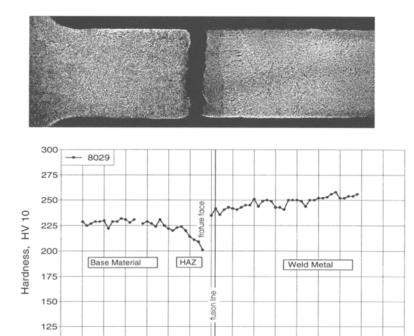
Failures in Base Metal or Over-Tempered Region at High Strain Rate



•Ductile failure occurs for tensile test and high-stress, short-time creep test



Low-to-Moderate Stress Failures Can Occur with Little Ductility



Creep Failure

Length of Specimen, mm

40

50

60

70

80

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100

10

20

30

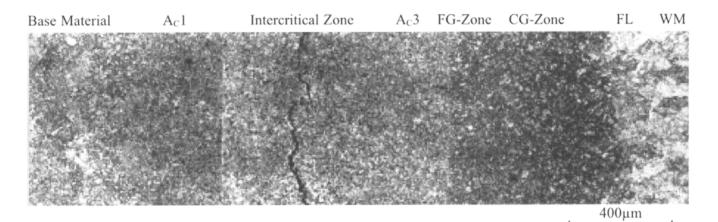


Type IV Cracking—A Problem for Ferritic/Martensitic Steels

- •Type IV cracking has become a major inspection problem in power plants worldwide
- •Evidence suggests cracking results from high stress across weldment and accumulation of creep cavitation damage in intercritically transformed zone—region that was heated between the A_{C1} and A_{C3} temperatures



Type IV Failures—Low Ductility



9Cr-1Mo-1WVNbN (E911) Steel

70 MPa, 650°C, 4295 h



Effect of Irradiation on Weldments Needs to be Evaluated

- •The effect of irradiation hardening at low temperatures on fracture needs to determined –Tensile studies
 - -Impact and fracture toughness studies
 - -Fatigue (high-cycle and low-cycle)
- •The effect of properties of weldments above temperature where hardening occurs needs to be evaluated
 - -Type IV cracking



Alternate Welding Techniques Need to be Evaluated

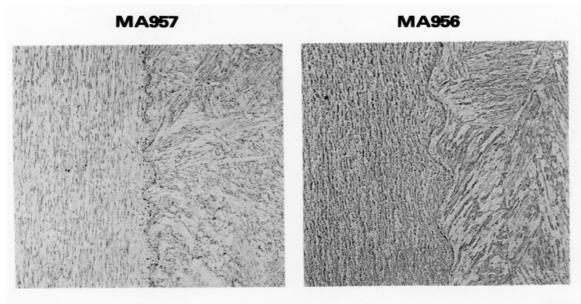
- •High-frequency induction welding
- Diffusion welding
 - –Used on F82H and MANET in fusion materials programs
 - -Leak-tight bonds with strength almost equivalent to base steels were achieved
- •Explosive Welding
 - -Process has been commercially developed
 - –Used to advantage in fast reactor programs
- •Pulsed-Magnetic Welding



Pulsed-Magnetic Welding is Method for Joining ODS Steels

• Pulsed Magnetic Welding Studies for FFTF

- Traditional fusion welding techniques do not work
 - Due to flocculation of Y_2O_3
 - flaring of tubing critical





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Repair Welds on Fusion Plant Will be Affected by Helium in the Steel

- •Limited data exist for HT9 with 0.3 and 1 appm He inserted by "tritium trick" and welded
- •No weld defects on controls or with 0.3 appm He
- •Discontinuous micro-cracks formed at prior austenite grain boundaries in steel with 1 appm He
- •Cracking occurred at high temperatures by shrinkage stresses in constrained plates during cooling from growth and coalescence of grain boundary He bubbles



Processing of Steels: Techniques are Well Established

- Processing of conventional and reducedactivation steels is well established
- •Variation on Processing May Provide Route to New Steels
- •Conventional processing techniques were used to produce A-21 precipitation-strengthened steel



A-21 IS STRENGTHENED BY CONVENTIONAL PROCESSING

- •A-21 steel: Fe-9.5Cr-3Co-1Ni-0.6Mo-0.3Ti-0.07C*
- •Processing of A-21 Steel
 - -austenitized at T>1100°C
 - -hot worked at 700-900 °C
 - -cooled to ambient temperature to form martensite -tempered 1 h at 650-750 °C

*Cobalt-free A-21 steel has been developed

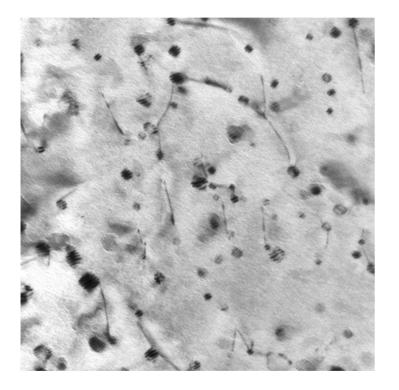


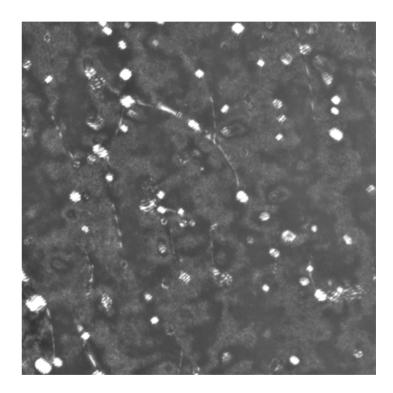
PROCESSING PRODUCES FINE DISPERSION OF TiC

- •Carbides dissolve at austenitization temperature (>1100°C)
- •Hot working at 700-900°C produces dislocations on which a fine distribution of TiC precipitates
- •Cooling to ambient temperature produces lowcarbon martensite (most of the carbon is in TiC)
- •During tempering, which may not be needed, no large $M_{23}C_6$ precipitates form (carbon tied up in TiC precipitates



A-21 CONTAINS HIGH NUMBER DENSITY OF FINE PRECIPITATES





•Particle size: 5-20 nm; Number Density: 4.7 x 10²¹ m⁻³



Welding Issues: Summary and Conclusions

- •Complicated microstructures of conventional welds in ferritic/martensitic steels affects fracture behavior
- •Effect of irradiation on weld failures needs to be established for reduced-activation steels
- •Alternate welding techniques need to be explored
- •Welds on steel containing helium need investigation to develop repair welding techniques



Processing Issues: Summary and Conclusions

- •Conventional processing techniques for conventional and reduced-activation steels are well established
- •Variation on the processing of steels may offer another route to steels with good elevatedtemperature strength

